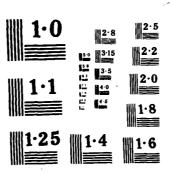
AD-A160 601	INTERACTION FATION A PETROVI	VE EFFE	TS OF	HIGH-	AND LO	W-FREQ	MENCY	LOADING N Y	17	2	
UNCLASSIFIED		-C-5056	95 811	-03164	O AFBA	IN - 63	F/G	20/11	NL		
	•										
		-			4				_		
										41	
							<u> </u>				
	1		==								
			<b>'</b>								<u> </u>
											$\prod$
		10.01									Ш
	ĺ	[									
											<u> </u>
								Н			





AFWAL-TR-85-4045

INTERACTIVE EFFECTS OF HIGH- AND LOW-FREQUENCY LOADING ON FATIGUE

MECHANICAL TECHNOLOGY INCORPORATED 968 ALBANY-SHAKER ROAD LATHAM, NEW YORK 12110

May 1985

Final Report for Period September 1982 - December 1984

Approved for Public Release, Distribution Unlimited

MATERIALS LABORATORY
AF WRIGHT AERONAUTICAL LABORATORIES
AIR FORCE SYSTEMS COMMAND
WRIGHT-PATTERSON AFB, OHIO 45433

#### NOTICE

When Government drawings, specifications, or other data are used for any purpose other than in connection with a definitely related Government procurement operation, the United States Government thereby incurs no responsibility nor any obligation whatsoever; and the fact that the government may have formulated, furnished, or in any way supplied the said drawings, specifications, or other data, is not to be regarded by implication or otherwise as in any manner licensing the holder or any other person or corporation, or conveying any rights or permission to manufacture use, or sell any patented invention that may in any way be related thereto.

This report has been reviewed by the Office of Public Affairs (ASD/PA) and is releasable to the National Technical Information Service (NTIS). At NTIS, it will be available to the general public, including foreign nations.

This technical report has been reviewed and is approved for publication.

DAVID I.G. JONES, Project Engineer

Metals Behavior Branch

Metals and Ceramics Division

JOHN P. HENDERSON, Chief Metals Behavior Branch

Metals and Ceramics Division

FOR THE COMMANDER

LAWRENCE N. HJELM, Asst Chief Metals and Ceramics Division

Materials Laboratory

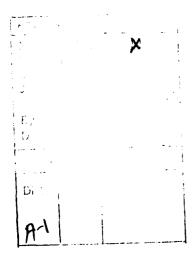
"If your address has changed, if you wish to be removed from our mailing list, or if the addressee is no longer employed by your organization please notify AFWAL/MLLN W-PAFB, OH 45433 to help us maintain a current mailing list".

Copies of this report should not be returned unless return is required by security considerations, contractual obligations, or notice on a specific document.

	REPORT DOCUME	NTATION PAGE				
14 REPORT SECURITY CLASSIFICATION	16. RESTRICTIVE MARKINGS					
Unclassified 24 SECURITY CLASSIFICATION AUTHORITY	None					
28. SECURITY CLASSIFICATION AUTHORITY	3. DISTRIBUTION/AVAILABILITY OF REPORT					
26 DECLASSIFICATION/DOWNGRADING SCHED	Approved f					
		distributi	on unlimit	2d		
4 PERFORMING ORGANIZATION REPORT NUM	BER(S)	5. MONITORING OR	GANIZATION R	EPORT NUMBER	3)	
85TR48		AFWAL-TR-85-4045				
6a. NAME OF PERFORMING ORGANIZATION	6b. OFFICE SYMBOL (If applicable)	7a. NAME OF MONIT	ORING ORGAN	IZATION		
Mechanical Technology Inc.		Materials Laboratory				
6c. ADDRESS (City, State and ZIP Code)	<u> </u>	7b. ADDRESS (City,	State and ZIP Cod	le)		
968 Albany-Shaker Road		AFWAL/MLLN				
Latham, New York 12110		WPAFB, Ohi				
	In			<del> </del>		
8a. NAME OF FUNDING/SPONSORING ORGANIZATION	8b. OFFICE SYMBOL (If applicable)	9. PROCUREMENT I	NSTRUMENT ID	ENTIFICATION N	JWREH.	
Materials Laborotory	AFWAL/MLLN	Contract F	33615-82-C	-5056		
8c. ADDRESS (City, State and ZIP Code)		10. SOURCE OF FUN	IDING NOS.			
Wright-Patterson AFB, Ohio 4	5433	PROGRAM ELEMENT NO.	PROJECT NO.	TASK NO.	WORK UNIT	
		ł	:			
11 T:TLE (Include Security Classification)		62102F	2420	01	24200135	
Interactive Effects of HCF/LC	F (U) see cover		·		<u>_</u>	
12 PERSONAL AUTHOR(S) A. Petrovich						
13a. TYPE OF REPORT 13b. TIME C	14. DATE OF REPOR	RT (Yr., Mo., Day	15. PAGE C	OUNT		
Technical Final FAOM 1 Se	1985 May		148			
16. SUPPLEMENTARY NOTATION				_		
17 COSATI CODES	19 SUBJECT TERMS (C					
FIELD GROUP SUB. GR.	18. SUBJECT TERMS (C	ontinue on reverse if ne	cessary and identi	Ty by block number	'	
1	1					
19. ABSTRACT (Continue on reverse if necessary and	l identify by block number	•,				
\	1		,	1 1 .1		
This report describes th						
mechanisms of fatigue and c material under high frequen						
to provide a basis for dama						
bined high and low frequence	v loading.	ign of affectar	- digine e		'der com	
T		furria: Do	v <b>Z</b> [ ],vk	4 (die	<i>) ;</i>	
rack propagation	like I Middle	/s	i. (* )		,	
Sock propagation life Appellance (Some life).						
			,			
20 DISTRIBUTION/AVAILABILITY OF ABSTRAC	CT .	21. ABSTRACT SECURITY CLASSIFICATION				
UNCLASSIFIED/UNLIMITED 🖾 SAME AS RPT.	DTIC USERS	Unclassifie	d			
22. NAME OF RESPONSIBLE INDIVIDUAL		22b. TELEPHONE NU		22c. OFFICE SYM	BOL	
Dr. D.I.G. Jones		(513) 2-5		AFWAL	MLLN	
<u> </u>		<u> </u>		L		

# **FOREWORD**

This report was prepared by Mechanical Technology Incorporated (MTI), Latham, New York for the Metals Behavior Branch, Metals and Ceramics Division, Materials Laboratory, Air Force Wright Aeronautical Laboratories, (AFWAL/MLLN), Wright-Patterson Air Force Base, under Contract F33615-82-C-5056. The work was administered under the direction of Dr. David I.G. Jones, AFWAL/MLLN. The program effort was conducted at MTI by A. Petrovich, W.F. Bessler and W.H. Ziegler. Additional support was provided by J. Walton, L.A. Peterson and L. Isley. This report describes work conducted from September, 1982 to December, 1984.





# TABLE OF CONTENTS

SECTION		PAGE
I	INTRODUCTION	1
II	SYSTEM CONSTRUCTION, EVALUATION OF SYSTEM DYNAMICS, AND SPECIMEN DESIGN	5
	A. Background On the Selection of Test Equipment	5
	B. System Selection and Construction	13
	C. Dynamic Evaluation of Test Specimens	21
	C.1 Modal Analysis of the Preliminary Specimen	24
	C.2 Strain Gage Evaluation of a Specimen with Lateral Reinforcement and Damping	33
III	EXPERIMENTAL TEST PROGRAM	47
	A. Fatigue Crack Growth Studies Conducted at 200 Hz $$	49
	B. Results of Combined Cycle Tests With an 1800 to 2000 Hz High Cycle Component and Comparison With Lower	
	Frequency Results	56
ΙΛ	EVALUATION OF MECHANISMS AND MODELLING ASSOCIATED WITH FREQUENCY EFFECTS AND COMBINED HIGH/LOW CYCLE INTERACTION	68
	A. Background	68
	B. Evaluation of Fatigue Crack Growth Mechanisms Under Combined Cycle Loading	82
	C. Consideration of High/Low Cycle Interactions in Crack Growth Life Prediction of Engine Systems	93
v	CONCLUSIONS	98
BIBLIOGRA	APHY	100
APPENDICE	ZS .	
	APPENDIX A: PERFORMANCE OF HIGH FREQUENCY SERVO-HYDRAULIC SYSTEM	103
	APPENDIX B: DATA PLOTS FOR ALL TESTS	107
	APPENDIX C: DATA LISTINGS FOR ALL EXPERIMENTS	131



# LIST OF TABLES

TABLE		PAGE
1	Specification for Instron Combined Cycle Test System	6
2	Natural Frequencies and Damping Factors for Resonant Modes .	29
3	Test Program Outline	48
4	Combined Cycle Tests Including a 200 Hz High Cycle Frequency	50
5	Combined Cycle Tests Including an 1800 to 2000 .Iz High Frequency Load Completed to Date	62
6	Conditions for the Onset of Minor Cycle Damage and Onset of Fast Fracture ( $\Delta$ K Values in MPa $\sqrt{m}$ )	80

FIGURE		PAGE
1	Schematic of Major/Minor Cycling Machine Control System .	7
2	Hydraulic Actuator and Servo-Valve	9
3	Standard Flapper-Nozzle Servo-Hydraulic Valve	10
4	Estimated Maximum Displacement Versus Frequency for the Akashi Servo-Valve Servo-Actuator Combination HV 10.0-1.2-37/5	11
5	Vibration Patterns for a Center Crack Panel Specimen	12
6	High-Frequency Servo-Hydraulic Test System with 2-inWide Center-Cracked Panel Specimen Installed	15
7	Center-Cracked Panel Specimen in High-Frequency Test Syste	m 16
8	Crack Length Measurement and Servo-Control System for High-Frequency Servo-Hydraulic System	17
9	Closed-Loop Control System for High- and Low-Frequency Components of Loading Profile	18
10	Load Cell Measurement of Load with High-Frequency Signal of 460 Hz	20
11	Photographs of the Preliminary Specimen Assembled and Disassembled	22
12	Drawings Showing the Dimensions of the Preliminary Specimen	23
13	Drawings Showing the Dimensions of the Preliminary Specimen Subjected to Modal Analysis	26
14	Relative Input Power Spectrum Used in the Modal Analysis .	27
15	Accelerometer Locations Used in the Modal Analysis	28
16	Mode Shapes for Case I: Mean Load 2000 lbs., Crack Total Length (2a) of 0.200 inches	30
17	Mode Shapes for Case II: Mean Load 4500 lbs., Crack Total Length (2a) of 0.200 inches	31
18	Mode Shapes for Case III: Mean Load 2000 lbs., Crack Total	32

FIGURE		PAGE
19	Specimen with End Reinforcement and Lateral Constraints Applied to the Crack Region	35
20	Diagram Showing Location of Damping Blocks and Glass Insulating Material	36
21	Dimensions of Specimen End Clamps	37
22	Diagram Showing Location of Strain Gages	38
23	Magnitude of Stress Versus Frequency in Locations 1 and 2 for Laterally Damped and Reinforced Specimen	39
24	Magnitude of Stress Versus Frequency in Locations 1 and 7 for Laterally Damped and Reinforced Specimen	40
25	Magnitude of Stress Versus Frequency in Locations 1 and 2 for Laterally Damped Specimen with Compression Rings and Load Cell Removal from System	42
26	Series of Oscilloscope Representations of Strain Gage Output #1 Versus #2 Showing the Degree of Bending and Out of Phase Vibration on the Specimen Crack Centerline for a Mean Load of 4000 lbs. and a Crack Length of .200"	
27	Series of Oscilloscope Representations of Strain Gage Output $\#1$ Versus $\#2$ Showing the Degree of Bending and Out of Phase Vibration on the Specimen Crack Centerline for a Mean Load of 2000 lbs. and a Crack Length of 0.200"	
28	Series of Oscilloscope Representations of Strain Gage Output #1 Versus #2 Showing the Degree of Bending and Out of Phase Vibration on the Specimen Crack Centerline for a Mean Load of 2000 1bs. and a Crack Length of 0.700"	
29	Characteristics of the High/Low Frequency Interaction Showing the Three Types of Behavior Observed in this Study. The Points Correspond to Testing with a Low Frequency $\Delta K$ of 20 MPa $\sqrt{m}$ , a Low Cycle Time of 10 Seconds and a High Cycle Frequency of 200 Hz	51
30	Results of Combined High/Low Frequency Test with a Low Cycle △K of 15 MPa √m and a Low Cycle Hold Time of 5 Seconds	53
31	Results of Combined Cycle Test with a Low Frequency △K of 20 MPa √m and a Hold Time of 5 Seconds. The Line is Drawn to Show the Sequence of Points	

FIGURE		PAGE
32	Results of a Combined Cycle Test with a Low Frequency $\Delta K$ of 30 MPa $\sqrt{m}$ and a Hold Time of 5 Seconds	55
33	Results of a Combined Cycle Test with a Low Cycle $\Delta K$ of 40 MPa $\sqrt{m}$ and a Hold Time of 5 Seconds	57
34	Comparison of Crack Growth Rate Versus High Cycle $\Delta K$ for Several Hold Times with a Low Cycle $\Delta K$ of 15 MPa $\sqrt{m}$	58
35	Comparison of Crack Growth Rate Versus High Cycle $\Delta K$ for Several Hold Times and a Low Cycle $\Delta K$ of 20 MPa $\sqrt{m}$	59
36	Comparison of Crack Growth Rate Versus High Cycle $\Delta K$ for Several Low Cycle $\Delta K$ Ranging from 15 to 40 MPa $\sqrt{m}$ with a Low Cycle Hold Time of 5 Seconds	60
37	Comparison of Crack Growth Rate Versus High Cycle $\Delta$ K for Several Low Cycle $\Delta$ K Ranging from 15 to 40 MPa $\sqrt{m}$ with a Low Cycle Hold Time of 180 Seconds	61
38	Comparison of Results for 200 and 1825 Hz for a Hold Time of 5 Seconds and a Low Cycle $\Delta$ K of 30 MPa $\sqrt{m}$	63
39	Comparison of Results for 200 and 1825 Hz for a Hold Time of 5 Seconds and a Low Cycle $\Delta$ K of 20 MPa $\sqrt{m}$	65
40	Comparison of Results for 10 and 200 Hz for a Hold Time of 180 Seconds and a Low Cycle $\Delta$ K of 30 MPa $\sqrt{m}$	
41	The Effect of Frequency on the Number of Cycles and Time to Failure of <b>U</b> -700 at 1400°F (760°C) and a Stress Range of 85 Ksi	70
42	The Percentage of Stage I Fracture in the Fatigue Zone as a Function of Cyclic Frequency at Temperatures of 1033, 1116, 1200 and 1255°K	72
43	Variation of FCG Rate (da/dN) with Stress Intensity Factor ( $\Delta$ K) and Frequency ( $\gamma$ ) at 823°K for Inconel 718 (Sinusoidal Load)	72
44	Schematic Comparison of the Air and Vacuum Crack Growth Behavior	74
45	Effect of Amplitude Ratio on Fatigue Crack Growth (FCG) Rate of Major and Minor Cycles	78

FIGURE		PAGE
46	Effect of Amplitude Ratio on Fatigue Crack Growth (FCG) Rate of Major and Minor Cycles	79
47	Linear Summation of FCG Rates (Damage A-Associated with Applied Major Cycle; B-Associated with Applied Minor Cycles; C-Given by Summation of Major and Minor Cycle Damage)	81
48	Analysis of Major-Minor Fatigue Crack Growth Rates in Terms of $\Delta K_{\mbox{RMS}}$	81
49	Scanning Electron Microscope (SEM) Photomicrograph of a Region of Specimen #28 in Which Only Low Cycle Loading was Applied	83
50	Scanning Electron Microscope (SEM) Photomicrograph of a Region of Specimen #28 in Which Combined Cycle Loading (with a 200 Hz High Cycle Load) was Applied	83
51	Scanning Transmission Electron Microscope (STEM) Photomicrographs of a Region in Which Only Low Cycle Loading was Applied	85
52	Scanning Transmission Electron Microscope (STEM) Photomicrographs of a Region in Which Combined Cycle Loading (with 200 Hz High Cycle Load) was Applied	87
53	STEM Photomicrographs of a Region on the Fracture Surface of Specimen #67 Corresponding to the Low Cycle Dominated Regime Where the High Cycle AK is Large Enough to Cause Retardation	89
54	STEM Photomicrographs of a Region on the Specimen #67 Fracture Surface Where the High Cycle Component Dominates Crack Growth	90
55	Comparison of Data (Points) with Growth Rate Predicted (Line) from a Linear Summation of Uncycled 200 Hz High Cycle Data and Pure Low Cycle Data	96
56	Comparison of Data (Points) with Growth Rate Predicted (Line) from a Linear Summation of Uncycled 1825 Hz High Cycle Data and Pure Low Cycle Data	97

## I. INTRODUCTION

A means of more fully utilizing the useful life of aircraft engine components is provided by the Retirement-for-Cause (RFC) life management concept. Under the RFC philosophy, components are inspected at intervals of operation such that a crack or other service induced defect just below the level of detectability cannot grow to a critical size between inspections. Those components with no observable flaw are returned to service with the assurance that if a fatigue crack develops, it will not grow to a size that will result in catastrophic failure of the component while in service. A cost savings results from the fact that by retiring the components on the condition of an observable flaw, components without flaws that would otherwise be retired by the probabilistic scheme would now be allowed to remain in service until cracking is apparent.

Implementation of the Retirement-for-Cause method requires advances both in non-destructive evaluation and crack growth life prediction. The primary requirement with regard to crack growth life prediction is an improvement in the accuracy of life prediction for the complex loading profile experienced by engine components. Important to the improvement of the accuracy of life prediction is an accounting of the interaction between the various components of the loading profile.

The life limiting loading profile experienced by an engine disk consists basically of a group of low frequency cycles associated with thermal gradients or centrifugal forces and superimposed high frequency loading associated with blade passage. The cycle period associated with the low frequency cycle (low cycle) loading is on the order of seconds to several hundred seconds. A wide range of loading rates and load levels may also be involved in the low cycle loading. The high frequency cycle (high cycle) loading would typically involve frequencies on the order of hundreds to several thousand hertz. Important to accurate life prediction is establishing the manner in which each of these features of the engine disk loading profile contribute to crack growth and how these features interact. The specific aspects of combined cycle loading that must be addressed are the following:

- Establishment of the limits of high cycle loading under which the disk can be safely operated.
- How cumulative damage rules should be applied when combined high cycle/low cycle loading contribute to crack growth.
- The degree to which the high cycle and low cycle loading influence each others contribution to crack growth.

A test system was designed and constructed specifically for the present study to provide adequate load levels up to 2000 Hz and minimize the frequency ranges over which dynamic complications in load application are present. A purely servo-hydraulic system based on an Akashi voice-coil servo-valve was used for all of the testing. The load frame and specimen were designed to minimize the number of system resonances that create undesirable specimen stress patterns and either complicate or invalidate the representation of stresses around the specimen crack. The test system constructed for their study is described in Section 2.

The specimen type used for this study was a center crack panel also described in Section 2. A clevis arrangement with provisions to clamp the specimen ends was used to grip the specimen. By securely clamping the specimen and providing additional lateral support, specimen resonances could be avoided at the selected test frequencies. Both the high frequency and low frequency was sensed by a load cell. It was recognized that resonances in the load frame and specimen could disturb the correlation between the load cell measurement and stresses in the specimen as well as provide significant bending stresses associated with resonant lateral vibration. Modal analyses of a preliminary specimen were performed to determine its natural frequencies and mode shapes over a range of steady load and crack length. These modal analyses indicated the specimen modifications required to make the specimen suitable for testing in bands of frequencies up to 2000 Hz. Test frequencies of 200 and 1825 Hz were chosen for most of this study. The absence of excessive bending stresses and a proper correlation between load cell measurement and specimen stresses was verified at these frequencies with strain gage measurement on the specimen. The precision in the high frequency  $\Delta K$  measurement required for this study made these specimen dynamic evaluations and detailed verification of specimen stress absolutely essential.

The high/low frequency loading profile used in this study is described in Section 2. The low frequency component was a trapezoidal waveform with a rise time  $(T_1)$  and fall time  $(T_2)$  of 0.5 seconds and with a hold time  $(T_0)$  of between 2 and 180 seconds. The high frequency loading was applied during the low frequency cycle hold period and typically ranged between 220 and 4450 Newtons (50 to 1000 lbs). The low frequency load levels  $P_1$  and  $P_2$  were varied during the tests such that the low frequency stress intensity factors  $K_1$  and  $K_2$  were maintained constant. The high frequency load range  $(P_0)$  was either increased during the test or maintained constant which in either case resulted in an increasing high frequency stress intensity factor range  $(K_0)$ . The low cycle R ratio  $(P_1/P_2)$  was 0.1 for all of the testing. All testing was performed at  $649^{\circ}C$ .

Section 3 presents the results of a series of crack growth tests that were performed on Inconel 718 at 649°C (1200°F). The following aspects of the high/low cycle interaction were investigated in the series of crack growth tests:

- the effect of high cycle frequency up to 2000 Hz on the low cycle/high cycle interaction
- the effect of low cycle stress intensity factor range ( $\Delta K_{LC}$ )
- variation of crack growth rate as a function of high cycle stress intensity factor range ( $\Delta K_{HC}$ ) over a  $\Delta K_{LC}$  range of 15 to 40 MPa  $\sqrt{m}$
- the influence of low cycle hold time between 2 and 180 seconds on the crack growth rate under combined cycle loading

The results of testing are summarized in curves representing crack growth rate versus high frequency  $\Delta K$  for constant low frequency cycle  $\Delta K$  range and low cycle hold time. Crack growth rate is reported in terms of growth per unit time at the upper level of the low cycle trapezoidal loading profile. The low cycle  $\Delta K$  ranges included in the testing were 15, 20, 30 and 40 MPa  $\sqrt{m}$ . The low cycle hold times included 2, 5, 10 and 180 seconds. Comparisons are made in Section 3 between the results for a high cycle frequency of 200 and 1825 Hz provided by this study and those for 10 Hz provided in Reference 1.

Section 4 also explores possible mechanisms associated with the combined cycle interaction. Correlations are made between the crack growth data and features

of the fracture surface. Means of .cdelling combined cycle crack growth rate are also discussed.

# II. SYSTEM CONSTRUCTION, EVALUATION OF SYSTEM DYNAMICS, AND SPECIMEN DESIGN

## A. Background on the Selection of Test Equipment

There have been several approaches to providing controllable load levels in the frequency regime above 100 Hz. One system used for high frequency fatigue testing is the electrodynamic shaker. Motion and forces in these systems are generated by interaction of a solenoid generated field with a moveable armature. Standard commercial electrodynamic shakers have force ratings up to 9000 lbs. with up to 100 g's of acceleration available to 3000 Hz. (2) Magnetrostrictive devices have also been used to generate forces and displacements of frequencies up to 1000 Hz. An example of a study of threshold crack growth conducted with magnetostrictive system is that of Reference 3.

The desire to test materials with a loading profile similar to that of an aircraft turbine engine has lead to the development of test systems that can apply low cycle high amplitude loading (on the order of 5000-20,000 lbs.) along with low amplitude high frequency (100 to 2000 lbs.) loading. A novel example of the machines developed for the application of combined cycle loading is the major/minor cycling system constructed by Instron Ltd. (4) The characteristics of this system are summarized in Table 1 and a schematic representation of the system appears in Figure 1. The high frequency loading component is applied by the electrodynamic shaker and the low frequency component by a hydraulic actuator. This is made possible by a specially developed isolation unit between the hydraulic actuator and shaker which allows the simultaneous application of loading by the hydraulic actuator and electrodynamic shaker.

An alternative to a combined servohydraulic/electrodynamic and purely electrodynamic system for the application of a combined high cycle/low cycle loading profile is a purely servohydraulic system based on a voice coil servo-valve. (5) The use of electrohydraulic servo-valves for material testing is widely practiced and is usually performed with a flapper-nozzle valve, which, in spite of its inherent low-frequency limitation, has been totally adequate for the testing of such material properties as creep, ultimate strength, yield strength, and low-frequency cyclic fatigue.

### TABLE 1

### SPECIFICATION FOR INSTRON COMBINED CYCLE TEST SYSTEM

**High Frequency Component:** 

Waveform:

Sinusoidal

Frequency Range: Max. Dynamic Load: 50 - 600Hz depending upon the specimen stiffness

±5kN

Low Frequency Component;

Waveform:

Trapezoidal

Minimum rise and fall times:

0.4 sec.

Dwell times:

0.1 - 99.9 secs. or 0.1 - 99.9 min.

Maximum Load Unidirectional tension or compression:

50kN

Load Frame:

**Number of Columns:** 

**Dynamic Load Rating:** 

±250kN

Max. Vertical Daylight: (between load cell & shaker)

700mm

Distance between columns:

661 x 305

Load Cell:

Fatigue Rating (Unidirectional):

Excitation:

50kN max. force

Load Measurement Accuracy Static:

5.6 volt DC ±1% of indicated force or ±0.2% of full scale, whichever

is the greater

Dynamic:

±3% of indicated force or ±0.2% of full scale, whichever

is the greater

Compensation is provided for changes in dynamic load reading caused by the mass of the Grip or Fixture.

### NOTE:-

Because of the high operating frequencies, the mass of the moving parts has a significant effect on the performance of the machine. The actual frequency range over which the desired dynamic force can be achieved is dependent upon the stiffness of the specimen. Details of the specimen should be given when ordering.

**Patents Pending** 

Instron Limited reserves the right to change details and specifications without notice

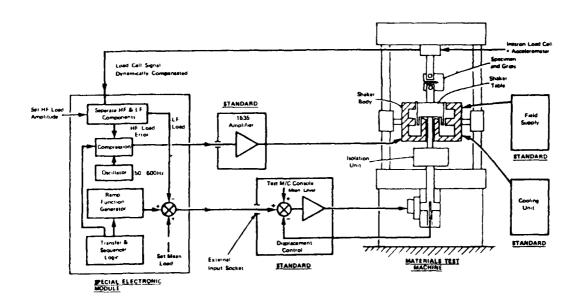


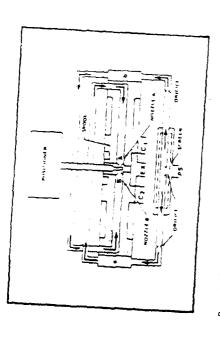
Figure 1 Schematic of Major/Minor Cycling

Machine Control System (4)

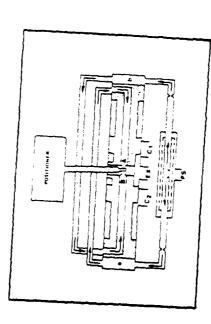
An example of a high frequency servo-hydraulic system is that developed by Akashi Ltd. (5) The Akashi vibration system employs a servo-valve and actuator with high-frequency capability based on a voice-coil-type servo-valve. In this system, the electrical drive signal directly causes servo-valve spool motion. The voice-coil valve thereby provides a significant advantage in high frequency input flow capability to the actuator. Also the Akashi servo-valve is optimized to reduce the impedance loading associated with high frequency, and its mechanical natural frequency has been established to favor frequencies at the higher end of its useful spectrum. The two types of servo-valves, the flapper-nozzle system and Akashi voice-coil system, are shown in Figures 2 and 3 respectively. Additional high frequency servo-hydraulic systems are manufactured by MTS (6) and Teem.

The ability of a test system to provide adequate displacement at the test frequency is the most important consideration for high frequency testing. It is difficult to accurately estimate the displacement capability of a shaker in specimen fatigue testing application in view of the complexity of the interaction between the actuator and load frame. A means of establishing a rough estimate is to determine the maximum deflection capability of the shaker with a test load of 50 lbs. and without the constraint of a load frame. Such a determination was made for an Akashi test system based on a pilot/slave servo-valve with a 5 gpm/37gpm flow capability and a 1.2 inch stroke actuator. The estimated deflection of this system is shown in Figure 4. This curve provided an adequate estimate of deflection for planning fatigue testing in a test system based on this servo-valve/actuator combination.

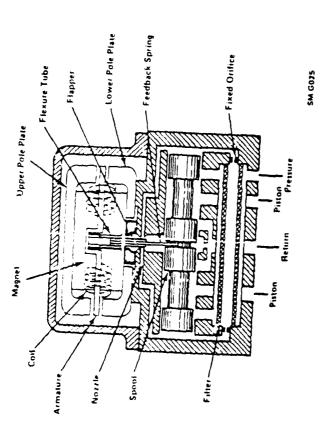
Another important consideration for fatigue crack growth testing is ensuring that the stresses around the growing crack are similar to those under quasistatic conditions. Only if this can be assured can results from one frequency be compared to another. Resonance in the load frame and specimen often disturb the patterns of stresses. Investigations of the manner in which specimen stresses can be distorted at high frequency have been carried out with the aid of modal analysis. Figure 5 shows the patterns of standing wave resonant vibration that can occur in a standard center crack panel specimen. Such dynamic complications to crack growth testing can be minimized by proper selection and design of the test system along with proper design of the specimen.



Boost System - Steady State with Spool On Center



Boost System - Steady State with Spool Off Center



Cross-Sectional View of 252 Servo Valve

Figure 2 Standard Flapper-Nozzle Servo-Hydraulic Valve

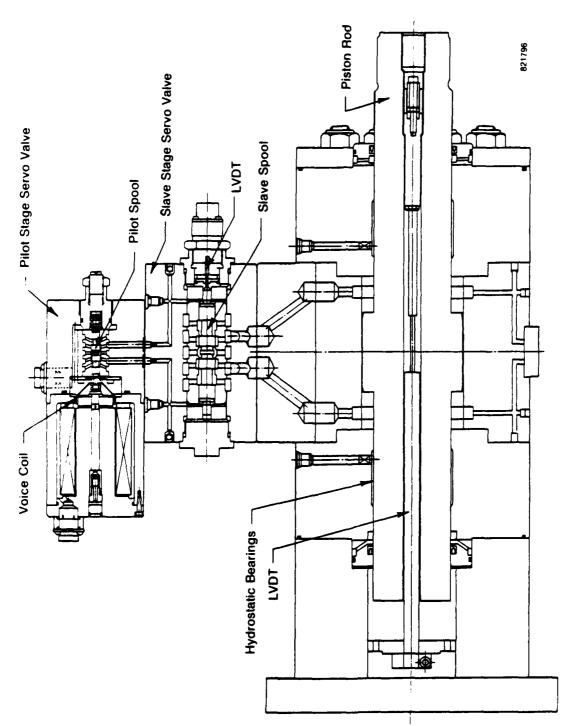


Figure 3 Hydraulic Actuator and Servo Valve (5)

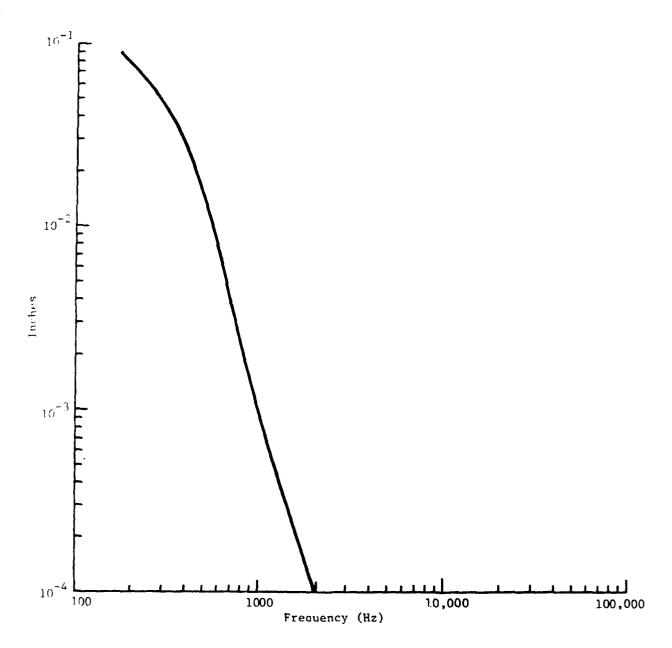
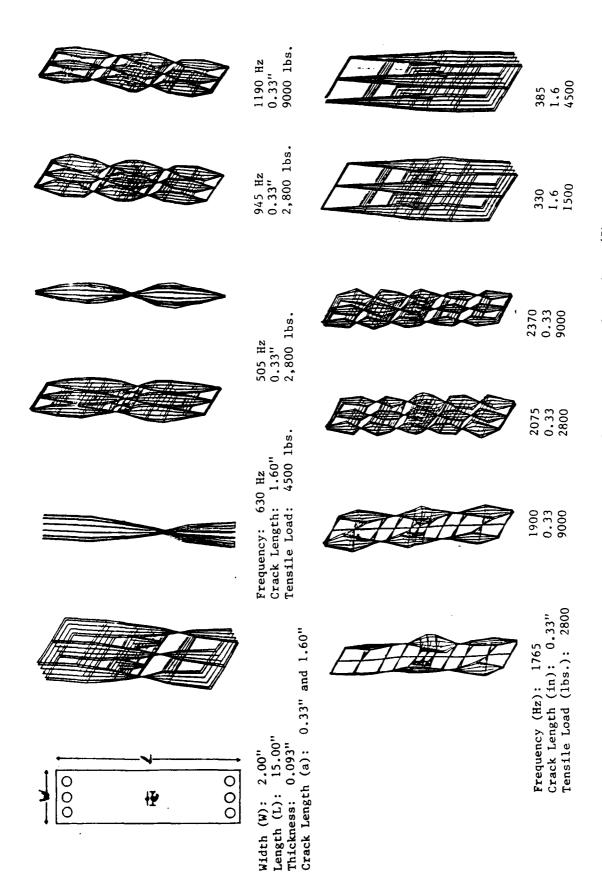


Figure 4 Estimated Maximum Displacement versus Frequency for the Akashi Servo-valve servo-actuator combination HV 10.0-1.2-37/5.

821644



Pigure 5 Vibration patterns for a center panel specimen (7)

# **B.** System Selection and Construction

The study of the phenomenon of high-frequency/low frequency load interaction in fatigue and creep crack growth required the construction of a test system that could provide adequate levels of load up to a frequency of 2000 Hz that could provide low cycle load levels up to 10,000 lbs., and allow precise control of both low-frequency and high-frequency loading during fatigue crack growth testing. The following options were available.

- A purely electrodynamic system with a maximum force rating on the order of 10,000 lb.
- A combined high-frequency electrodynamic/low-frequency servo-hydraulic system with some type of isolation system to remove large preloads from the electrodynamic system during high-frequency vibration.
- A purely servo-hydraulic system based on a voice-coil servo-valve.

A preliminary evaluation indicated that the purely electrodynamic and purely servo-hydraulic options would permit a load frame sufficiently rigid to preclude dynamic complications at the higher frequencies. It was also determined that a servo-hydraulic system was less expensive for the load ranges required for this program. It was feared that the combined servo-hydrualic and electrodynamic system with its isolation system had such an extended load frame that a large number of resonances would make testing and verification of loading extremely difficult particularly at higher frequencies. Therefore, because of compatibility with existing MTI equipment, relatively low cost, and ability to construct compact frames and fixturing around the specimens and actuator, the purely servo-hydraulic option was chosen.

The purely servo-hydraulic test system constructed for this program is based on an Akashi voice-coil servo-valve with a frequency capability that far exceeds the more conventional flapper-nozzle servo-valve. The Akashi servo-actuator incorporates all the features of the fatigue-rated actuators currently used today, as well as several key design improvements. The piston-to-cylinder clearance is manufactured to allow operation without a piston seal, thereby eliminating a wear item and, more importantly in high-frequency operation, removing the cause of waveform distortion associated with seal motion during pressure reversal. Hydrostatic bearings are employed to provide high side load

capability and to eliminate any metal-to-metal contact at the bearings. Piston rod seals are not used, thus eliminating the largest factor of friction in the system and also removing an element that must be periodically serviced, i.e., replacing the seal and refinishing the piston rod.

The high-frequency servo-hydraulic equipment purchased by MTI for this program includes the following:

- Akashi Servo-Actuator Model HV 10.0-1.2-37/5 (10,000 lb. maximum status load
- Akashi Servo-Valve (pilot/slave) Model SV 5/SV 37
- \* Akashi Servo-Controller Model SC-1
- · Akashi Servo-Amplifier Model SA-400
- Akashi Manifold Model HM-40

The Akashi servo-actuator (Model HV 10.0-1.2-37/5) was installed in a load frame already in operation at MTI. Figure 6 shows the servo-hydraulic testing system with specimen, induction heating coil, and crack-length-measuring telemicroscope in place. Figure 7 shows, in greater detail, the specimen, induction heating coil and specimen gripping arrangement. A standard clevis was used to provide loading to the specimen. To reduce unnecessary deflection, the 2-in.-wide center cracked panels are reinforced at the ends gripped by the clevis.

During the high-frequency experiments, noise levels reached 135 db, and it was necessary to construct a sound-deadening enclosure around the system. The noise reduction provided by this enclosure was sufficient to reduce the noise to an acceptable level in surrounding offices and work areas.

The high-frequency servo-valve receives signals from the servo-controller and servo-amplifier. The command signal applied to the servo-controller is the sum of a high- and low-frequency signals which are controlled independently by the PDP-11/04 computer that interfaces the servo-hydraulic system. A schematic diagram of the control and data acquisition system is shown in Figure 8. The high- and low-frequency signals must be controlled independently becasue the system gain (i.e., the load range per unit input signal) is different for the high- and low-frequency portions of the command signal. Figure 9 schematically

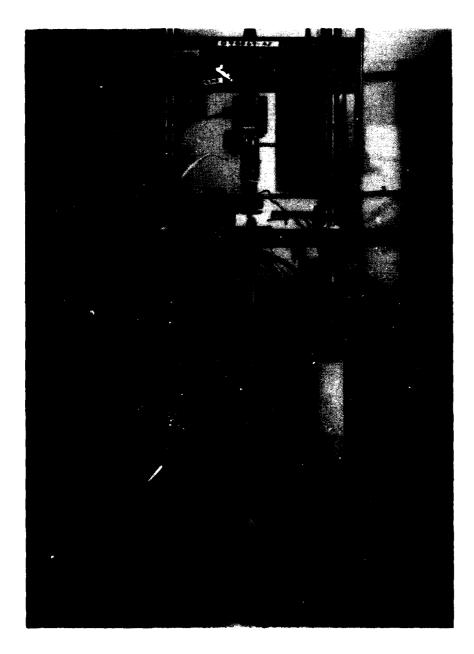
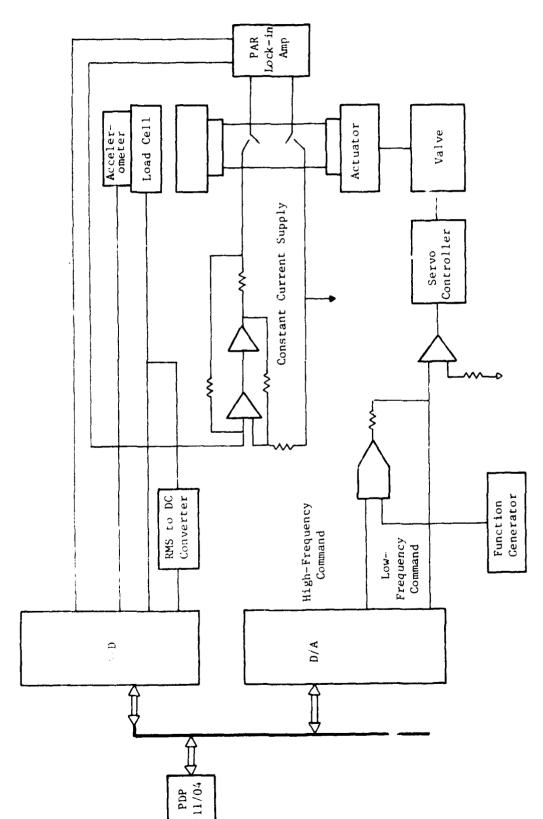


Figure 6 High Frequency Servohydraulic Test System with 2" Width Center Cracked Panel Specimen Installed



Figure 7 Center-Cracked Panel Specimen in High-Frequency Test System (The specimen is heated inductively, and the crack length is monitored by both a 20X telemicroscope and an AC direct potential crack-measurement system.)



Grack Length Measurement and Servo Control System for High-Frequency Servo Hydraulic System (Based on the Akashi Servo Controller and Actuator with Latest Modification) Figure 8

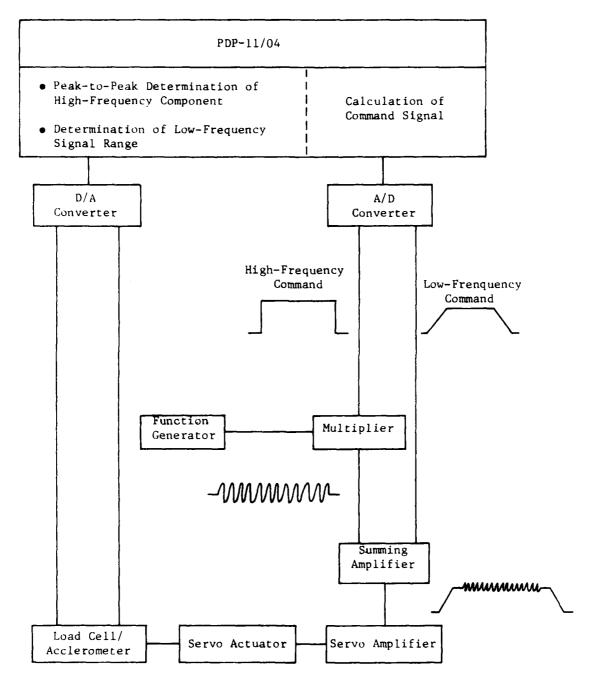


Figure 9 Closed-Loop Control System for High- and Low-Frequency Components of Loading Profile

illustrates the control of the high- and low-frequency signals. The loading profile resulting from the summation of the high- and low-frequency components is shown in Figure 10a which corresponds to the load response of the system with the 2-in.-wide center-cracked panel and a high-frequency component of 460 Hz. The high-frequency segment of the load cell signal is shown in greater detail in Figure 10b.

A servo-hydraulic vibration system test report for the Akashi servo-valve and servo-actuator is included in Appendix A. The first chart in this report shows the maximum performance in terms of acceleration for the Akashi servo-actuator used in this test program. This chart provides the acceleration as a function of frequency for a 60.75 and 108.7 kg payload. The second and third charts in this report show the acceleration as a function of frequency for a specific signal level input to the servo-amplifier for a 60.75 kg and 108.7 kg payload respectively. Such charts are useful in estimating the actuator displacement capability in a materials testing load frame.

The basic types of specimens used in crack growth testing are the compact tensile specimen, the edge-cracked-panels, the center-cracked panel and bend-type specimens, plus many variations on these basic types. For high-frequency testing, several important dynamic considerations have to be addressed if the desired loading conditions are to be achieved at the crack. The most important consideration is the compliance (extension per unit load) for the range of  $\Delta K$  (K = stress intensity factor) to be covered in the test program. The specimen has to provide as low a compliance as possible consistent with accurate measurement of crack length and of crack growth rate. (Minimizing compliance is important because, as frequency increases, the possible deflection of both servo-hydraulic and electrodynamic systems rapidly decrease). Therefore, the center-cracked-panel type was chosen for use in this program because of it relatively low compliance for a given  $\Delta K$  range.

In the planning of the test program, frequency ranges for the high and low-frequency  $\Delta K$  were selected. frequencies. A specimen and loading system were then selected to accomplish testing in these ranges. The dynamic limitations of the system had to be considered. For example, in the frequenc, regime of 1000 Hz and above, the maximum displacement that a servo-hydraulic system can provide is critically dependent on the design of the servo-valve and servo-actuator. Simi-

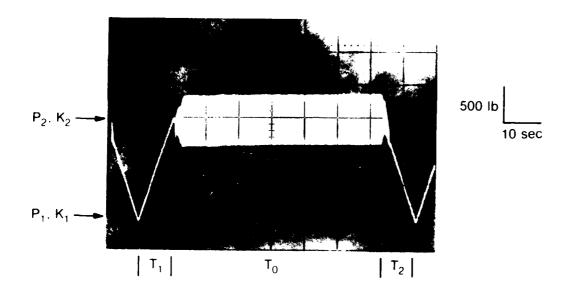


Figure 10a Combined high/low frequency loading profile.

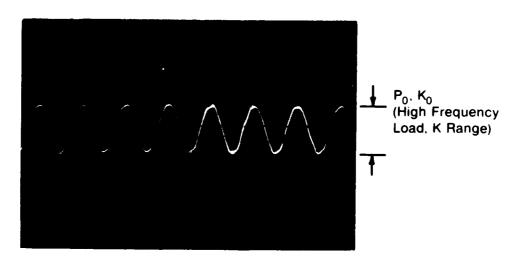


Figure 10b High frequency component with expanded time scale.

lar considerations apply to purely electrodynamic and combined servo-hydraulic/electrodynamic systems. The specimen dimensions and starting crack length (a) were, therefore, chosen based on system displacement limitations and desired high-frequency  $\Delta K$  range.

## C. Dynamic Evaluation of Test Specimens

Strain gage measurements on a preliminary center crack specimen (Figure 11 and 12) used in the test program showed that while it was adequate for 200 Hz testing it was inadequate for frequencies above 500 Hz. A series of system resonances resulted in substantial bending stresses and a generally poor correlation between load cell measurements and strain gage measurements of stress in the crack region. A modal analysis was performed on this preliminary specimen to determine how it should be modified to reduce the density of resonances in the frequency regime beyond 500 Hz. The resulting modifications on the specimens were successful in providing several frequency bands above 500 Hz in which undesirable dynamic stresses were eliminated.

An important objective of the present work is to compare the fatigue crack growth behavior of the material at several frequencies. It is, therefore, extremely important to ensure that the loading pattern is the same at these frequencies. A series of specimen designs were evaluated with strain gages and a successful design evolved. The criteria established for the specimen were as follows:

- Bending stresses and out of phase components of stress at the specimen crack line must be less than 5% of the high cycle amplitude (this is a requirement of ASTM standard E647 extended to cover all dynamic disturbinges of stress uniformity).
- The load measurement must have an appropriate relationship to the stream in the crack region throughout the test.
- Stresses at several locations in the crack stress field must have an appropriate relationship to each other.

In view of the fact that resonant frequencies shift as the crack grows, the following condition must also be fulfilled by the specimen:

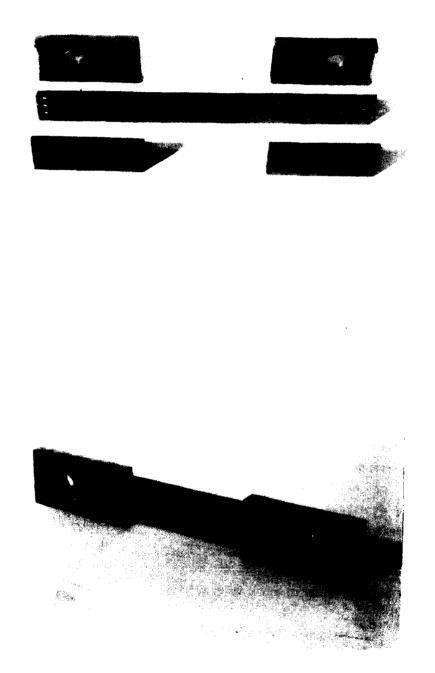


Figure 11 Photographs of the Preliminary Specimen used for 200 Hz Assembled and Disassembled

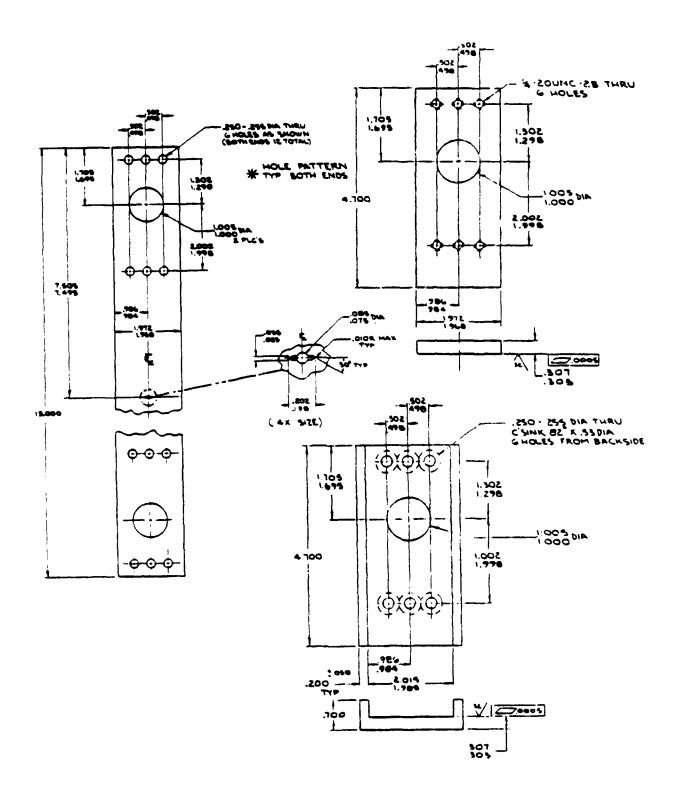


Figure 12 Drawings Showing the Dimensions of the Preliminary Specimen Used for 200 Hz Testing

- The above conditions must apply over a range of at least 200 Hz in the trequency regime of interest over the entire range of loading and crack length experienced during a test.

The interval of 200 Hz was chosen based on the expected variation in resonant frequency resulting from specimen material modulus variation as the specimen is brought to the test temperature. This can be demonstrated by considering the expression for the ringing frequency of an undamped plate:

$$v_n = \frac{1}{2\pi} \kappa_n^2 \left(\frac{EIg}{\gamma S}\right)^{1/2} \tag{2.1}$$

where  $K_n$  is determined by the boundary conditions,

and where

S = specimen cross-sectional area

Y = density

E = elastic modulus

I = moment of inertia

g = accelerometer of gravity

2 = length of specimen

With the average 10% reduction in elastic modulus over the specimen region participating in the vibration mode, a 5% change is expected in resonant frequency or about 100 Hz at a resonant frequency of 2000 Hz.

In Sections C.1 and C.2, the results and conclusions of the modal analysis are presented along with a description of the specimen adopted for testing around 2000 Hz and a summary of the strain gage dynamic evaluation of this specimen at 1825 and 2000 Hz.

# C.1 Modal Analysis of the Preliminary Specimen

After the evaluation of the preliminary specimen (shown in Figures 11 and 12) with strain gages, it was apparent that some modification of the specimen would be required to make it suitable for testing near 2000 Hz. Comparison of strain gage response from opposite surfaces indicated that bending stresses well beyond that permitted by ASTM standard E647 existed over most of the frequency range

from 1000 to 2000 Hz. It was felt that a modal analysis should be performed on the specimen to determine the resonant frequencies and mode shapes of these resonances. The information gained from this modal analysis was subsequently used to determine modifications that would be necessary to make the specimen suitable for 2000 Hz testing.

The preliminary specimen with the clamping fixtures extended (Figure 13) was placed in the load frame in the usual manner and a static load was applied. An accelerometer was attached to the specimen or load frame at one of 43 locations. The specimen was then excited with a random signal having the spectrum shown in Figure 14. The signal was provided with a Scientific Atlanta Vibration Controller which has the capability of open loop or closed loop vibration control. The modal analysis was conducted with open loop excitation. The accelerometer was moved successively to the locations on the specimen shown in Figure 15 and the random vibration was applied. Additional locations on the load frame were included but the levels of vibration were considerably less than those on the specimen. The accelerometer response and shaker excitation were simultaneously recorded. The data was processed using a Hewlett Packard 6451C Fourier Analyzer System which calculates the transfer function between the input and response at points on the specimen and load frame. An analytical model was curve fitted to the transfer function data and modal parameters such as natural frequency, damping factor, and mode shape were identified. The system software also has the capability of providing animated representations of the mode shapes. Modal parameters for the following mean load and crack length (3a) cases were evaluated:

```
- Mean load = 2000 lbs., crack length (2a) = 0.20"
```

Table 2 lists the natural frequencies and damping factors for the resonant modes for each of these cases. The mode shapes for the three cases are shown in Figure 16 through 18.

The modal analysis on this specimen demonstrated the following:

<sup>-</sup> M an load = 4500 lbs., crack length (2a) = 0.20"

<sup>-</sup> Mean load = 2000 lbs., crack length (2a) = 0.95"

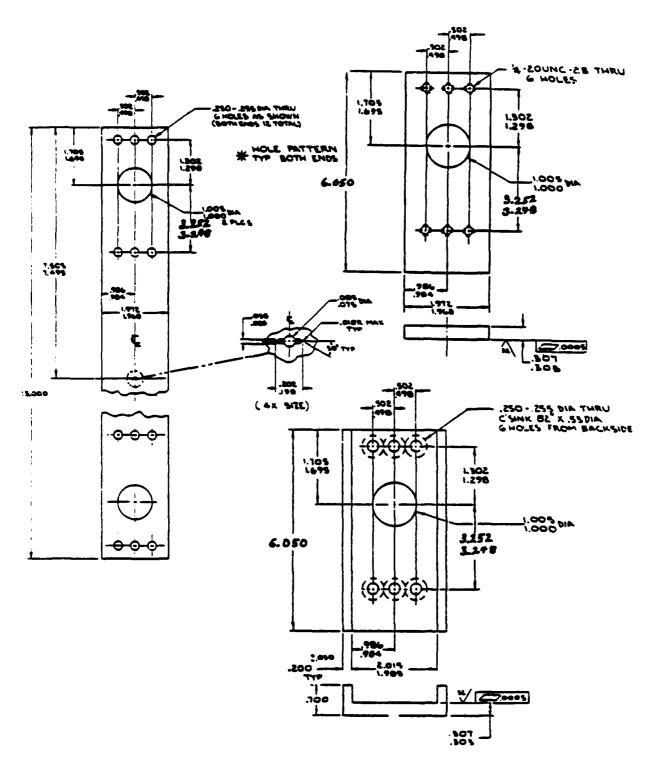


Figure 13 Drawings Showing the Dimensions of the Preliminary Specimen Subjected to Modal Analysis

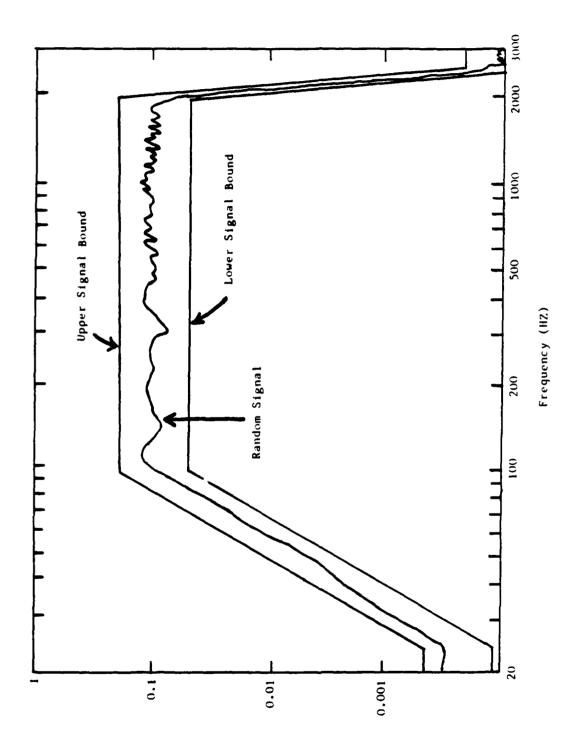


Figure 14 Relative input power spectrum used in the modal analysis

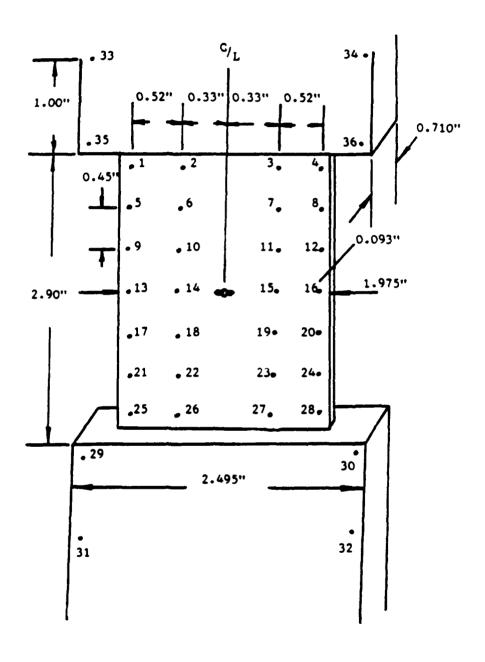


Figure 15 Accelerometer locations used in the modal analysis

### TABLE 2 NATURAL FREQUENCIES AND DAMPING FACTORS FOR RESONANT MODES

CASE I: Mean load = 2000 lbs, total crack length (2a) = 0.20"

MODE	NAT. FREQ. (HZ)	DAMP. FACT. (%)	DAMP. COEFF. (RAD/SEC)
1	765.0115	.8677	41.7089
2	1439.8879	3.0185	273.2087
3	1709.9106	1.4115	151.6672
4	1861.4131	1.5402	180.1682
5	1955.4885	.4472	54.9416

CASE II: Mean load = 4500 lbs, total crack length = 0.20"

MODE	NAT. FREQ.	DAMP. FACT. (%)	DAMP. COEFF. (RAD/SEC)
1	827.7138	3.6762	191.3148
2	1483.2148	1.0132	94.4276
3	1673.6045	.9470	99.5824
4	1806.6233	2.1325	242.1211
5	1953.5444	.3507	43.0522

CASE III: Mean load = 2000 lbs, total crack length = 0.95 "

MODE	NAT. FREQ. (HZ)	DAMP. FACT. (%)	DAMP. COEFF. (RAD/SEC)
1	646.3500	.3708	15.0568
2	795.7371	2.7693	138.5117
3	1381.7729	3.9457	342.8291
4	1690.2246	1.4620	155.2824
5	1883.5137	1.4233	168.4624

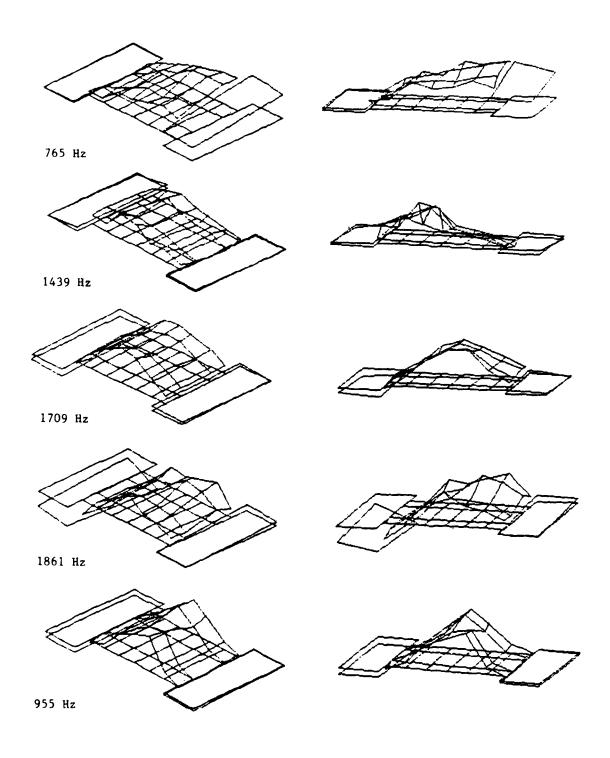


Figure 16 Mode Shapes for Case I: Mean Load 2000 Lbs, Crack Total Length (2a) of 0.20"

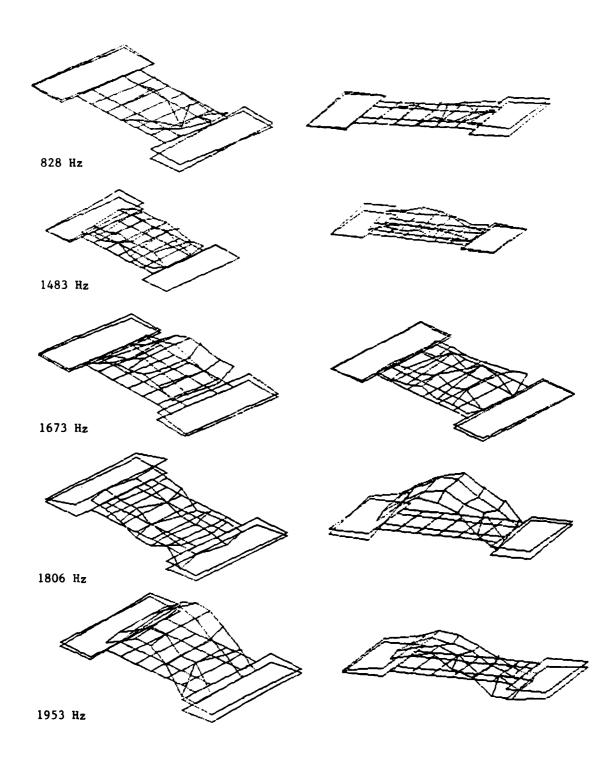


Figure 17 Mode Shapes for Case II: Mean Load 4500 Lbs, Crack Total Length (2a) of 0.20"

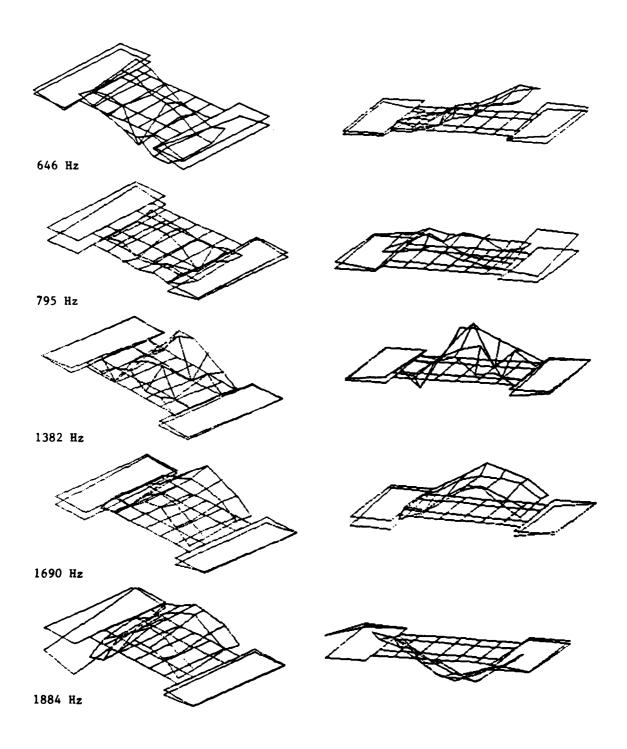


Figure 18 Mode Shapes for Case III: Mean Load 2000 Lbs, Crack Total Length of 0.95"

- The load level and crack length changes expected during a test can shift resonant frequencies substantially, in some cases as much as 200 Hz.
- While these resonances involve the entire load train and possibly the frame as well, the most severe lateral deflection is in the specimen.
- There was a cluster of resonances from 1400 to 2000 Hz with an average of 200 Hz spacing between them.
- There was a rather clear field between 750 Hz to 1200 Hz where no resonances developed for any of the cases.

The behavior shown in the modal analysis corresponds well to that shown by strain gage response of this preliminary specimen. Strain measurements confirmed that there is a regime between approximately 800 to 1200 Hz in which there are no complications from dynamic bending stresses. In the regime between 1200 to 2000 Hz, strain measurements likewise confirmed the series of resonances that result in significant bending strains that cannot be tolerated in a test. In several of the mode shapes the greatest deflection is in the thin part of the specimen. It was felt that if these deflections can be reduced significantly by lateral support and damping, the specimen may provide a satisfactory stress distribution at the crack region.

#### C.2 Strain Gage Evaluation of a Specimen with Lateral Reinforcement and Damping

The strain gage evaluation of the preliminary specimen showed that bending strains were excessive over most of the frequency between 500 and 2000 Hz with the possible exception of the range 1000 to 2000 Hz. The ranges of frequency over which the strains on either side of the specimen are in phase and of equal magnitude was extremely limited and certainly less than the 200 Hz that is required due to shifting resonances during a test.

Recognizing that the preliminary specimen with the clamping arrangement extended as much as possible was still unsuitable for testing above 1200 Hz, several experiments were performed to determine the effectiveness of various approaches to spreading out or damping resonant vibrations. The modal analyses showed that the most extreme deflection is in the unclamped portion of the specimen. It appears that since this region is the most compliant it acts somewhat

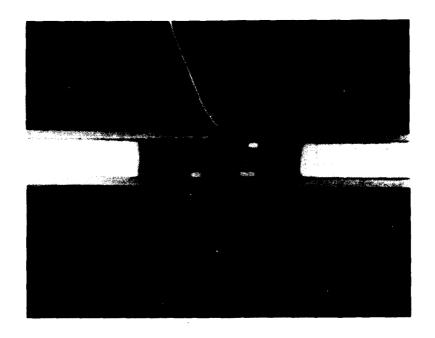
as a hinge between the load train elements above and below it. Three additional modifications were, therefore, made to the specimen and system:

- the construction of lateral buttressing that would reinforce the specimen in the unclamped region with respect to out of plane motion yet have minimal effect on tensile stress distribution
- removal of elements in the load train which may add to unwanted deflections
- increasing the size of the clamping fixture to improve stiffness in the lateral direction

Figure 19 shows the lateral reinforcements applied to the specimen. A schematic diagram of the central region of the specimen indicating the elements of the lateral support are shown in Figure 20. Parallel elements were loaded against the unclamped surfaces of the specimen near the crack. By applying this load through several layers of glass cloth, the support still had substantial compliance in shear and would not significantly affect the distribution of tensile stresses in the specimen. The glass cloth was also expected to provide damping of lateral vibration. Specific dimensions of the lateral support and clamping fixtures are shown in Figure 21.

A series of experiments were conducted on this specimen to establish its suitability and also to establish a set of procedures for verifying an appropriate stress distribution on each specimen prior to each experiment. The set of experiments involved strain gage measurements in the locations shown in Figure 22.

The first group of experiments involved the amplitude measurement of two strain gages on opposite surfaces when a 2500 lb. preload and a high cycle amplitude were applied. Figure 23, for example, shows the output of strain gage 1 and 2 as a function of frequency. Over most of the frequency range the correlation is acceptable indicating a satisfactorily low level of bending stresses at the crack line over most of this frequency range. There are, however, several frequency ranges in which large discrepancies occur. The stress amplitudes as shown by strain gages 1 and 7 on the same side of the specimen (Figure 24) likewise show a good correlation for this uncracked specimen over most of the



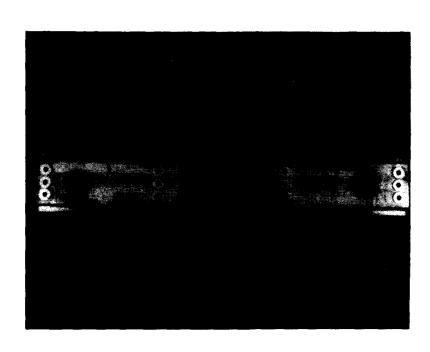


Figure 19 Specimen with End Reinforcement and with Lateral Constraints Applied to the Crack Region

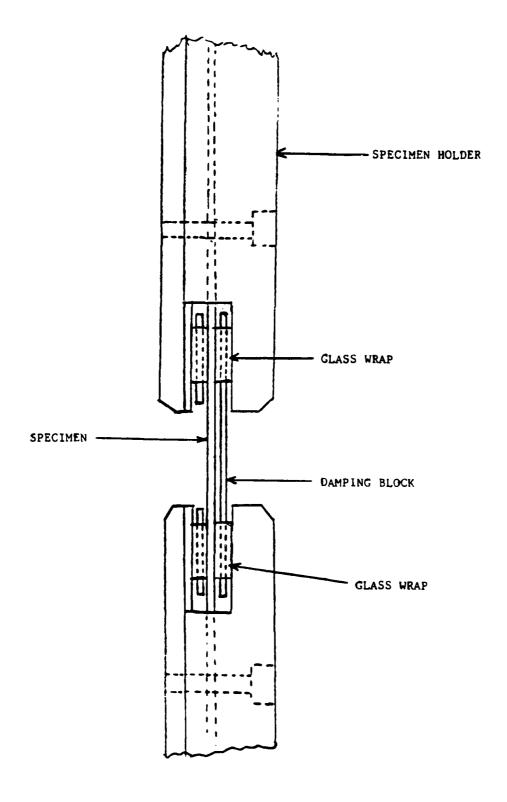


Figure 20 Diagram Showing Location of Damping Blocks and Glass Insulating Material

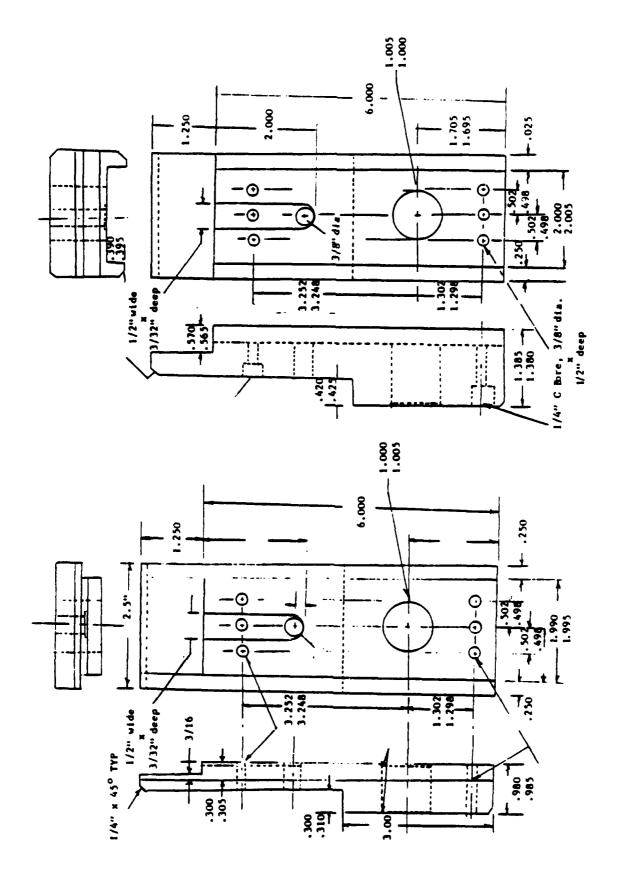


Figure 21 Dimensions of Specimen End Clamps

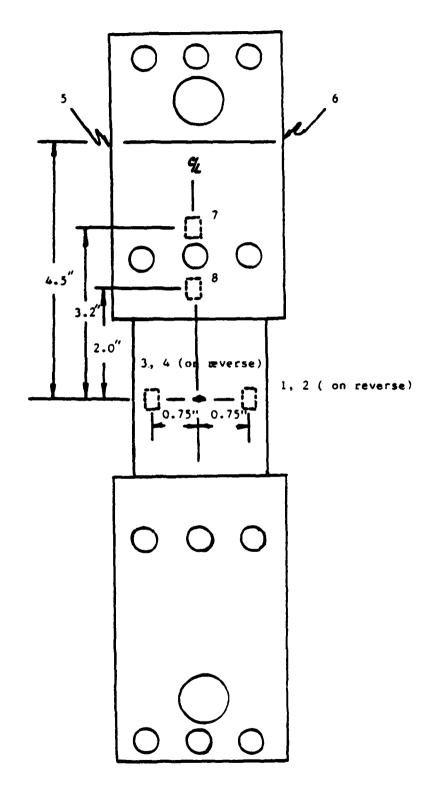


Figure 22 Diagram Showing Location of Strain Gages

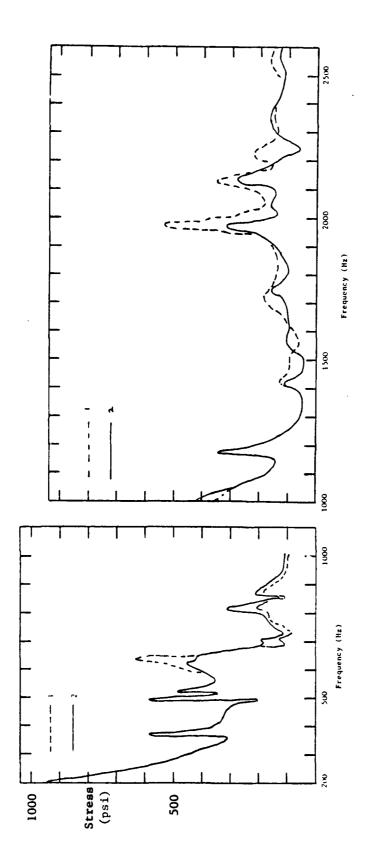


Figure 23 Magnitude of Stress Versus Frequency in Lccations 1 and 2 for Laterally Damped and Reinforced Specimens

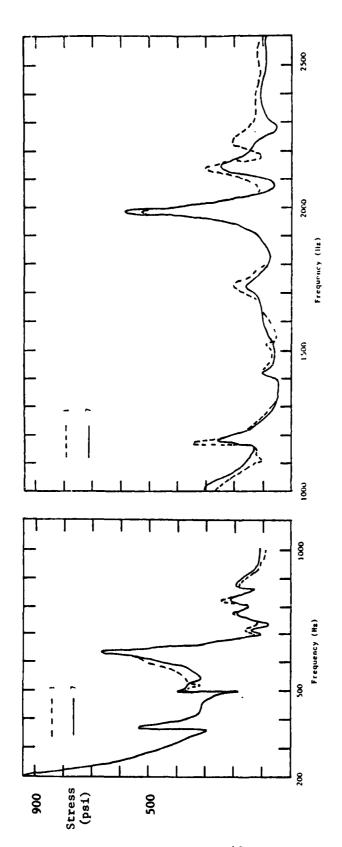


Figure 24 Magnitude of Stress Versus Frequency in Locations 1 and 7 for Laterally Damped and Reinforced Specimens

frequency range investigated but with significant discrepancies in certain narrow ranges of frequency.

The effect of reducing the length of the load train by removing the load cell and compression rings was evaluated. As shown in Figure 25, the effect of eliminating these objects was to modify the ranges over which differences in amplitude between strain gages 1 and 2 occurred. While the correlation between these outputs is improved in the frequency range between 1900 and 2200 Hz, in other ranges the correlation showed little change and in some cases deterioration. It was found in subsequent tests that removing the compression rings alone can improve the correlation of strain gage measurements in the frequency range between 1900 and 2200 Hz.

This series of experiments, involving the measurement of stresses at various locations on the specimen, demonstrated that there are frequency ranges above 1000 Hz in which dynamic stresses do not dist rb the stress pattern associated with tensile loading. Satisfied that the specimen with lateral damping could provide satisfactory test results in some frequency band near 2000 Hz experimentation was carried further to identify frequency bands in which testing could be carried out and also establish procedures that could be used to verify the appropriateness of the stress distribution prior to each test. The verification on each specimen is necessary because there is a possibility that the effectiveness of the lateral support may depend on the procedures of assembly.

The verification procedures adopted involved the measurement and comparison of both phase and amplitude on opposite surfaces of the specimen. An initial experiment was carried out over a range of frequencies near 2000 Hz to determine how the relative magnitude and phase vary with changing load and crack length. The strain measurements were along the crack line at locations 1 and 2 of Figure 22. Measurement of the output of strain gages 1 and 2 were made and displayed on an oscilloscope as 1 versus 2. In the absence of bending stresses the resultant would be a line at 45° from the x or y axis, i.e. the stresses would be in phase and of equal magnitude. Resonant vibrations are apparent as a deviation from this pattern. An appropriate stress distribution would have a maximum peak to peak deviation of 5% from the ideal 45° trace. This condition would also be required over a 200 Hz interval around the chosen test frequency in order to ensure that resonances are not "swept in" by increasing the specimen temper-

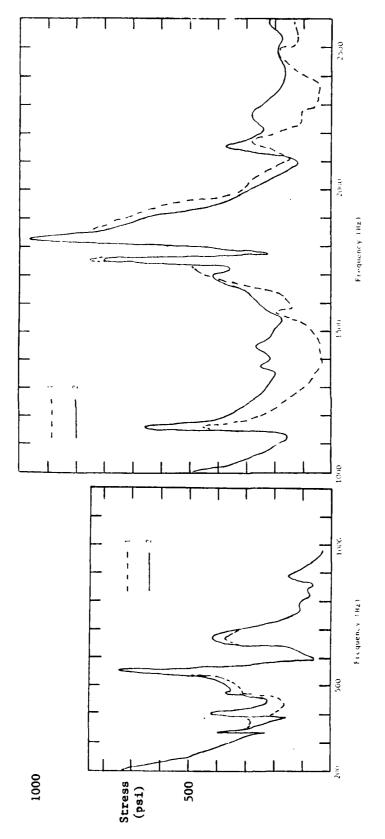


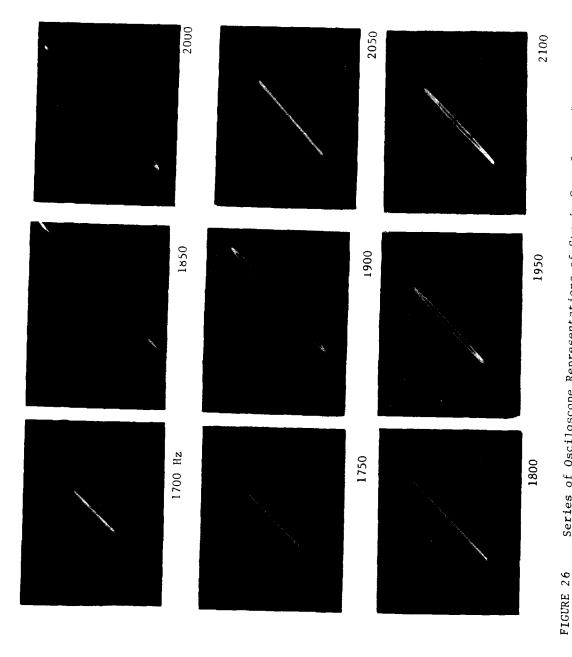
Figure 25 Magnitude of Stress Versus Frequency in Locations 1 and 2 for Laterally Damped Specimens with Compression Rings and Load Cell Removed from System.

rane. It is also required over the load range and crack length experienced in a typical experiment. Figures 26 through 28 show the results of measurements over a range of trequency, crack length and mean load. These measurements were made to experial specimens to ensure that bending stresses were consistently below the acceptable level from specimen to specimen.

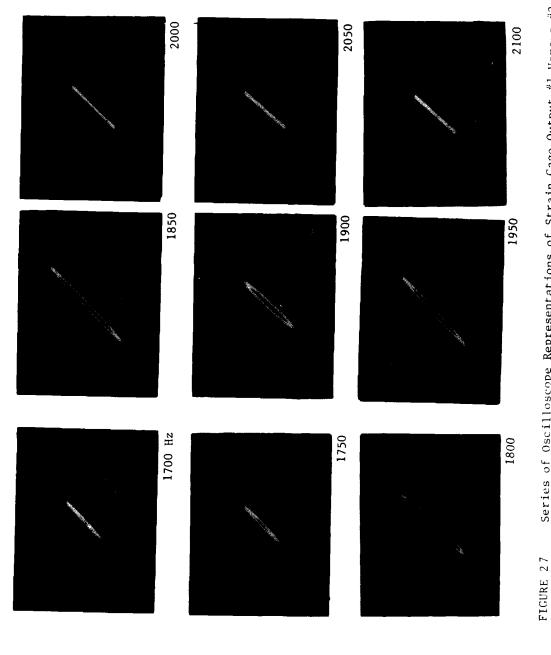
Prior to each crack growth, test strain gage measurements were made in this manner over a 200 Hz frequency interval to ensure that the specimen was properly assembled. This procedure was adopted to ensure that errors in assembly that might reduce the effectiveness of the lateral support had not occurred.

The coat sensing for the high frequency load range was performed with the remote load cell. For frequencies near 2000 Hz it was required to apply a correction ta terminal the measured load in order to properly represent the stresses in the animity of the specimen crack. Load cell output for a given crack length and applied load was measured as a function of strain gage output at 20 and 200 Hz. The proportionality at these two frequencies was consistently the same. The proportionality between load cell output and strain gage output was then measured at 1825 and 2000 Hz. The correction factor that must be applied to load cell measurement in order to provide the same proportionality as at the lower trequencies was established for the range of load and crack length measurement experenced in a typical experiment. The correction factor variation was 12% over a typical range of test conditions.

Sensing load directly on the specimen at locations 7 or 8 shown in Figure 22 was considered. However, in view of the fact that elevated temperature strain gages would be required and that strain gages on the specimen are frequently destroyed near 2000 Hz, it was decided to perform tests with remote load sensing. With remote sensing a much higher testing productivity was achieved with perhaps a small sacrafice of absolute high frequency load measurement accuracy.



Series of Osciloscope Representations of Strain Gage Output #1, versus #2 Showing the Degree of Bending and Out of Phase Vibration on the Specimen Crack Centerline for a Mean Load of 4000 lbs. and a Crack Length of .200".



Series of Oscilloscope Representations of Strain Gage Output #1 Versus #2 Showing the Degree of Bending and Out of Phase Vibration on the Specimen Crack Centerline for a Mean Load of 2000 lbs. and a Crack Length of 0.200".

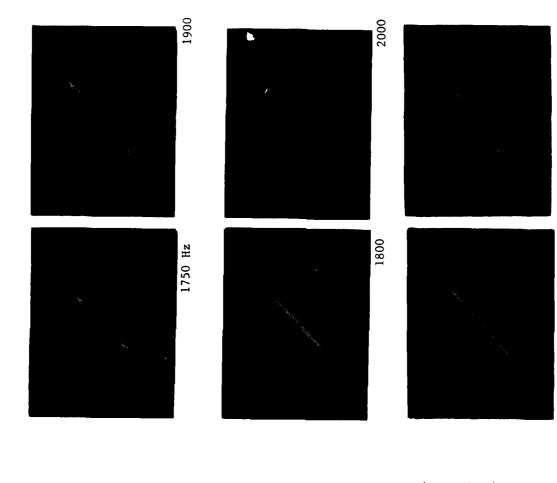


FIGURE 28 Series of Oscilloscope Representations of Strain Gage
Output #1 versus #2 Showing
the Degree of Bending and Out
of Phase Vibration on the
Specimen Crack Centerline for
a Mean Load of 2000 lbs. and
a Crack Length of 0.700".

### III. EXPERIMENTAL TEST PROGRAM

The test program objective was the establishment of the relationship between crack growth rate in an aircraft engine disk material and those parameters associated with the high- and low-frequency loading experienced in the engine first- and second-stage disks. The information provided by this testing program was to be applicable to life prediction of tlawed engine components and provide guidance to the implementation of the retirement-for-cause engine maintenance and design concept. Consequently, the test parameters selected for this program were based on the loading conditions experienced by aircraft engine disks.

Considering the nature of the loading and the requirements of aircraft engine design, the following were included in the experimental program:

- Determination of the nature of the transition from low cycle to high cycle dominated behavior over ranges of both low cycle and high cycle stress intensity factor range ( $\Delta K$ ).
- Establishment of the high-cycle transition  $\Delta K$  over as wide a frequency range as possible.
- Major cycle (low cycle) hold times in the regimes in which both fatigue and creep crack growth dominate. Cycle times from a few seconds to several hundred seconds were included.
- A temperature typical of those experienced by the aircraft engine disk.
   For the Inconel 718 specimens used in this study, 1200°F was chosen.
- A sufficient level of replication to eliminate the influence of material variability on the test results and indicate the level of consistency of the experimental system.
- The influence (if any) of the high-cycle loading on crack growth below the high-cycle transition.

The test program summarized in Table 3 was designed to address these aspects of high- and low-frequency interaction in crack growth. The low cycle waveform used throughout the testing program was a trapezoidal loading profile with ramp times of 0.5 seconds and hold times ranging from a second to essentially infinity (steady mean load). The high-frequency loading was applied during the hold period only.

# TABLE 3 TEST PROGRAM OUTLINE (ALL FESTING AT 1200°F)

Objective	Conditions of Test
Evaluate high-cycle threshold AK and crack growth rate versus high frequency AK in the creep crack growth regime.	Selected low-frequency $\Delta K$ values with long hold times (60 seconds or greater) maintained throughout the test and with varying high-frequency $\Delta K$ values.
Evaluate high-cycle threshold AK and crack growth rate versus high-frequency AK in the fatigue crack growth regime.	Selected low-frequency $\Delta K$ values with the shortest practical hold times (probably on the order of 1 to 10 seconds) maintained throughout the test and with varying high-frequency $\Delta K$ values.
Evaluate the effect of high- frequency loading on low- cycle crack growth in the re- gime of transition between creep and fatigue-dominated crack propagation.	Selected low-frequency hold times with specific low-frequency $\Delta K$ levels maintained throughout the test and with varying high-frequency $\Delta K$ levels.
Evaluate the effect of high- frequency loading at several low-frequency cycle R ratios in the creep and fatigue crack growth regimes.	Varying high-frequency $\Delta K$ values emphasizing the high-cycle transition regime with specific low cycle hold times maintained throughout the test.
Evaluate the effect of temp- erature on the high-cycle transition.	Varying high-frequency $\Delta K$ values emphasizing the transition high-cycle regime with specific low-frequency $\Delta K$ and hold times and selected temperatures.

### A. Fatigue Crack Growth Studies Conducted at 200 Hz

Combined cycle tests with a high cycle trequency of 200 Hz were conducted for low cycle parameters in a test matrix. All testing in this matrix was carried out with a low cycle R ratio of 0.1 and a test temperature of  $1200^{\circ}F$ . This matrix included low cycle maximum K values ranging from 15 to 40 ksi  $\checkmark$ in (corresponds to a  $\Delta$ K of 15 to 40 MPa  $\checkmark$ m since the R ratio was 0.1) and low cycle hold times ranging from 2 to 180 seconds. Table 4 shows the conditions of tests completed and the number identifying the test. All conditions in the test matrix were applied in at least one test. In several cases replicated tests were conducted. Data plots for all of these tests may be found in Appendix A, with corresponding listings in Appendix B.

The low cycle  $\Delta K$  ranges included in the testing were 15, 20, 30 and 40 MPa  $\sqrt{m}$ . The low cycle hold times that included 2, 5, 10 and 180 seconds were expected to cover the regimes in low frequency loading in which the low cycle crack growth is time dominated (creep crack growth) and the regime in which the number of low frequency cycle influences crack growth rate (combination of creep and fatigue crack growth). The lower end of the low cycle hold period range (i.e., 2 and 5 seconds) was expected to show the effect of accumulated low frequency cycles on the low cycle crack growth rate. The series of data plots representing crack growth rates versus high cycle  $\Delta K$  for constant low cycle  $\Delta K$  and low cycle hold time obtained in this study show several interesting trends.

In the curves of crack growth versus high frequency  $\Delta K$  distinct regimes can be seen. As shown in Figure 29, three types of behavior were observed over the range of low frequency  $\Delta K$  and hold times investigated. In type 1, the crack growth rate versus high cycle  $\Delta K$  remained relatively constant in the low cycle dominated regime prior to the rapid increase in crack growth rate in the high cycle dominated regime. Type 2 behavior was characterized by retardation of crack growth rate by the high frequency cycle in the low cycle dominated regime. Type 3 behavior was typical of the lowest low cycle  $\Delta K$  studied, in which the low cycle  $\Delta K$  was below the crack growth threshold and no crack growth could be measured in the low cycle dominated regime. In all these cases distinct low cycle and high cycle dominated regimes could be observed. However, the transition between these two regimes was not always distinct due to the retardation effect.

TABLE 4: COMBINED CYCLE TEST INCLUDING A 200 HZ HIGH CYCLE FREQUENCY

(All testing was conducted at 649°C (1200°F) and with a low cycle R ratio of 0.1)

		LOW CYCLE HOLD TIME (sec)			
		2	5	10	180
	15	TEST #'s	21, 23	20	42, 43
Low Cycle Maximum	20	35, 37	30, 28	26, 27	39, 40
K (MPa √m)	30	46	31, 32	47	41
	40	36, 38	33, 34	48	44

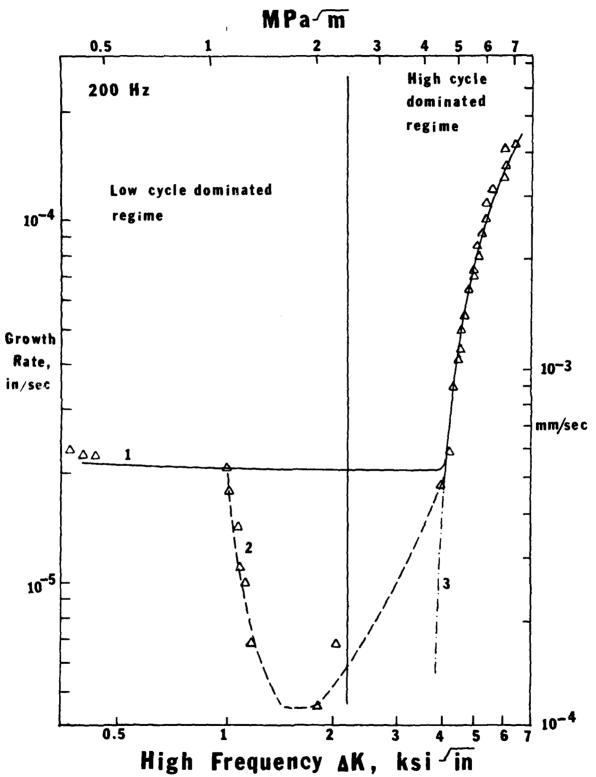


FIGURE 29 Characteristics of the High/Low Frequency Interaction Showing the Three Types of Behavior Observed in This Study. The Points Correspond to testing with a Low Frequency  $\Delta$  K of 20 MPa  $\sqrt{m}$ , a Low Cycle Time of 10 Seconds and a High Cycle Frequency of 200 Hz.

Figure 30 shows a curve corresponding to a low cycle  $\Delta K$  of 15 MPa  $\sqrt{m}$  and a hold time of 5 seconds. The data in this representation corresponds to a test with increasing high frequency  $\Delta K$ . Prior to data acquisition in the increasing high cycle  $\Delta K$  mode, the crack was allowed to grow with a systematically decreasing  $\Delta K$  until a crack growth rate on the order of 5 x  $10^{-4}$  mm/sec (2 x  $10^{-6}$  inches/sec) was achieved. This precaution was taken to eliminate the effects on crack growth of the prior precycling. The data presented in Figure 30 is characteristic of threshold fatigue crack growth data which generally exhibits increasing growth rate and decreasing slope with increasing  $\Delta K$  when crack growth versus  $\Delta K$  is plotted on log-log axes. The lower level of this curve corresponds to a growth rate of 1.3 x  $10^{-8}$  inches per high frequency cycle (3.30 x  $10^{-7}$  mm/cycle) which would definitely be in the threshold regime for Inconel 718.

Figure 31 shows the results of a test conducted with a low frequency AK of 20 MPa √m and a hold time of 5 seconds. It was carried out with a sequence of loads intended to illustrate an important aspect of the retardation effect that is very pronounced at a low frequency ∆K of 20 MPa /m. The line drawn through the experimental points has arrows drawn to show the sequence of points as they occurred during the test. The initial loading up to point A seems to give rise to a measurable retardation, and changing the high frequency load range to that at point B rapidly accelerates the retardation. This results in a more severe retardation in the 0.762mm (0.030) inches of growth beyond point B than was accomplished in the 2.79mm (0.110 inches) of growth with the high cycle AK range around point A. (Each point represents 0.010 inches, 0.254mm, of crack growth). Beyond point B the crack growth rate decreases rapidly, reaches a minimum value and then starts to increase. At point C just beyond the minimum value of crack growth, a lower high frequency load range was applied (the new level of high cycle AK is represented by point D). The crack growth rate increases from point D to E showing a gradual elimination of the retardation effect. At point E the load range was again increased to point F and crack growth continued in the high cycle don nated regime.

As the low frequency  $\Delta K$  increases, the retardation effect generally becomes less pronounced. The data for a low cycle  $\Delta K$  of 30 MPa  $\sqrt{m}$  and a low cycle hold time of 5 seconds appears in Figure 32. While the high frequency load results in a factor of four reduction in crack growth rate for a low cycle  $\Delta K$  of 20 MPa  $\sqrt{m}$ , at a low cycle  $\Delta K$  of 30 MPa  $\sqrt{m}$  the reduction in crack growth rate is only a factor

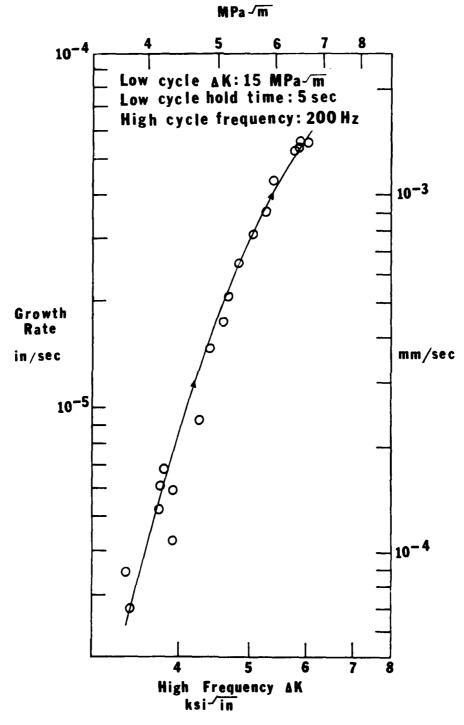


FIGURE 30 Results of Combined High/Low Frequency
Test With a Low Cycle AK of 15 MPa √m
and a Low Cycle Hold Time of 5 seconds.



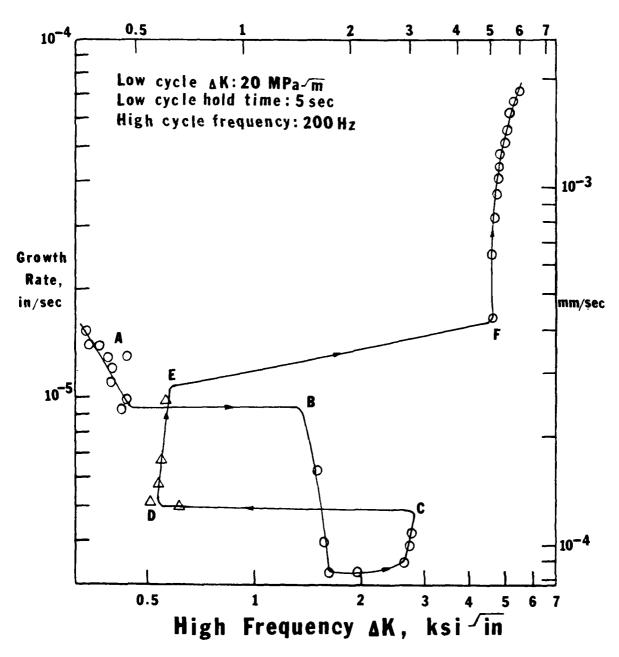


FIGURE 31 Results of Combined Cycle Test With a Low Frequency  $\Delta K$  of 20 MPs  $\sqrt{m}$  and a Hold Time of 5 Seconds. The Line is Shown to Show the Sequency of Points.

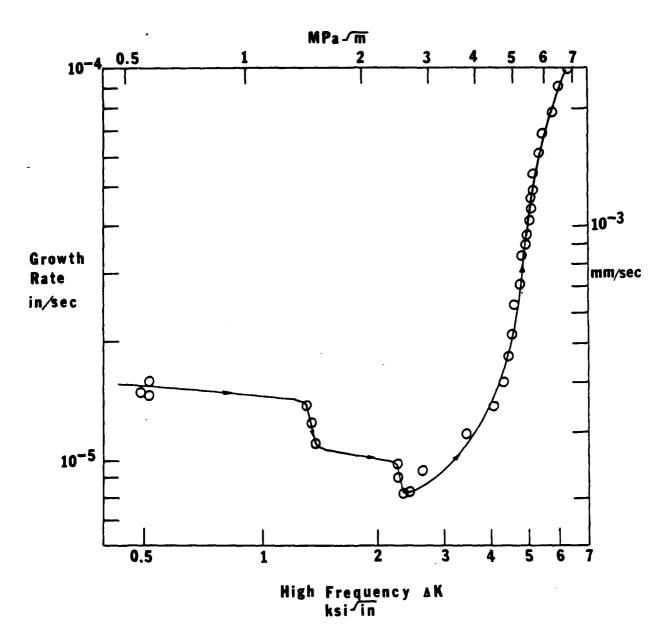


FIGURE 32 Results of a Combined Cycle Test With a Low Frequency  $\Delta K$  of 30 MPa  $\sqrt{m}$  and a Hold Time of 5 Seconds.

of 2. As shown in Figure 33, a low cycle  $\Delta K$  of 40 MPa  $\sqrt{m}$  shows no measurable retardation associated with high frequency loading.

Figure 34 shows the effect of varying cycle time on the crack growth behavior with a low cycle  $\Delta K$  of 15 MPa  $\sqrt{m}$ . No distinct trend is apparent and there is little deviation between these curves. Figure 35 shows the effect of cycle time ranging from 2 seconds to 180 seconds on the crack growth behavior with a low cycle  $\Delta K$  of 20 MPa  $\sqrt{m}$ . The only significant feature in this group of tests is that with a 180 seconds hold time there appears to be a more severe retardation.

A comparison of crack growth rate versus high cycle  $\Delta K$  for a hold time of 5 seconds and several values of low cycle  $\Delta K$  appears in Figure 36. As expected the crack growth rate in the low cycle dominated regime increases as  $\Delta K$  increases. A similar comparison is made in Figure 37 but with a low cycle hold time of 180 seconds and roughly the same behavior can be observed.

## B. Results of Combined Cycle Tests with an 1800 to 2000 Hz High Cycle Component and Comparison with Lower Frequency Results

The parameters covered by the 1800 to 2000 Hz combined cycle testing are indicated in Table 5. Tests 60 through 66 were performed on specimens from a second heat of material. The crack growth rate in the low cycle dominated regime was quite different from the previous batch of material. The newer material has crack growth rates lower by a factor of 6 to 8 at some low cycle  $\Delta K$  levels. Since, it is desirable to evaluate the effect of frequency without the complication of lot to lot material variation, fatigue crack growth testing near 2000 Hz was also performed on material from the older lot of material. Tests 67 through 75 represent tests from the same lot used for the 200 Hz tests. The complete set of data plots and listings for the 1800 to 2000 Hz combined cycle tests may be found in Appendices B and C respectively.

The dynamic tests performed on the laterally supported and damped specimen indicated that there is greater consistency in dynamic behavior at 1825 Hz than at 2000 Hz. There is also a greater high frequency load capability at 1825 Hz.

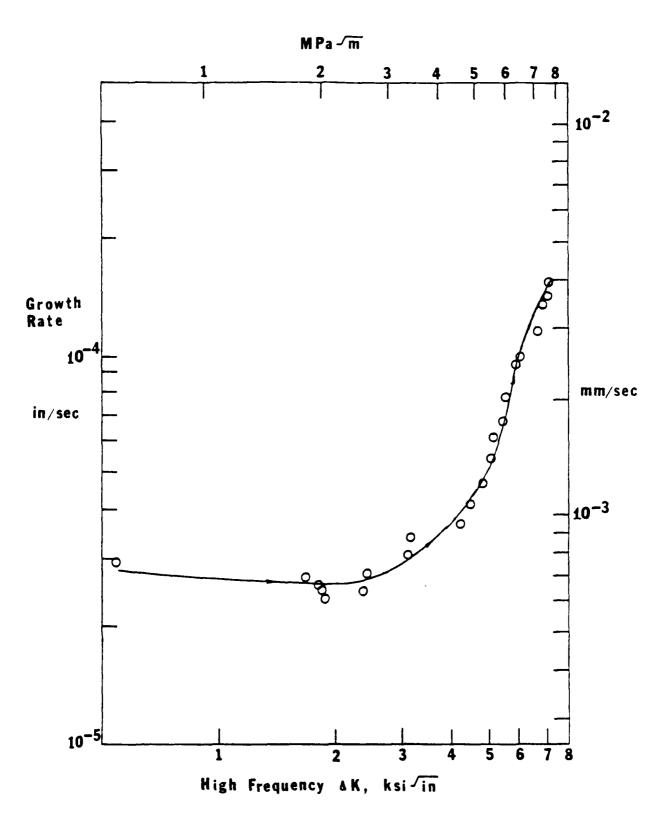


FIGURE 33 Results of a Combined Cycle Test with a Low Cycle  $\Delta$ K of 40 MPa  $\sqrt{m}$  and a Hold Time of 5 Seconds.

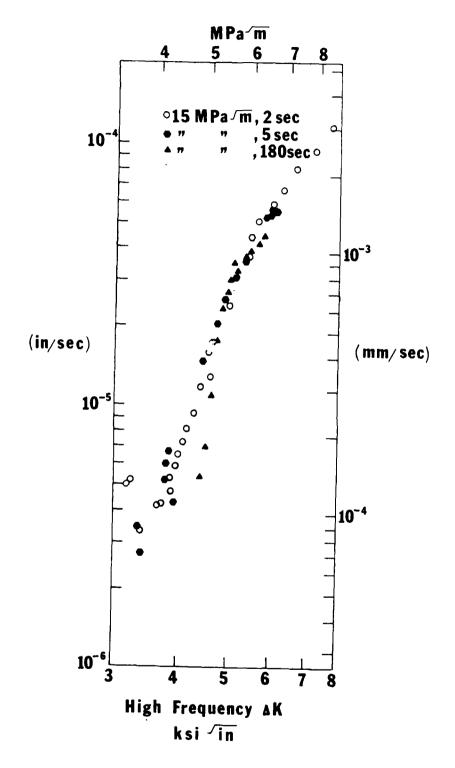


FIGURE 34 Comparison of Crack Growth Rate Versus High Cycle  $\Delta K$  for several Hold TImes With a Low Cycle  $\Delta K$  of 15 MPa  $\sqrt{m}$ .

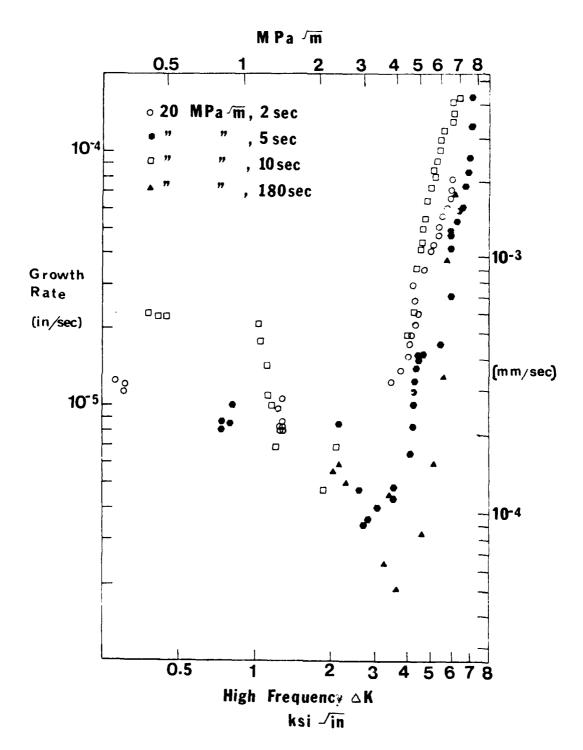


FIGURE 35 Comparison of Crack Growth Rate Vorsus High Cycle ΔK for Several Hold Times and a Low Cycle ΔK of 20 MPa √m.

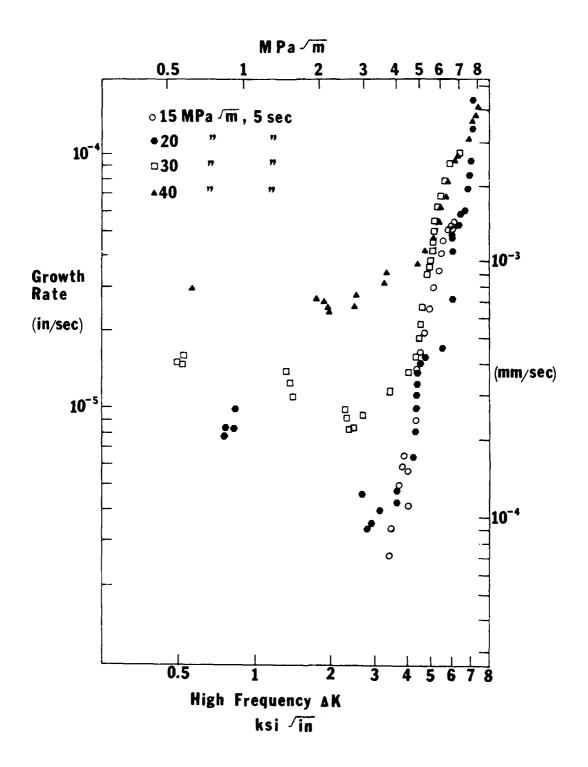


FIGURE 36 Comparison of Crack Growth Rate Versus High Cycle  $\Delta K$  for Several Low Cycle  $\Delta K$  ranging From 15 to 40 MPa  $\sqrt{m}$  With a Low Cycle Hold Time of 5 Seconds.

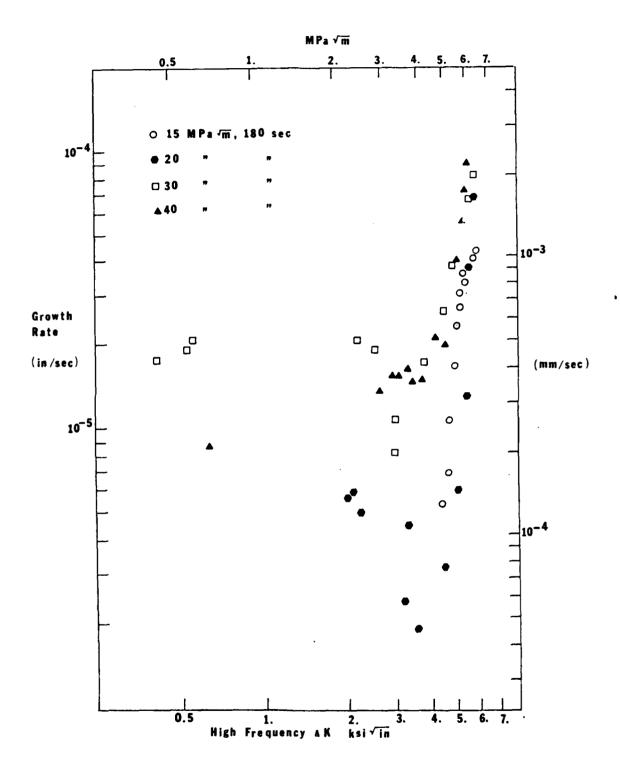


FIGURE 37 Comparison of Crack Growth Rate Versus HTgh Cycle  $\Delta K$  for Several Low Cycle  $\Delta K$  Ranging From 5 to 40 MPa  $\sqrt{m}$  With a Low Cycle Hold Time of 180 Seconds.

TABLE 5: COMBINED CYCLE TEST INCLUDING AN 1800 to
2000 Hz HIGH FREQUENCY LOAD COMPLETED TO DATE

		LOW CYCLE HOLD TIME (sec)						
		2	5	10	180			
Low Cycle Maximum  K (ksi (in)	20	TEST #'s	67	60				
	25		63					
	30		64, 68	61, 62				
	40		66, 65					

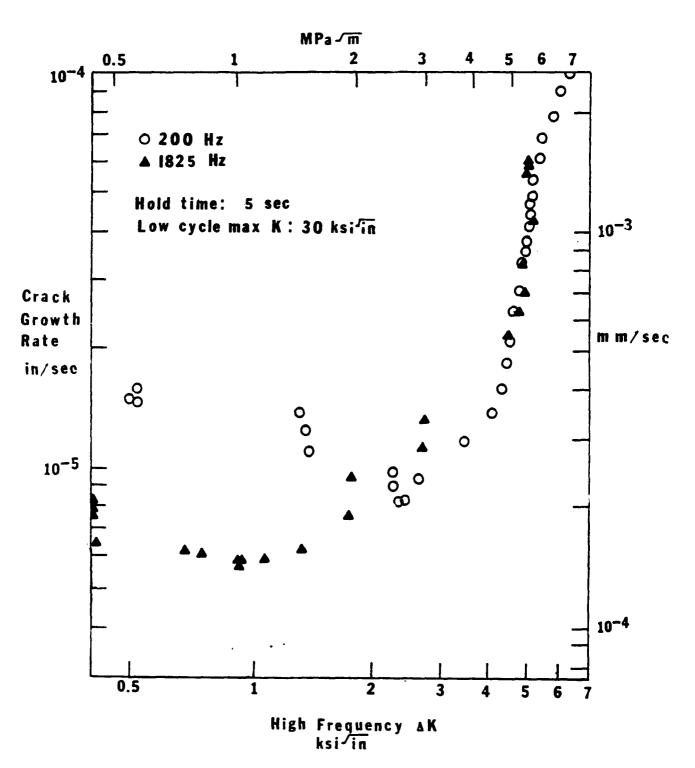


FIGURE 38 Comparison of Results for 200 and 1325 Hz for a Hold Time of 5 Seconds and a Low Cycle  $\Delta K$  of 30 MPa  $\sqrt{m}$ .

Tests 60 through 66 in this higher frequency series were conducted with several high cycle frequencies in the range of 1800 to 2000 Hz. The advantages of testing at 1825 Hz became apparent and, therefore, beyond test 66, 1825 Hz was used as the high cycle frequency. Comparison of results for 200 Hz and 1825 Hz tests may be found in Figures 38 and 39 which present data for a 5 second hold time and for a  $\Delta$ K of 30 MPa  $\sqrt{m}$  and 20 MPa  $\sqrt{m}$  respectively. A feature that the 1825 Hz tests show in these figures is a less pronounced retardation than at 200 Hz. Other tests performed near 2000 Hz show similar results. In Figure 39 for a low cycle  $\Delta$ K of 20 MPa  $\sqrt{m}$ , on the onset of high cycle activity appears to occur at a lower high frequency  $\Delta$ K at 1825 Hz than at 200 Hz. A distinct low cycle dominated range of high frequency  $\Delta$ K is apparent at both frequencies.

A comparison between combined cycle loading with a high frequency component of 200 Hz and 10 Hz is presented in Figure 40. The 10 Hz data is from Reference 1. The apparent onset of high cycle behavior is about the same. The initial slopes of the high cycle dominated regime are significantly different with the 200 Hz data having a larger slope. This behavior would be expected in the high cycle dominated regime in which the number of high frequency cycles determines the rate of crack growth.

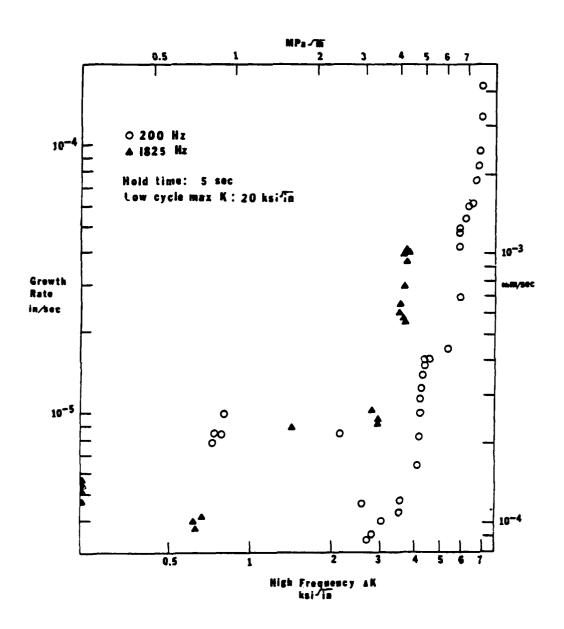


FIGURE 39 Comparison of Results for 200 and 1825 Hz for a Hold Time of 5 Seconds and a Low Cycle  $\Delta\kappa$  of 20 MPa  $\sqrt{m}$  .

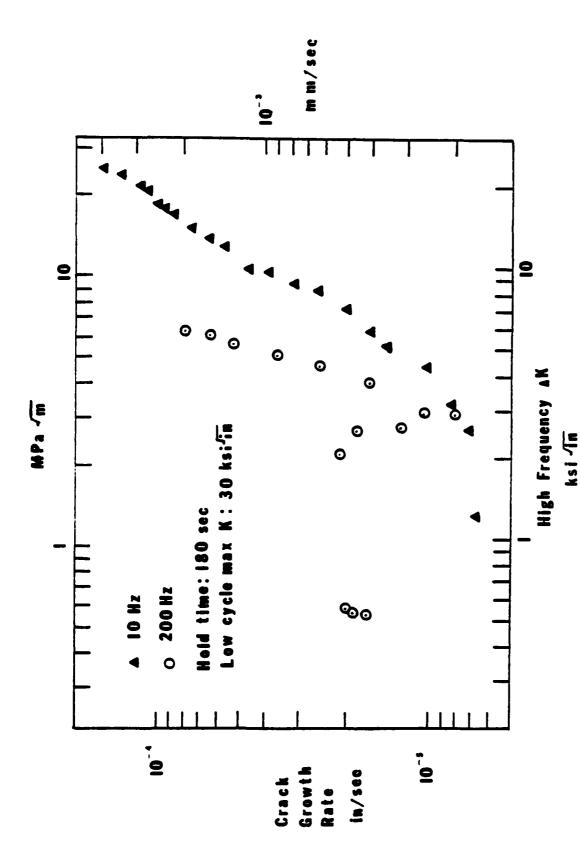
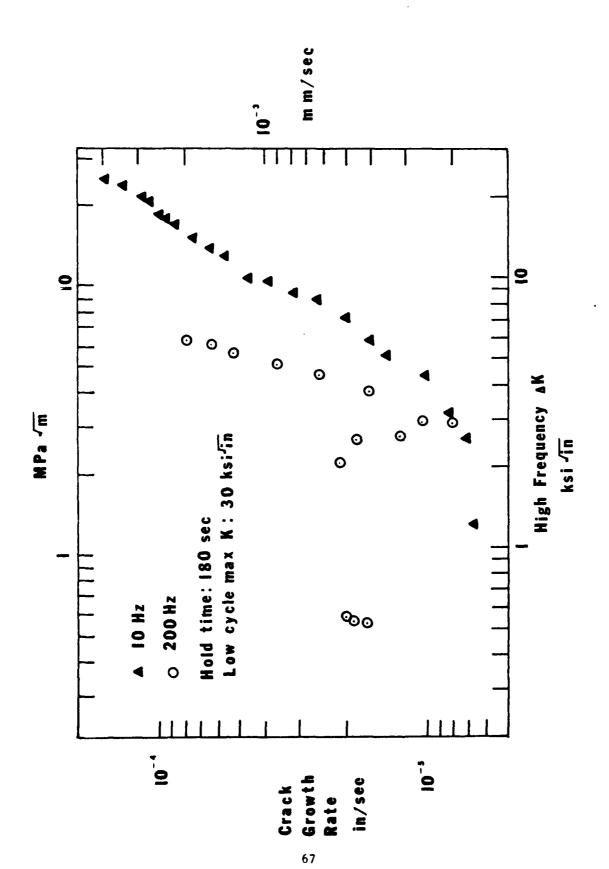


FIGURE 40 Comparison of Results for 10 and 200 Hz for a Hold Time of 180 Seconds and a Low Cycle  $\Delta k$  of 30 MPa  $4m_{\odot}$ 



## IV EVALUATION OF MECHANISMS AND MODELLING ASSOCIATED WITH FREQUENCY EFFECTS AND COMBINED HIGH/LOW CYCLE INTERACTION

There have been several studies of frequency effects in nickel base and other alloys up to frequencies of 20,000 Hz. Investigation of combined high and low frequency interactions in fatigue have also been performed. In this section the important observations and conclusions from these studies will be summarized. The test results from this program will then be discussed in the context of these previous studies.

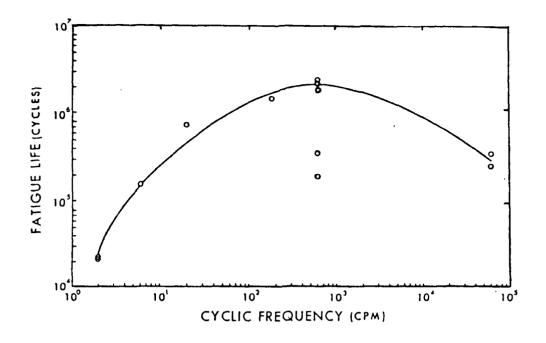
## A. Background

References 8-11 provide a review of mechanisms that apply to fatigue crack growth of nickel base alloys at elevated temperatures. These papers deal with both the initiation and propagation of fatigue cracks and the influence of frequency on these processes. Frequency effects are evaluated in terms of the effect of frequency on "slip character", which is the degree to which dislocations disperse during plastic deformation. The two extremes in slip character that nickel base alloys have exhibited are planar slip and wavy or homogeneous Planar slip is characterised by the concentration of dislocations in planar arrays with planar shear offsets produced on polished surfaces transverse to the crack plane and parallel to the direction of propagation. This type of slip and its associated deformation is favored by low stacking fault energy, ordering, the presence of coherent precipitates, low temperatures, and small strains. Austenitic stainless steel and nickel base alloys both exhibit planar slip at ambient temperatures. Wavy or homogeneous slip on the other hand, is characterized by uniformly distributed, nonplanar dislocation arrangements with an associated rumpling of the surface transverse to the crack plane and parallel to the growth direction. Wavy slip is favored by high stacking fault energies, incoherent precipitates or particles, large strains, and elevated temperatures. Most metals including stainless including stainless steels and nickel base alloys exhibit wavy slip at temperatures greater than 0.4Tm (Tm = melting temperature) because a thermally activated process allows dislocations to cross slip and climb out of their original slip planes. Wavy slip can occur in both transgranular and intergranular fracture modes. The fact that wavy slip occurs by a time dependent, thermally activated process in iron and nickel base alloys at elevated temperatures has significant impact on the frequency dependence of

fatigue. As the frequency or strain rate increases the degree of slip dispersal decreases, i.e., when the characteristic time constant associated with slip dispersal becomes larger in relation to the cycle time associated with deformation, slip becomes more concentrated on certain planes. It has in fact been observed that as cycling frequency increases, slip becomes similar to that observed at ambient temperatures in nickel base alloys. At higher frequencies as with lower temperatures, planar slip tends to dominate.

Fatigue life over the broad range of frequency from .033 Hz to 1000 Hz for Udimet 700 at 1400°F (760°C) is presented in References 8 and 9 and shown in Figure 41. From .033 Hz to 10 Hz, fatigue life at the given strain range increases by a factor of 100. Over this frequency range there are changes in the site of crack initiation. At the lowest frequency initiation occurs at surface connected grain boundaries and the initial mode of fracture in intergranular. As frequency increases in this range of frequency, intergranular cracking generally associated with creep and oxidation become less dominant giving way to transgranular fracture. At a frequency of 3 Hz, the fracture is almost entirely transgranular. With an increase in frequency from 10 to 1000 Hz, fatigue life is reduced by a factor of seven because of the concentration of deformation in fewer slip bands and the resulting accelerated crack initiation and propagation. Reference 9 suggests that the main reason for the reduced fatigue life beyond a frequency of 10 Hz may be associated primarily with the number of cycles required for crack initiation.

The nature of crack initiation has been shown to change with changing frequency. Stage I and Stage II are two classifications of fatigue crack initiation. Stage I crack initiation is favored by low temperatures and high frequencies, i.e., the same conditions that lead to planar slip. Low frequencies and high temperature on the other hand, favor Stage II crack initiation. Additional observations regarding the influence of frequency on crack initiation in nickel base alloys is given in Reference 10. Fatigue cracks in Udimet 700 at 1400°F (760°C) are shown to initiate in an intergranular mode from a surface intitation site at frequencies of 0.033 to 0.33 Hz. The crack then extends intergranularly along the surface to a depth of 1 to 3 grain liameters below the surface and then changes to a transgranular Stage II mode. At frequencies in the range 3 to 1000 Hz, crack propagation began in the Stage I transgranular mode and changed to a Stage II mode. It was also observed over the many specimens examined that low



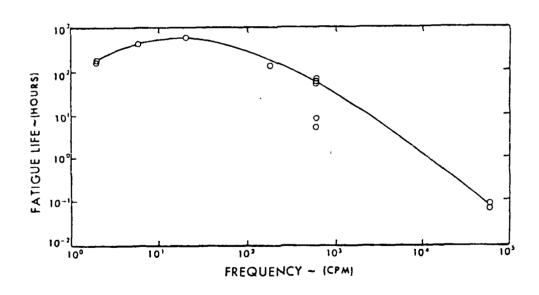


Figure 41 The Effect of Frequency on the Number of cycles and Time to Failure of **V**-700 at 1400°F (760°C) and a Stress Range of 85 Ksi. (9)

trequencies favor surface intergranular crack initiation and intergranular crack propagation. High frequencies on the other hand favor subsurface initiation at grain boundaries or twin boundary intersections and transgranular crack propagation.

A study of the influence of cyclic frequency on the fatigue properties of single crystal MAR-M-200<sup>(11)</sup> showed results similar to those for Udimet 700. Testing was performed on MAR-M-200 at frequencies from 0.033 Hz to 1030 Hz over the temperature range 1400°F (760°C) to 1800°F (982°C). The number of cycles to failure at 1400°F (760°C) and 1550°F (787°C) reached a peak in the range of 1 to 10 Hz. Stage I crack initiation was favored at the lower temperatures and higher trequencies and Stage II crack initiation at the higher temperatures and lower frequencies. At 1030 Hz crack initiation and propagation occurred entirely in the Stage I mode with facets corresponding to 110 slip planes. Generally, the amount of Stage I fracture varied according to temperature and frequency as shown in Figure 42. The nature of the fracture was attributed to degree of slip homogeneity. Stage I is favored by inhomegeous planar slip and Stage II is tavored by homogeneous slip. In almost all specimens of MAR-M-200 cracks initiated at subsurface micropores.

Clavel and Pineau<sup>(12)</sup> studied the effects of frequency and wave form on the fatigue crack growth of Alloy 718 (a nickel base superalloy) in the frequency range between  $5 \times 10^{-3}$  Hz and 20 Hz at  $298^{\circ}$ K and  $823^{\circ}$ K. The variation in fatigue crack growth rate with frequency that they observed at  $823^{\circ}$ K is summarized in Figure 43. Consistent with the observations on Udimet 700, Nimonic 90, and MAR-M-200 single crystals, the fatigue crack growth rate decreases with increasing frequency in the regime in which environmental and time dependent material deformation processes creep effects can operate. Fractography revealed that this decrease in fatigue crack growth rate (FCGR) is accompanied by a change in fracture mode from intergranular to transgranular. Suprisingly, they observed through TEM examination of the substructure that higher strain rates promoted more homogeneous plastic deformation while low strain rates favor inhomogeneous deformation and the formation of twins. The crystallographic aspects of the fracture surface observed at room temperature in the threshold regime is attributed to the decohesion along the twin deformation bands.

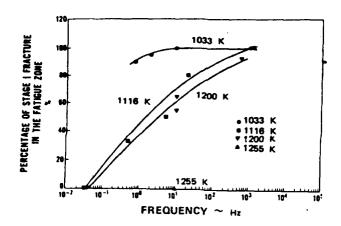


FIGURE 42 The Percentage of Stage I Fracture in the Fatigue Zone as a Function of Cyclic Frequency at Temperatures of 1033, 1116, 1200, and 1255 K. (9)

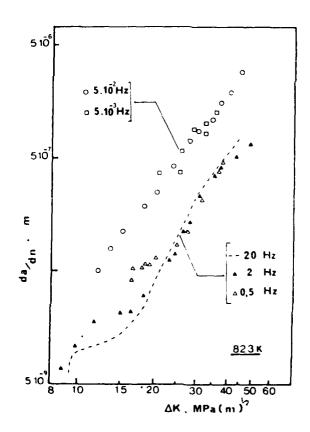


FIGURE 43 Variation of FCG Rate (da/dN) With Stress Intensity Factor ( $\Delta$ K) and Frequency ( $\checkmark$ ) at 823 K (Sinusoidal Load) for Inconel 718. (12)

A paper by Sullivan et. al. (13) discusses the effect of cycling frequency in the very low frequency regime for the nickel base superalloys Udimet 700 and MAR-M-200. Creep tests were performed on these materials at 955°C in air with periodic unloading. The time intervals between unloadings were on the order of 15 minutes to 5 hours. The main observation made regarding the effects of unloading is that it produced accelerated creep rate in both Udimet 700 and directionally solified MAR-M-200.

Scarlin<sup>(14)</sup> also studied the effect of frequency in the range  $10^{-4}$  to  $10^2$  Hz on the fatigue crack growth of two nickel base superalloys: Nimonic 105 at 750°C and IN 738 LC at 850°C. The results show the expected decreasing crack growth rate with increasing frequency.

The influence of environment on the frequency dependence of fatigue has also been investigated. For example, in the lower frequency regime (up to 1.7 Hz) Solomon and Coffin (15) studied the effect of frequency on the fatigue crack growth of A286 at  $1100^{\circ}$ F in both air and vacuum. They observed generally that the crack growth mode and frequency dependence of crack growth rate varied with frequency in the manner shown in Figure 44. This representation of the crack growth data shows that specimens tested in air and vacuum both have frequency regimes of intergranular and transgranular fracture but with different behavior in the lower cycle regime. Likewise, both air and vacuum tested specimens have a frequency above which the crack growth rate is independent of frequency. This study also shows that the dependence of crack growth rate (in growth per cycle) may be represented as follows.

$$\frac{dC}{dN} = \phi c (\Delta \varepsilon_{\rho})^{\alpha} v^{k-1}$$
 (1)

where  $\Delta \epsilon_{\rho}$  is the plastic strain range for the specimen used in their experiments, C is the measured crack length, dC/dN is the crack growth rate and  $\phi$ ,  $\alpha$ , and k are constants. Each regime shown in Figure 44 is characterized by a different value of k; the pure cycle dependent regime has a k value of 1.0.

The tests performed in vacuum generally have a lower growth rate than in air. The difference in fatigue crack growth rate, however, decreases with increasing frequency and the results converge at high frequency. These results suggest

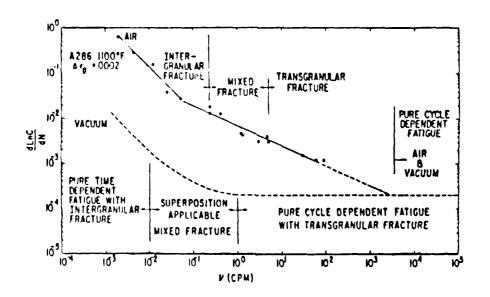


FIGURE 44 Schematic Comparison of the Air and Vacuum Crack Growth Behavior.

that as frequency increase the effect of environment reduces, or in effect the crack at higher frequencies out runs the processes of oxidation that degrade the fatigue crack growth properties.

Another study demonstrating the influence of environment on frequency effects in the fatigue crack growth involves 200 Maraging steel in a salt water environment and is reported in Reference 16. A significant frequency effect on the crack growth of this alloy in salt water in the frequency range 0.17 Hz and 3.3 Hz. At 3.3 Hz, it was found that the salt water solution had little effect on the crack growth rate as compared to the results in air. There was a factor of ten increase in crack growth rate when the frequency was reduced to 0.017 Hz. This along with the comparisons for different gaseous environments demonstrates that environment may be responsible for much of the frequency effect.

An important aspect of frequency effects in component life prediction is the effect of frequency on the threshold stress intensity factor range ( $\Delta K_{th}$ ). There are reports on the aspect for several materials. Mautz and Weiss<sup>(17)</sup> reported the effects of frequency on  $\Delta K_{th}$  for D6ac steel at room temperature for both air and argon environments. No frequency effects on threshold behavior were observed for an air environment between frequencies of 100 and 375 Hz. In dry argon, however, the results for 100 Hz were slightly higher than those at 375 Hz.

A very extensive study of fatigue crack growth properties of titanium alloys used in aircraft engine compressors was performed by Beyer, Sims and Wallace (3). Frequency effects up to 1000 Hz on the fatigue crack growth properties of Ti-6Al-2Sn-4Zr-6Mo, Ti-8Al-1Mo-1v, and Ti-6Al-2Sn-4Zr-2Mo were investigated at room, 600°F, 800°F, 900°F and 1000°F for several R ratios for crack growth rates down to the threshold regime. For the higher R ratios such as 0.5 and 0.7 there is a considerable reduction for all three alloys when the frequency is increased to 1000 Hz from 0.17 Hz at elevated temperatures. The threshold stress intensity factor likewise reduced on increasing the frequency to 1000 Hz. The variation in crack growth in the frequency range 0.017 Hz to 30 Hz was much less than that between 30 and 1000 Hz.

The highest test frequency in fatigue testing that we were able to find in the literature was that used by St. Stanzl and Mitsche (18) who performed crack

growth tests on 0.04%C steel, chromium steel 20A13 (0.29.C, 13%a), and pure molybdenum at 20 kHz. They conclude that their results in terms of crack growth rate versus  $\Delta K$  are similar to those for 10 Hz provided by another investigation.

Combined high cycle/low cycle loading has been investigated for several materials. The frequencies for the high cycle and low cycle components represented in these studies cover a very broad range in both loading components. At the lowest extreme in low cycle loading there is the low cycle frequency of zero with the high cycle frequency in the range that will with sufficient amplitude cause fatigue crack growth. This combined cycle interaction is often referred to as creep-fatigue interaction. (19-24) Another group of papers and reports (1, 25-28) deal with high cycle/low cycle interaction where the high and low cycle components correspond to those that are encountered in rotating machinery. The low cycle component has a cycle period on the order of seconds to several hundred seconds and the high cycle frequency ranges from 10 Hz to several thousand Hz.

Several studies have shown that load cycling can have an effect on the creep rate. Both increases and decrease in creep rate have been observed when cycling is applied. The softening has been attributed to and increased mobility of piled up dislocations as a result of the fatigue cycling assisting the dislocations to overcome obstacles and "friction" stress fields in the slip plane. The hardening effect has been explained in terms of migration of solute atoms or dispersed point defects towards free dislocations. Venkiteswaran et. al. (19) who studied the precipitation hardened alloy Inconel Alloy X-750 attributed the reduction in creep rate due to an applied fatigue cycle to the formation of complex dislocation tangles and vacancy condensation along dislocation lines. A change in fracture mode from intergranular to transgranular was also observed with the application of the 555 to 910 Hz fatigue loading.

Atanmo and McEvily<sup>(24)</sup> reported on the creep-fatigue interaction during crack growth of aluminum alloy 5052 at 400°F. Conducting tests with ramped loading and hold times ranging from 30 to 65 seconds, as well as with steady load, they-observed that cyclic-creep lifetimes can exceed creep lifetimes, perhaps as a result of the reversal of the creep process at the crack tip during the off-load period of the test.

There are several investigations of high cycle/low cycle interaction motivated by design considerations in rotating machinery such as generating plants, gas turbines and compressors. These studies all involve a low cycle loading component consisting of a trapezoidal waveform with a high frequency component applied during the upper level hold time as shown in Figure 45. Included in these studies are those performed at Portsmouth Polytechnic Institute on Ti-6Al-4V. (25-27) The dwell period for this test was 6.8 seconds. Testing was performed at room temperature with a high cycle frequency of 150 Hz. Fatigue crack propagation experiments with increasing load and high cycle load levels were undertaken using minor to major amplitude ratios (Q) of 0, 0.1, 0.2 and 0.3. Figure 46 shows the effect of amplitude ratio (Q) on the measured FCG rates.

A series of tests were conducted to determine the value of high frequency  $\Delta K$  under major-minor cycling corresponding to measureable influence of the high cycle component on crack growth. A step down procedure was used to determine the threshold for high cycle activity. Table 6 lists the conditions for the onset of minor cycle damage and onset of fast fracture. The authors also evaluated the appropriate manner of predicting crack growth rate under combined cycle loading. The two approaches to crack growth prediction evaluated by the authors are the linear summation of the major and minor cycle crack growth rates measured individually and the representation of the complex loading in terms of its RMS value. A comparison of the experimental results with the predicted results are shown in Figures 47 and 48. In these cases, the crack growth is dominated by the high cycle (minor cycle) loading and the prediction of both the linear summation and RMS representation are satisfactory.

Goodman and Brown<sup>(1)</sup> report on several combined cycle tests on alloy 718 at 649°C with a high cycld frequency of 10 Hz and the same loading profile as used in the studies of References 25 and the present study. Features similar to those found in the present study at 200 and 1825 Hz were observed including retardation in the low cycle dominated regime and the existence distinct high and low cycle dominated regimes. Also included in the program conducted by Goodman and Brown, tests with a high cycle frequency of 100 and 200 Hz.

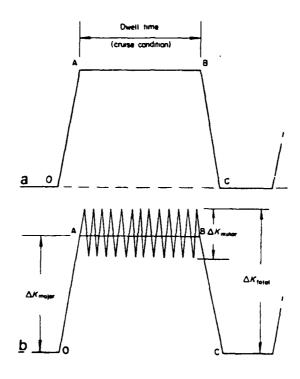


FIGURE 45 a - Major Cycles Only; b - Major and Minor Cycles (Minor/Major Amplitude Ratio Q =  $\Delta K_{minor}/\Delta K_{major}$ )

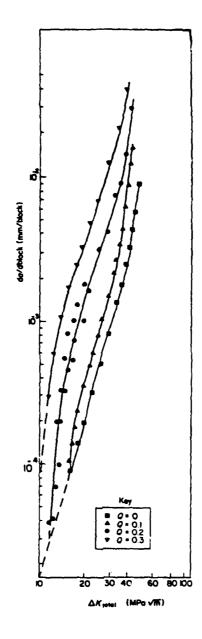
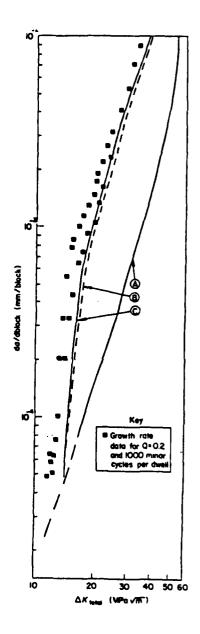


FIGURE 46 Effect of Amplitude Ratio on FCG Rates for Major and Minor Cycles. (25)

TABLE 6 Conditions for the Onset of Minor Cycle Damage and Onset of Fast Fracture (  $\Delta K$  Values in MPa  $\sqrt{m})$  .

Ampli- tude ratio	j-	Onset of minor cycle activity			Onset of fast fracture	
	R	$\Delta K_{\text{minor}}$	ΔK <sub>major</sub>	ΔK <sub>total</sub>	ΔK <sub>minor</sub>	$\Delta K_{\text{total}}$
0.02	0.982	1.5	75.0	75.8	1.3	63.1
0.04	0.965	1.6	40.0	40.8	2.5	63.1
0.1	0.914	1.7	17.0	17,9	6.0	63.3
0.2	0.835	2.1	10.5	11.6	11.6	63.6
0.3	0.762	2.3	7.7	8.8	16.7	63.9



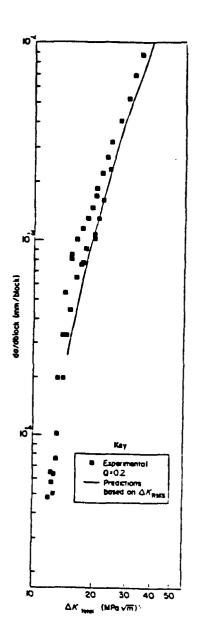


FIGURE 47 Linear Summation of FCG Rates (Damage: A-Associated With Applied Major Cycle: B-Associated With Applied Minor Cycles: C-Given by Summation of Major and Minor Cycle Damage). (25)

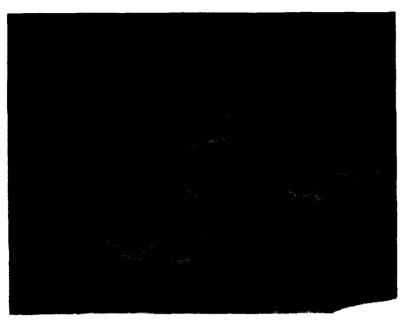
FIGURE 48 Analysis of Major-Minor Fatigue Crack Growth Rates in Terms of  $\Delta \kappa_{RMS}$  (25)

## B. Evaluation of Fatigue Crack Growth Mechanisms Under Combined Cycle Loading

The study of Venkiteswaran et. al.,  $^{(19)}$  reports the results of creep testing with a superimposed small vibratory stress on the axial creep behavior of a high temperature nickel base alloy, Inconel X-750. This work demonstrated that the creep rate was lower and rupture life higher by an order of magnitude when a 500 to 900 Hz vibratory stress was applied transverse to the axial creep load. This effect was attributed to the formation of complex dislocation tangles, vacancy condensation along dislocation lines and crack tips and also a change in fracture mode from purely intergranular fracture to a mixture of intergranular, fatigue and cleavage modes. It was suggested that the application of the high frequency loading, therefore, made creep crack propagation more difficult along the matrix containing  $\gamma'$  precipitates. Since the heat treated Inconel 718 used in this study likewise contains  $\gamma'$  (Ni3 Al-Ti) as well as  $\gamma''$  (Ni3Cb) precipitates, this mechanism could apply in the present study. The changing mode of fracture observed by them is consistent with the fractographic features of the combined cycle crack growth specimen that we investigated.

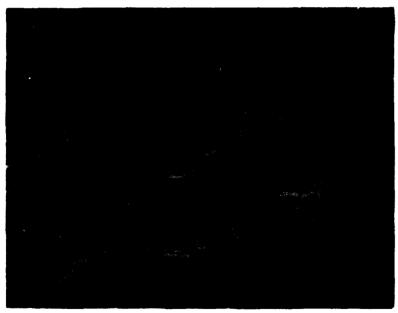
Fractographic examination was performed on specimens subjected to 200 Hz cyclic load in order to obtain information regarding fracture mechanisms. Areas on a 200 Hz specimen were examined by Scanning Electron Microscopy (SEM) and Scanning Transmission Electron Microscopy (STEM). All of these photographs correspond to specimen 28. The fracture surface includes areas corresponding to creep crack growth with no high frequency loading and areas corresponding to combined high/low cycle loading with crack growth both in the low and high cycle dominated regimes. The fracture surfaces of these regions show distinct differences. Figures 49 and 50 show SEM photomicrographs of the purely low cycle and combined cycle regions respectively. The purely low cycle region shows intergranular fracture typical of creep crack growth. With the application of a high frequency load range at a level that maintained the low cycle dominated behavior, the fracture becomes predominantly transgranular with the appearance of fatigue striations.

Replicas were taken of the fracture surface and subsequently shadowed with chromium and coated with a film of carbon. These replicas were then examined in an SEM with a transmitted beam. The resulting photomicrographs for a region of low cycle loading only, are shown in Figure 51. The intergranular nature of the



1000x E5631

FIGURE 49: Scanning Electron Microscope (SEM) photomicrograph of a region in which only low cycle loading was applied. (Specimen #28)



1000x E5636

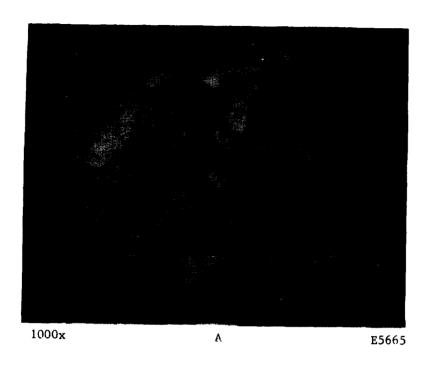
FIGURE 50: Scanning Electron Microscope (SEM) photomicrograph of a region in which combined cycle loading (with 200 Hz high cycle load) was applied.

fracture is clearly shown in these photomicrographs. In Figure 52 are shown the STEM photomicrographs for a region which had experienced combined cycle loading. The striation pattern is this predominantly transgranular fracture seems to show grouping of striations. At the higher magnification of 10,000x the pattern appears to be obscured, probably by oxidation at the test temperature of 649°C.

Additional STEM photomicrographs were made on a specimen tested with a high cycle frequency of 1825 Hz (specimen 67). Without high frequency cycles applied, the fracture surface showed the expected intergranular fracture. Figure 53 shows the fracture surface in the low cycle dominated regime where the high cycle  $\Delta K$  is large enough to cause retardation. A striation pattern is apparent. Figure 54 shows the fracture surface well into the high cycle dominated regime. The striation pattern in this region is more pronounced and shows a greater spacing corresponding to the increased crack growth.

The relationship between fatigue crack growth and high cycle  $\Delta K$  for constant low cycle  $\Delta K$  show three regimes. At the lower limit of  $\Delta K_{HC}$  the low cycle loading dominates the rate of fatigue crack growth. In an intermediate range of  $\Delta K_{HC}$ , the high cycle loading causes a retardation of the crack growth rate. At the highest values of  $\Delta K_{HC}$ , crack growth rate is dominated by the high cycle loading with crack growth determined by the number of high frequency cycles. The low and high cycle dominated regimes are distinct but the transition between the two regimes is obscured by the retardation effect. The behavior of alloy 718 at 649°C revealed by this study is similar to that shown by Goodman and Brown  $^{(1)}$  who studied the interactive effect of this alloy at 649°C with a high cycle frequency of 10 Hz. In their investigation, distinct low and high cycle dominated regimes as well as a regime of  $\Delta K_{HC}$  where retardation occurred were also apparent. The investigation of Powell et. al.,  $^{(25)}$  on Ti-6-4 showed regimes of  $\Delta K_{HC}$  where the high cycle loading was either active or inactive, but a retardation effect was not apparent.

The retardation effect was unexpected and an experiment was carried to gain insight into its origin and characteristics. The experiment summarized in Figure 31 shows the rate (with respect to crack length) at which the retardation effect develops and also the rate at which it relaxes. There seems to be a crack growth interval of about 1mm (0.0394 inches) required for the retardation effect



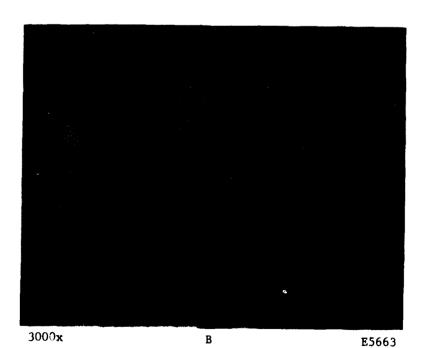


FIGURE 51: Scanning Transmission Electron Microscope (STEM) photomicrographs of a region in which only low cycle loading was applied.

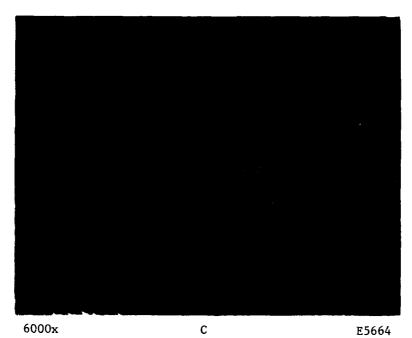
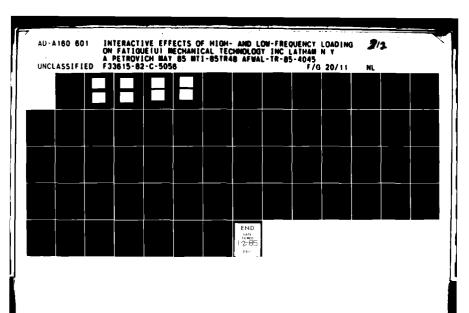
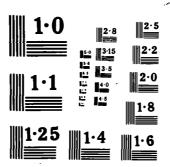
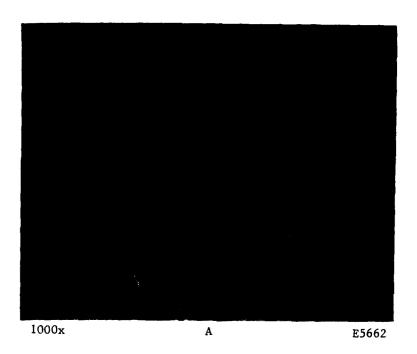


Figure 51 (Cont'd) Scanning Transmission Electron Microscope (STEM) photomicrographs of a region in which only low cycle loading was applied.







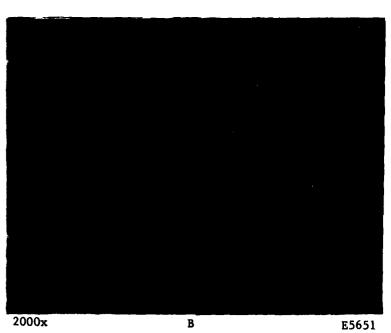
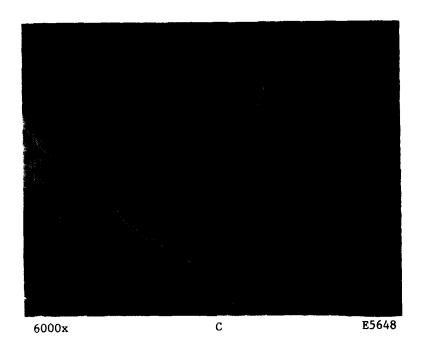


FIGURE 52: Scanning Transmission Electron Microscope (STEM) photomicrographs of a region in which combined cycle loading (with 200 Hz high cycle load) was applied.



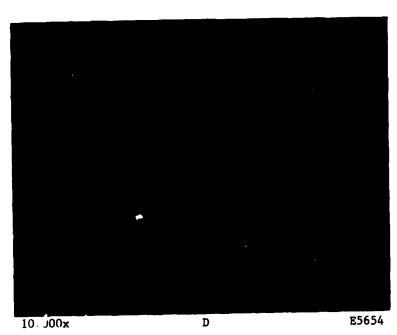
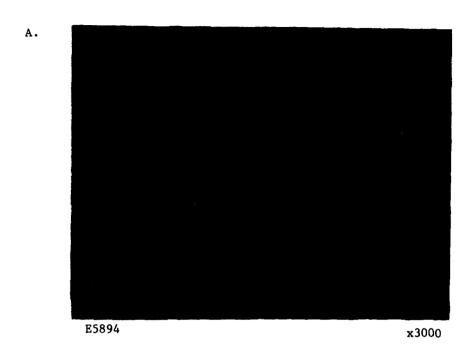


Figure 52 (Cont'd) Scanning Transmission Electron Microscope (STEM) photomicrographs of a region in combined cycle loading (with 200 Hz high cycle load) was applied.



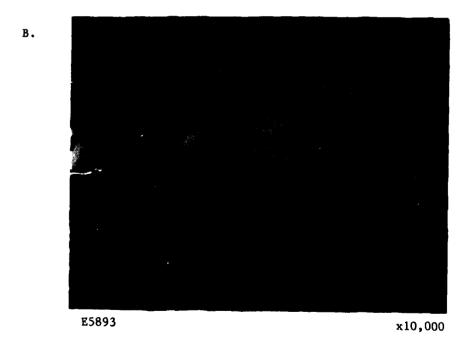
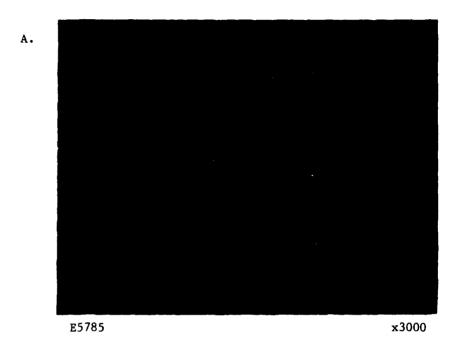


Figure 53 STEM Photomicrographs of a Region on the Fracture Surface of Specimen #67 Corresponding to the Low Cycle Dominated Regime where the High Cycle K is Large Enough to Cause Retardation



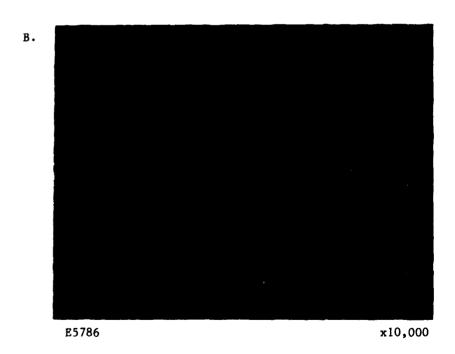


Figure 54 STEM Photomicrograph of a Region on the Specimen #67 Fracture Surface Where the High Cycle Component Dominates Crack Growth.

to subside. The plastic zone size associated with the crack, however, is a small function of this length.

The size of the plastically deformed region (R) ahead of the crack as given by the Dugdale model (28) neglecting the effects of creep is:

$$R = \left\{ \sec[1/2\pi(\sigma/\sigma y)] - 1 \right\} a$$

where  $\sigma$  is the applied stress,  $\sigma y$  is the yield strength and a is the half crack length.

As calculated using this expression at the crack length corresponding to the retardation relaxation in Figure 31 and assuming a yield strength 980 MN/m² (140 ksi) the plastic zone size is 0.20mm (0.008 inches). A possible explanation of the fact that the affected region is considerably larger than the calculated plastic zone length is that creep stress relaxation results in a larger characteristic zone where structural changes important to retardation effects occur. Reference 29 demonstrates that crack tip stresses can be modified significantly by creep. The most significant influence of creep relaxation as shown by Reference 29 is the reduction of the stress gradient beyond the crack tip, i.e., the development of a more uniform distribution of stress in a region that includes the above calculated "plastic zone" and an area further from the crack tip. However, it is unlikely that creep can have such a pronounced influence on the crack stress distribution.

An alternative explanation of the long relaxation interval is suggested by the study of Venkiteswaran (19) that showed that the high frequency loading affects the creep rate versus stress constitutive properties. The modification of the creep rupture processes may in turn modify the residual plastic deformation remaining in the wake of the advancing crack (crack closure). (30) Such a concept would allow the possibility of the effect persisting well beyond the above calculated plastic zone without postulating a significant modification in the crack tip stresses due to creep relaxation effects. The crack growth interval of four or five times the plastic zone size is in fact characteristic of the development of closure effects.

cycle frequency is applied. However, when the retardation effect occurs with this lower high cycle frequency, a longer time period is required to reach the minimum crack growth rate. This would be expected if the retardation effect is related to the accumulated number of high frequency cycles.

A feature of the high cycle loading revealed by this and other studies is that there is a sharply defined value of transition  $\Delta K$  ( $\Delta K_{\rm tr}$ ) associated with the dominance of  $\Delta K_{\rm HC}$  for crack growth under combined cycle loading. For the range of conditions investigated, the influence of high cycle loading on crack growth below  $\Delta K_{\rm tr}$  can essentially be ignored. This is consistent with the observations of Powell et. al. (25) who performed combined cycle crack growth experiments on Titanium - 6 - 4 at ambient temperatures with a high cycle frequency of 150 Hz and with all of the combined cycle testing results of Goodman and Brown (1) performed on Inconel 718 at 649°C for a high cycle frequency of 10 Hz. Furthermore, there is little variation of this transition  $\Delta K$  with frequency distinguishable beyond the  $\Delta K_{\rm tr}$  variation intrinsic to the material.

Considering the various features of the crack growth rate beyond  $\Delta K_{\text{tr}}$ , i.e., that growth rate depends on number of cycles, that it is sharply defined by a threshold value, and that is shows a relationship between growth rate and  $\Delta K_{\text{HC}}$  similar to that for stage I crack growth leads to the conclusion that it could be represented by a relationship of the form:

$$\frac{da}{dN} = C \left(\Delta K_{HC} - \Delta K_{tr}\right)^{m}$$

where C is a constant and da/dN is crack growth rate in terms of crack extension per high frequency cycle. This relationship has been used to describe crack growth in the threshold regime (stage I) with a constant R ratio (Kmin/ $_{\rm Kmax}$ ). The R ratio in the high cycle dominated regimes in the experiment conducted in this study, varies a small amount since they correspond to increasing  $\Delta K_{\rm HC}$  and constant  $\Delta K_{\rm LC}$  tests. The  $\Delta K_{\rm tr}$  is the above expression is expected to vary with low cycle  $\Delta K$  and perhaps hold time but as shown by the present investigation it is essentially invariant with respect to frequency. This feature is helpful to the design since an acceptable level of high cycle loading can be established without concern for the frequency of the superimposed high frequency load.

is essentially invariant with respect to frequency. This feature is helpful to the design since an acceptable level of high cycle loading can be established without concern for the frequency of the superimposed high frequency load.

The investigation of Gell and Leverant  $^{(8)}$  showed a pronounced influence of frequency on fatigue life of nickel base alloys in the frquency range of 10 to 1000 Hz. In this range they observed that fatigue life decreases with increasing frequency. Comparing the results of the present investigation with those of Goodman and Brown,  $^{(1)}$  there is little variation in  $\Delta K_{tr}$  over the frequency range to 2000 Hz. This fact and the fact that the crack growth rate per cycle versus  $\Delta K_{HC}$  beyond  $\Delta K_{tr}$  does not increase with increasing frequency leads to the conclusion that the decreasing fatigue life with increasing frequency observed by Gell and Leverant  $^{(8)}$  is associated primarily with crack initiation.

## C. Consideration of High/Low Cycle Interactions in Crack Growth Life Prediction of Engine Systems

Attention has been devoted recently to the effects of gas turbine engine load spectra on crack propagation. This is a result of increased performance requirements for U.S. Air Force gas turbines and the resulting high operating stresses and severe service environments experienced by gas turbine components. Many of the investigations are associated with the development of the advanced life management concept and focus on engine disks.

An important aspect of life prediction under engine loading spectra is the interaction of the low and high cycle components in crack growth of turbine disks. The cycle period associated with the low frequency cycle (low cycle) loading is on the order of seconds to several hundred seconds. A wide range of loading rates and load levels may also be involved in the low cycle loading. The high frequency cycle (high cycle) loading would typically involve frequencies on the order of hundreds to several thousand hertz. Important to accurate life prediction is establishing the manner in which each of these features of the engine disk loading profile contribute to crack growth and how these features interact. The specific aspects of combined cycle loading that must be addressed are the following:

- Establishment of the limits of high cycle loading under which the disk can be safely operated.
- How cumulative damage rules should be applied when combined high cycle/low cycle loading contribute to crack growth.
- The degree to which the high cycle and low cycle loading influence each others contribution to crack growth.

There are previous studies of load spectrum interaction in crack growth of aircraft engine components that deal with periodic overloads, overload/underload combinations and periods of sustained load interspersed with relatively constant amplitude loading. An example of such a study in that of Macha et. al. (31) which considered these effects on IN-100 and evaluates the applicability of crack growth rate models for engine complex loading spectra. Another study that addresses the effects of flight loading in military gas turbine operation on the fatiuge crack growth of IN-100 and Waspaloy is summarized in References 32 and 33. This study addresses the effect of overload ratio and the effect of the number of cycles between overloads.

The simplest approach to crack growth prediction is a linear summation of crack growth on a cycle by cycle basis for the given loading profile. However, this approach has been shown to be inadequate for many situations of variable amplitude loading where retardation or acceleration can result from certain sequences of loading. Various approaches have been established to account for these effects including models based on crack closure (34) and the interactions in the crack yield zone (35).

In the present study, the applicability of a linear summation of high and low cycle crack growth contribution in predicting combined high/low cycle crack growth was investigated for Alloy 718 with a high cycle component of 200 and 1825 Hz. Figures 55 and 56 show a comparison between a combined cycle test result and a linear summation of crack growth rate calculated from crack growth data for the low and high cycle contributions measured individually. The manner of summing the individual high and low cycle components is shown schematically in these figures. The individual contributions were measured in an experiment with increasing  $\Delta K_{HC}$  superimposed on steady (not cycled)  $\Delta K_{LC}$  and in an experiment with pure low cycle loading with a triangular waveform and an R ratio of

0.1. For a high cycle frequency of 200 Hz, a low cycle  $\Delta K$  of 30 MPa  $\sqrt{m}$  and a hold time of 180 seconds, Figure 55 shows a reasonable correspondence between actual results and those predicted from a linear summation in the high cycle dominated regime only. For the case of a high cycle frequency of 1825 Hz on the other hand there appears to be deviation in the high cycle dominated regime associated with a difference in  $\Delta K_{\rm tr}$  and a fair correspondence of crack growth rate in the low cycle dominated regime. These trends, however, are not necessarily representative. In this study, as well as that of Goodman and and Brown (1) a substantial variation in low cycle crack growth rate was apparent for tests carried out under identical conditions. Likewise, for given values of  $\Delta K_{\rm LC}$  hold time and frequency, a variation in  $\Delta K_{\rm tr}$  of 20% was apparent. The results of linear summation show a deviation from the combined cycle data that is in the range of variation in crack growth rate behavior for a given set of combined cycle parameters.

With some qualifications, a linear summation provides an adequate representation of combined cycle crack growth rate. In applying the linear summation approach to design, one must be aware of the fact that the low cycle crack growth rate, the retardation behavior and  $\Delta K_{\mbox{\scriptsize tr}}$  can vary. An appropriate design curve to account for combined crack growth rate is the dashed line construction of Figure 55. The upper bound on low cycle crack growth is the horizontal dashed line. The upper bound on high cycle dominated crack growth is represented by the dashed line on the right side of the diagram. Together, the two curves define an upper bound on crack growth rate in the low cycle dominated, the retardation, and the high cycle dominated regimes. The crack growth rate predicted in the retardation regime would be a significant over estimate. However, this is necessary because the extent of retardation is not easily predicted and its benefit should, therefore, be ignored. Another important factor that must be kept in mind in applying a linear summation rule is that the relationship between crack growth rate and high cycle loading with a high level of mean load (i.e., large enough to cause creep crack growth) is not necessarily unique. For nickel base alloys, crack growth resulting from steady loads has been shown to exhibit non equilibrium behavior. In the present study, this effect was apparent when a small high cycle component was superimposed on a large steady load.

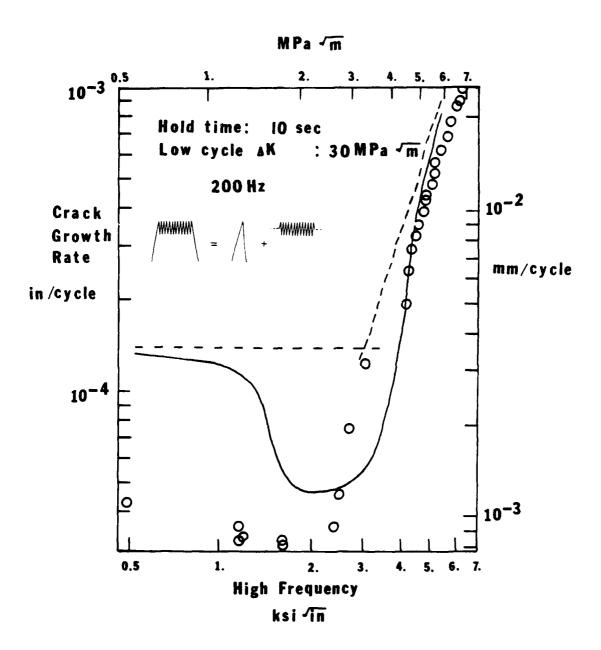


Figure 55 Comparison of Data (Points) with Growth Rate Predicted (Line)
From a Linear Summation of Uncycled 1825 Hz High Cycle Data
and Pure Low Cycle Data

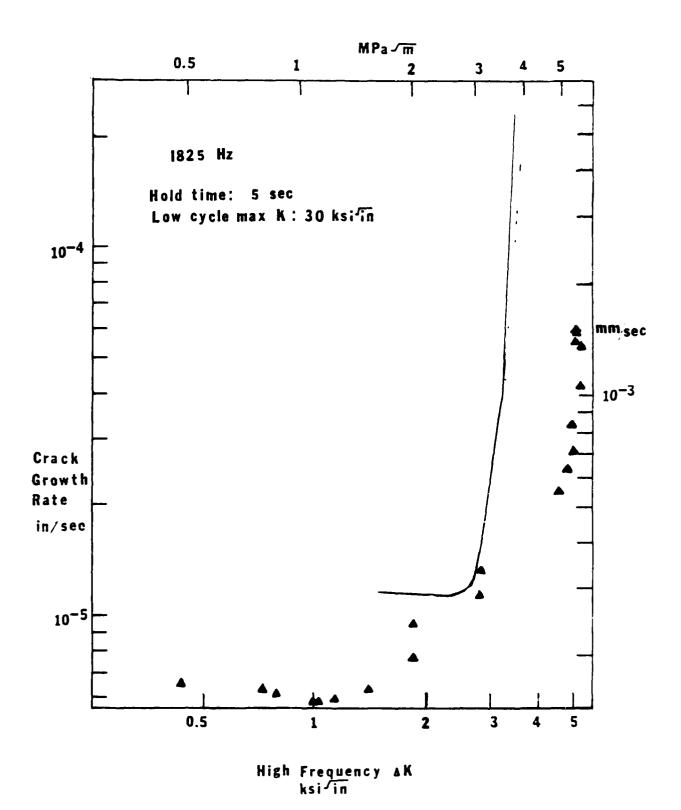


Figure 56 Comparison of Data (Points) With Growth Rate Predicted (Line) From a Linear Summation of Uncycled 1825 Hz High Cycle Data and Pure Low Cycle Data

## **V CONCLUSIONS**

The crack growth rate for Alloy 718 at  $649^{\circ}\text{C}$  was measured for combined cycle loading over ranges of low cycle  $\Delta K$ , high cycle  $\Delta K$ , low cycle hold time, and high cycle frequency. Several interesting trends in combined cycle crack growth were revealed. Generally, at lower values of high cycle  $\Delta K$ , crack growth is dominated by the low cycle components of the load spectrum and with a sufficient level of high cycle loading, crack growth is determined predominantly by the accumulated number of high frequency cycles. These two features of combined loading crack growth behavior were consistent and straight forward. The nature of the transition from the low cycle to high cycle regime, however, depends significantly on the value of low cycle  $\Delta K$  and to a lesser extent on high cycle frequency and low cycle hold time. The transition from low to high cycle dominated crack growth is obscured by a retardation effect.

This interactive effect is apparent under combined cycle loading generally at low values of  $\Delta K$  and values of  $\Delta K_{HC}$  in a range between the purely low and high cycle dominated regimes. Fractographic examination reveals that it is associated with a change in crack growth mechanism from one characterized by intergranular fracture for pure low cycle loading to transgranular fracture for combined cycle loading. The degree of crack growth retardation appears to decrease with increasing low cycle  $\Delta K$  and also decreases with increasing high cycle frequency. Another interesting featrue associated with the retardation effect is that a crack growth interval of several plastic zone sizes is required for its development or relaxation. Considering this fact and the fact that it becomes increasingly less pronounced with increasing  $\Delta K_{LC}$ , leads to the conclusion that the retardation effect is associated with a change in the extent of plastically deformed material left in the wake of the advancing crack when high cycle loading is applied.

The slope of the log (crack growth rate per unit time) versus log  $\Delta K_{HC}$  curve beyond  $\Delta K_{tr}$  increases with increasing frequency as one would expect for a situation where crack growth rate in the high cycle regime is dependent on the number of cycles. The shape of the curve in the high cycle dominated regime is similar to that experienced for near threshold behavior (Stage I) observed in constant R ratio tests. (The R ratio of the high cycle loading in these studies varies with  $\Delta K_{HC}$ ).

A crack growth rate prediction based on the linearly summed contributions for the high and low cycle components of loading correlate well in the case of a high cycle frequency of 200 Hz with some deviation in the low cycle dominated regime. For the case of 1825 Hz some deviation was observed for the high cycle dominated regime. This may be associated with the intrinsic variation in  $\Delta K_{tr}$ .

## **BIBLIOGRAPHY**

- Goodman, R.C. and Brown, A.M., "High-Frequency Fatigue of Turbine Blade Materials", Report #AFWAL-TR-82-4151 (Materials Laboratory, Air Force Wright Aeronautical Laboratories), Contract No. F33615-79-C-5108, October 1982.
- 2. Ling descriptive brochures of electrodynamic shakers, Ling Electronics Inc. 1525 South Manchester Ave. Anaheim, California 92803.
- 3. Beyer, J.R., Sims, D.L., Wallace, R.M., "Titanium Damage Tolerant Design Data for Propulsion Systems", Air Force Materials Laboratory Report, AFML-TR-77-101.
- Instron Limited, "Major/Minor Cycling System", Instron Limited, Coronation Rd, High Wycombe, Bucks HP123SY.
- 5. Ling Akashi brochure on 2000 Hz hydraulic system.
- MTS Systems Corporation, "Dynamic High Frequency Test System", Application note, MTS Systems Corporation, Box 24102, Minniapolis, Minnesota 55424.
- 7. Thomas Lagnese, private communication.
- Gell, M., Leverant, G.R. "Mechanisms of High-Temperature Fatigue," ASTM, STP 520, 1973, p.37.
- 9. Gell, M. and Organ, F.E., "The Effect of Frequency on the Elevated Temperature Fatigue of a Nickel-Base Superalloy, "Metall. Trans., Vol.2, April 1971, p.943.
- 10. Gell, M., Leverant, G.R. and Wells, C.H., "The Fatigue Strength of Nickel-Base Superalloys," Achievement of High Fatigue Resistance in Metals and Alloys, ASTM STP 467, American Society for Testing and Materials, 1970, pp.113-153.
- 11. Leverant, G.R. and Gell, M. "The Influence of Temperature and Cycle Frequency on the Fatigue Fracture of Cube Oriented Nickel-Base Superalloy Single Crystals" <u>Metallurgical Transaction A</u>, Vol.6A, p.367, Feb. 1975.
- 12. Clavel, M. and Pineau, A., "Frequency and Wave Form Effects on the Fatigue Crack Growth Behavior of Alloy 718 at 298°K and 823°K", Metal-lurgical Transactions A, Vol.9A, p.471 (1978).
- 13. Sullivan, C.P., Webster, G.A., and Piearcey, B.J., "The Effect of Stress Cycling on the Creep Behavior of a Wrought Nickel-Base Alloy at 955°C", Journal of the Institute of Metals, Vol.96, p.274, (1968).
- 14. Scarlin, R.B., "Effects of Loading Frequency and Environment on High Temperature Fatigue Crack Growth in Nickel-Base Alloys", <u>Fracture</u> 1977, Vol.2, p.849, Proceedings of ICF4, Waterloo, Canada, June, 1977.

- 15. Solomon, H.D., and Coffin, L.F., Jr., "Effects of Frequency and Environment on Fatigue Crack Growth in A-286 at 1100°F", ASTM STP 520, 1973, p.112.
- 16. Eisenstadt, R. and Smail, D.L. "The Effect of Frequency on Cyclic Crack Growth in 200 Managing Steel in a Salt Water Environment", Fracture 1977, Vol.2, p.911, Proceedings of ICF4, Waterloo, Canada, June, 1977.
- 17. Mautz, and Weiss, "Mean Stress and Environmental Effects on Near Threshold Fatigue Crack Growth, "Cracks and Fracture, ASTM STP601, American Society for Testing and Materials, 1976, pp.154-168.
- 18. St. Stanzl, and Mische, R., "High Frequency Fatigue of Metals, Crack Initiation and Propagation", Fracture, 1977, Vol.2. p.249, ICF4, Waterloo, Canada, June, 1977.
- 19. Venkiteswaran, P.K., Ferguson, D.C. and Taplin, D.M.R., "Combined Creep-Fatigue Behavior of Inconel Alloy X-750", <u>Fatigue at Elevated Temperatures</u>, ASTM STP520, American Society for Testing and Materials, 1973, pp.462-472.
- 20. Davies, P.W. and Wilshire, B., "Some Observations on the Creep and Fracture of Nimonic 80A Under Combined Creep/Fatigue Conditions", Journal of the Institute of Metals, Vol.97, p.15, (1969).
- 21. Price, A.T. "Creep-Fatigue Behavior of Polycrystaline Zinc", Journal of the Institute of Metals, Vol. 95, p.87, (1967).
- 22. Melika, A.H. and Evershed, A.V., "The Dependence of Creep Behavior on the Duration of a Superimposed Fatigue Stress", <u>Journal of the Institute of Metals</u>, Vol.88, p.411.
- 23. Kamel, R. and Bessa, F.A., "Effect of Superimposed Small Vibrations on the Static Creep Behavior of Polycrystalline Zinc", Acta Metallurgical, Vol.13, p.19, (1985).
- 24. Atommo, P.N. and McEvily, A.J., Jr., "Creep-Fatigue Interaction During Crack Growth", Fatigue at Elevated Temperatures, ASTM STP 520, American Society for Testing and Materials, 1973, pp.157-165.
- 25. Powell, B.E. Duggan, T.V., and Jeal, R.H., "The Influence of Minor Cycles on Low Cycle Fatigue Crack Propagation", International Journal of Fatigue, Vol.4, No.1, (1982).
- 26. Powell, B.E., "The Onset of Minor Cycle Activity", Interim report on Contract "AFOSR-82-0077 (Air Force Office of Scientific Research), Portsmouth Polytechnic, Portsmouth, U.K., March, 1982
- 27. Powell, B.E., and Henderson I., "Predicting Fatigue Crack Growth Rates", Interim report on contract #AFOSR-82-0077, (Air Force Office of Scientific Research), Portsmouth Polytechnic, Portsmouth, U.K., June, 1982.
- 28. Dugdale, D.S., "Yielding of Steel Sheets Containing Slits", J. Mech. Phys. Solids, (1960) p.100.

- 29. To, K.C., "A Phenomenological Theory of Subcritical Creep Crack Growth Under Constant Loading in an Inert Environment", "International Journal of Fracture", Vol.11, No.4, August 1975, p.641.
- 30. Elber, W., "Fatigue Crack Closure Under Cyclic Tention". "Engineering Fracture Mechanics, Vol.2, 1970, pp.37-45.
- 31. Macha, D.E., Grandt, A.F., Jr., and Wicks, G.J., "Effects of Gas Turbine Engine Load Spectrum Variables on Crack Propagation, "Effects of Load Spectrum Variables on Fatigue Crack Initiation and Propagation, ASTM STP 714. D.F. Bryan and J.M. Potter, Eds., American Society for Testing and Materials, 1980, pp.108-127.
- 32. Larsen, J.M. and Annis, C.G., Jr., "Observation of Crack Retardation Resulting from Load Sequencing Characteristic of Military Gas Turbine Operation", Effect of Load Spectrum Variables on Fatigue Crack Initiation and Propagation. ASTM STP 714, D.F. Bryan and J.M. Potter, Eds., American Society for Testing and Materials, 1980, pp.91-107.
- 33. Larsen, J.M., Schwartz, B.J., and Annis, C.G., "Cumulative Damage Fracture Mechanics Under Engine Spectra", Air Force Materiasl Laboratory Report II AFML-TR-79-4159, January, 1980.
- 34. Newman, J.C., Jr., "A Crack-Closure Model for Predicting Fatigue Crack Growth Under Aircraft Spectrum Loading", J.B. Chang and C.M. Hudson, eds., ASTM STP748, 1981, pp.52-84.
- 35. Johnson, W.S., "Multi-Parameter Yield Zone Model for Predicting Spectrum Crack Growth, "Methods and Models for Predicting Fatigue Crack Growth Under Random Loading, J.B. Chang and C.M. Hudson, eds., ASTM STP748, 1981, pp.85-102.

## APPENDIX A

PERFORMANCE OF HIGH FREQUENCY SERVO-HYDRAULIC SYSTEM

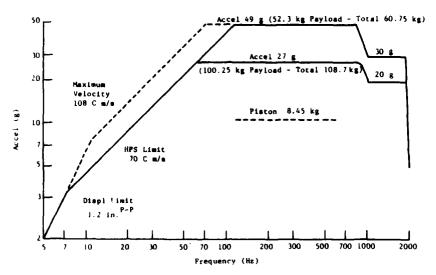


Figure A-1 HVIO-1.2-37/5 Test Data (Vertical)

Maximum Performance (ront Mount Type)

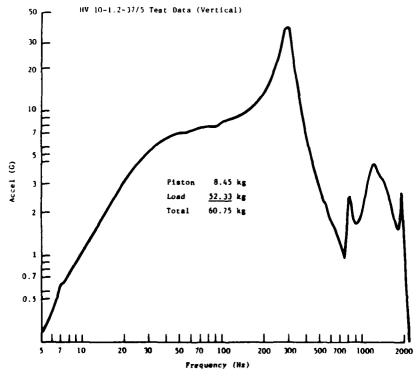


Figure A-2 Front Mount Type Constant Input-Frequency Response

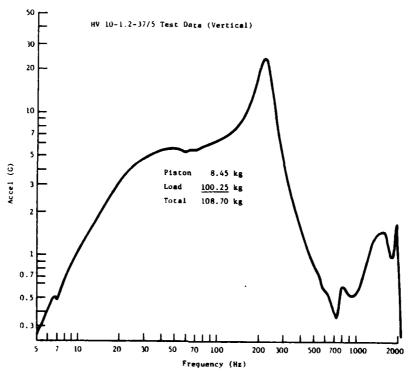
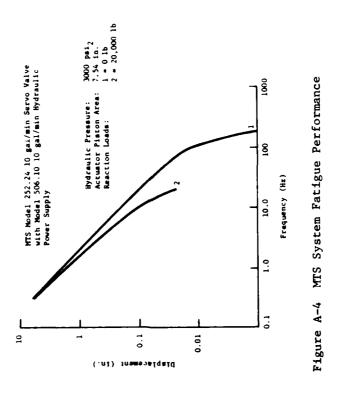


Figure A-3 Front Mount Type Constant Input-Frequency Response



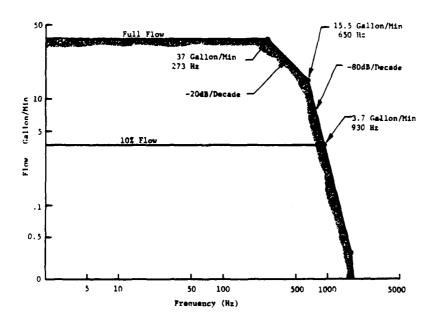


Figure A-5 Flow Versus Frequency for Akashi 37 gpm servo-valve

## APPENDIX B

DATA PLOTS FOR ALL TESTS

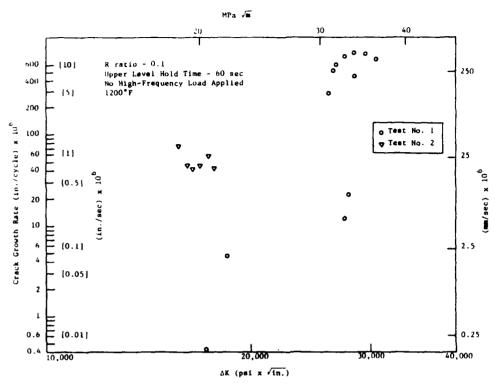


Figure B-1 Test No. 1 and No. 2

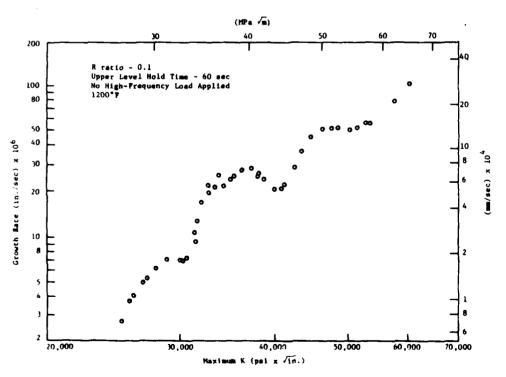
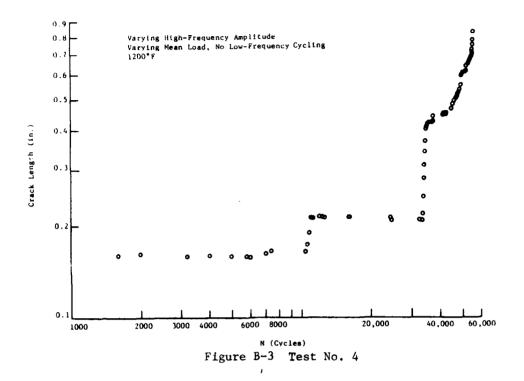
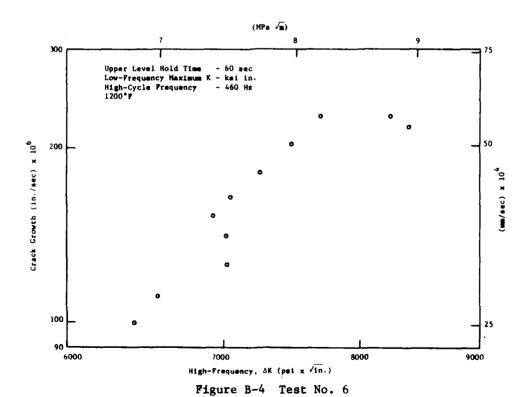


Figure B-2 Test No. 3





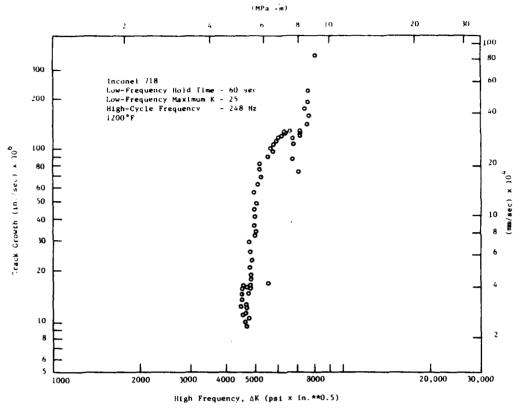


Figure B-5 Test No. 7

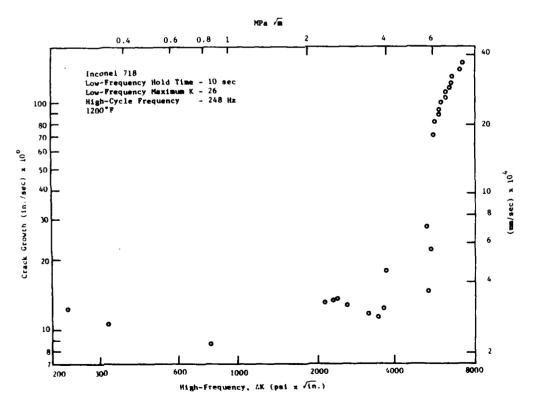


Figure B-6 Test No. 8

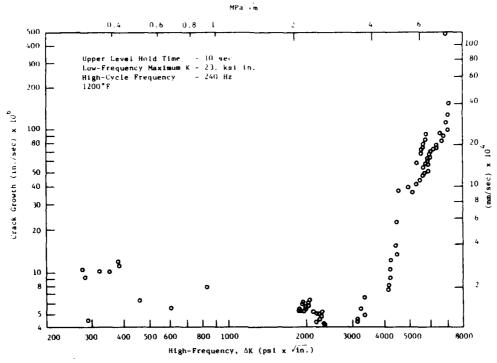


Figure B-7 Test No. 9

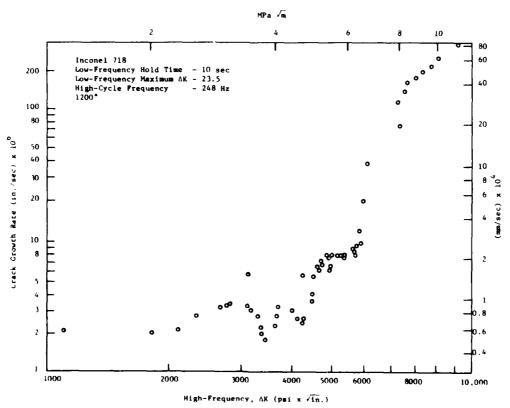


Figure B-8 Test No. 10



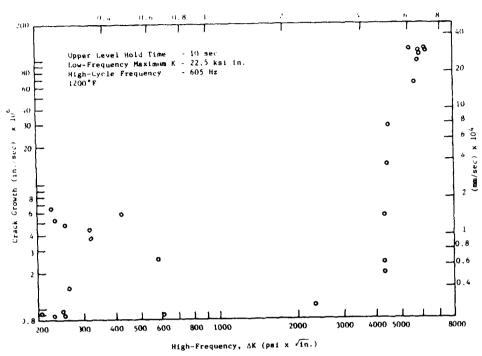


Figure B-9 Test No. 11

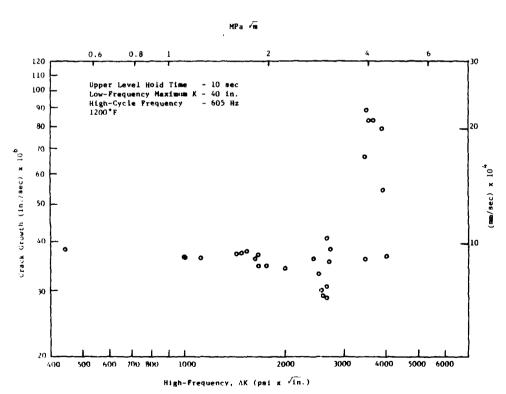
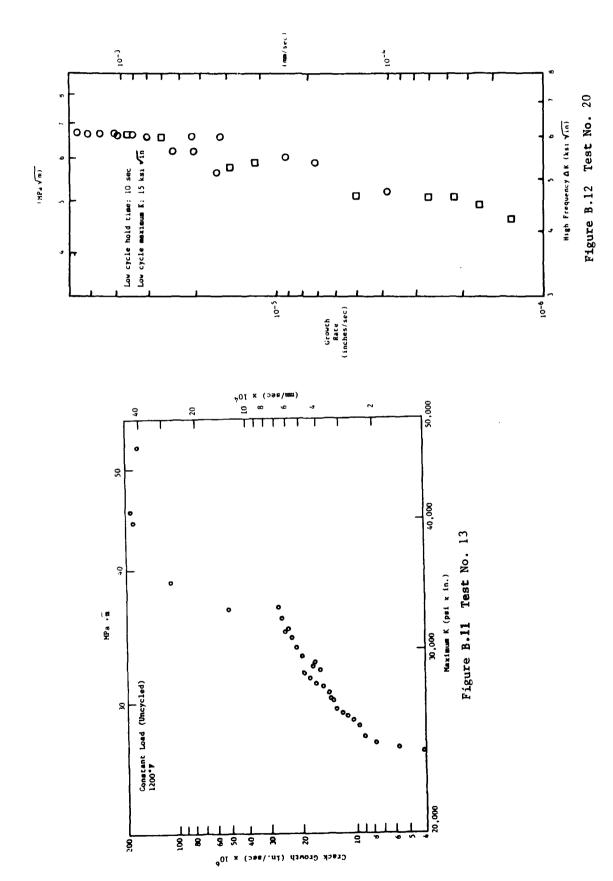
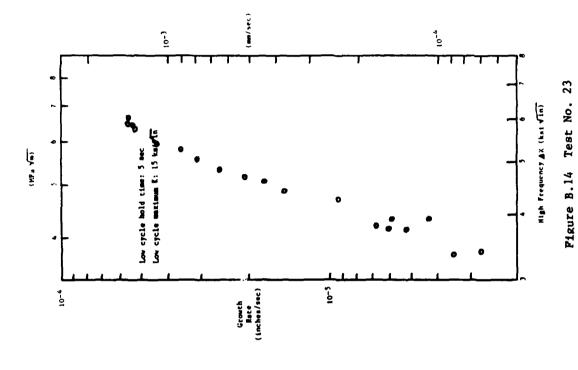
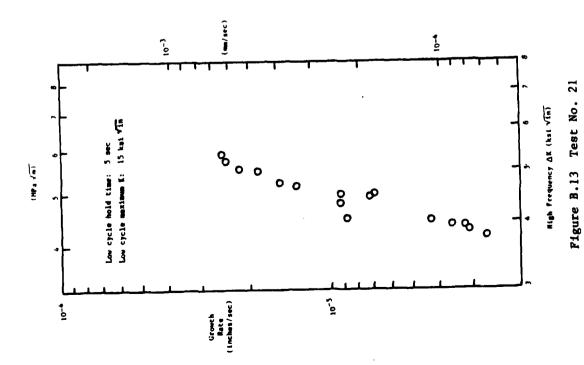
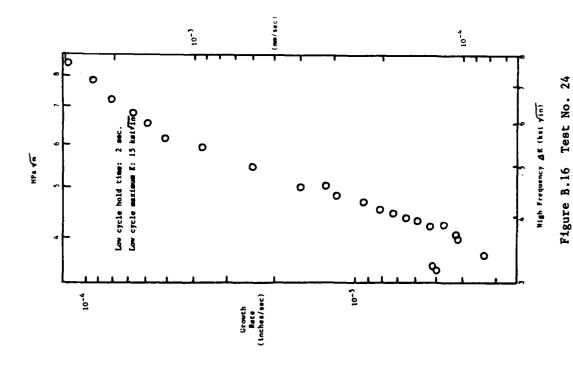


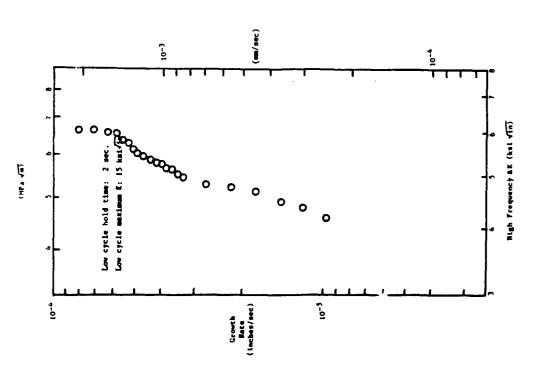
Figure B-10 Test No. 12











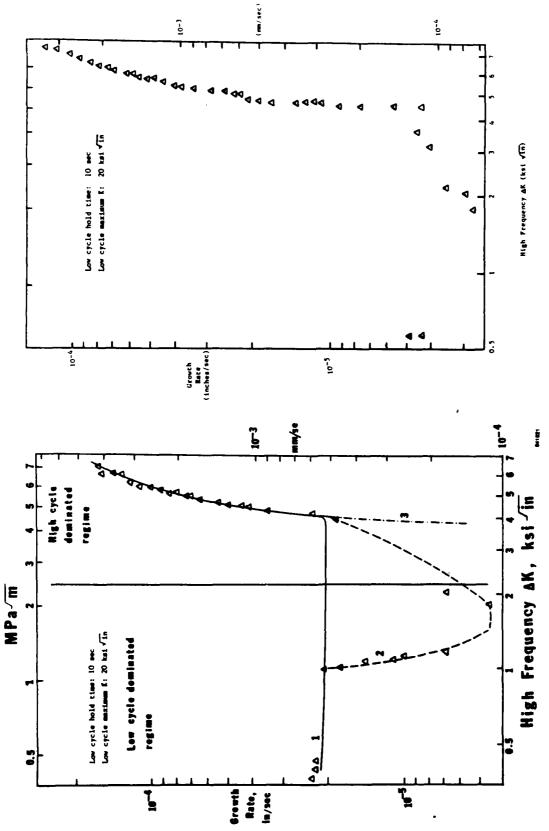


Figure B.18 Test NO. 27

Figure B.17 Test No. 26

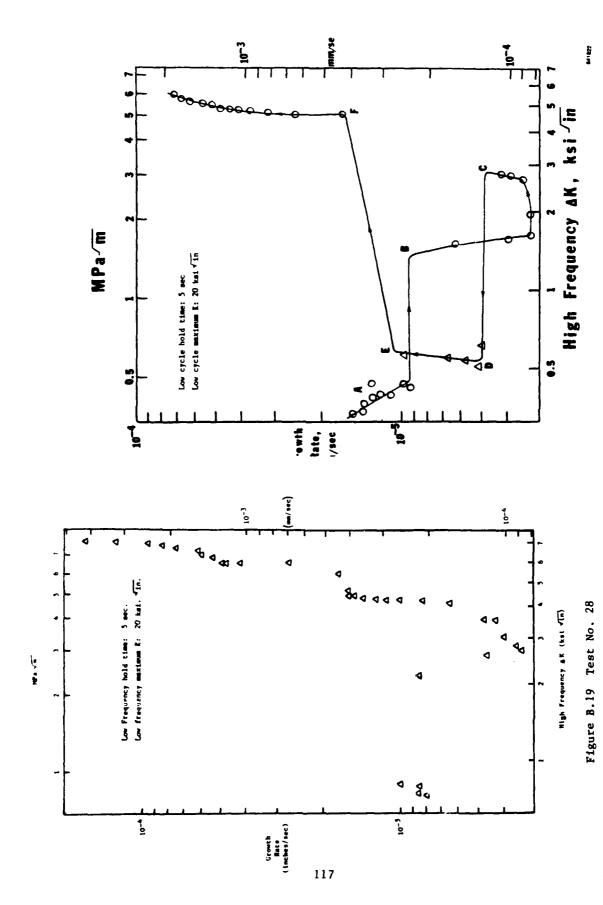


Figure B. 20 Test No. 30

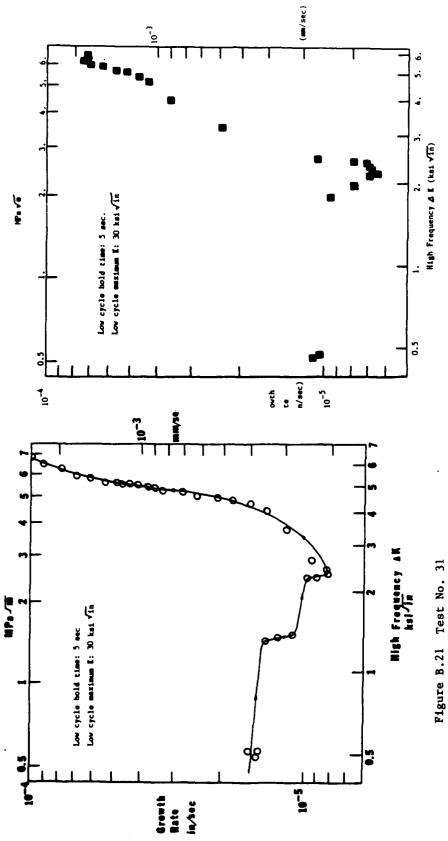
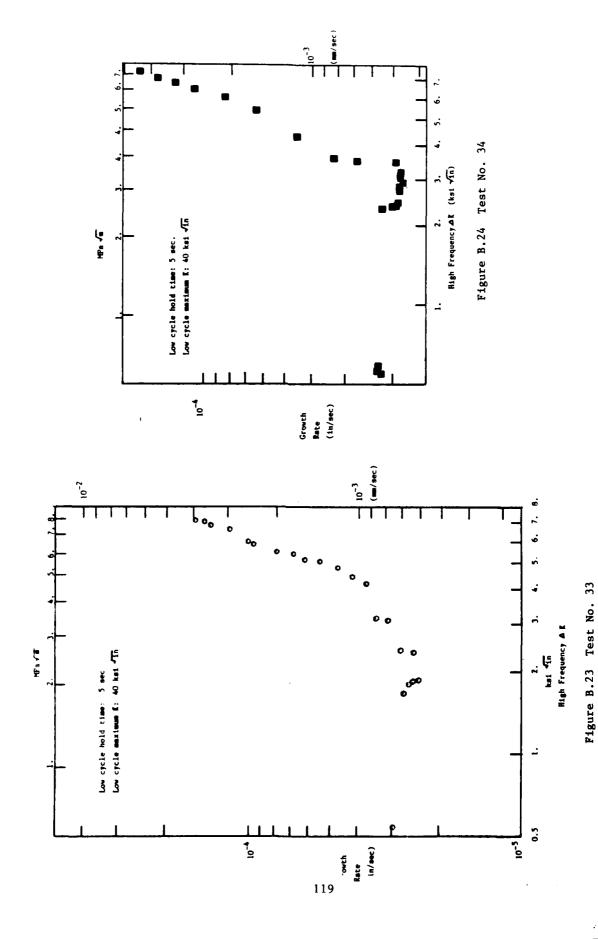


Figure B.22 Test No. 32



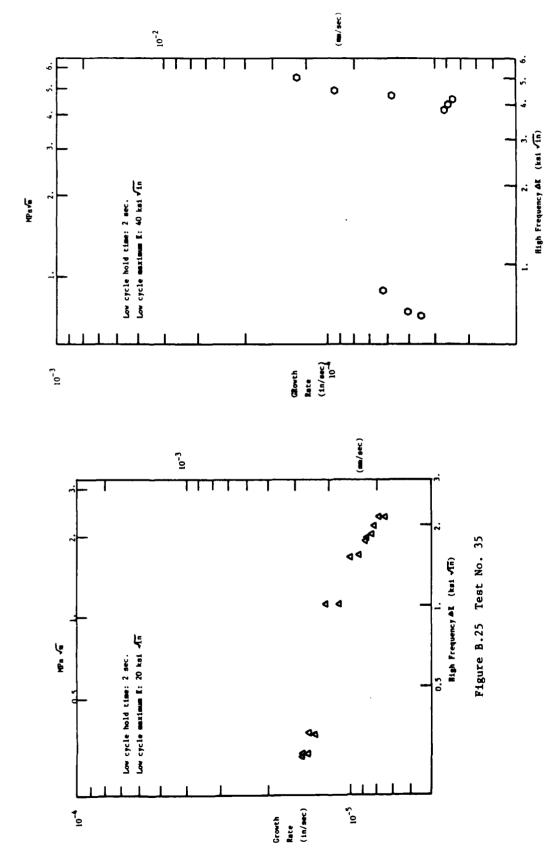
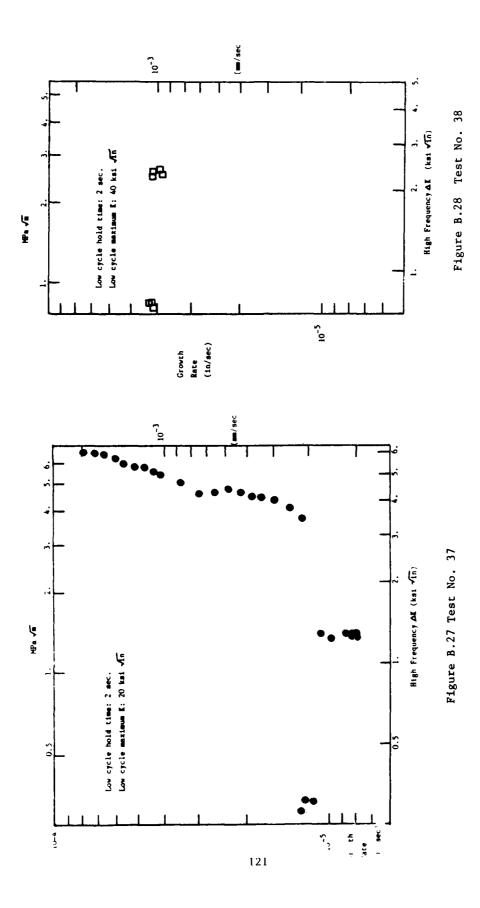


Figure B.26 Test No. 36



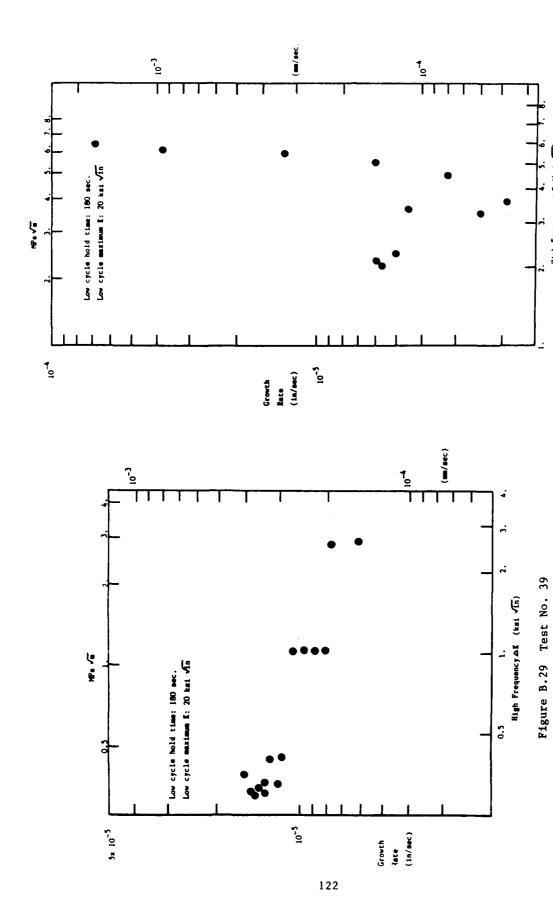
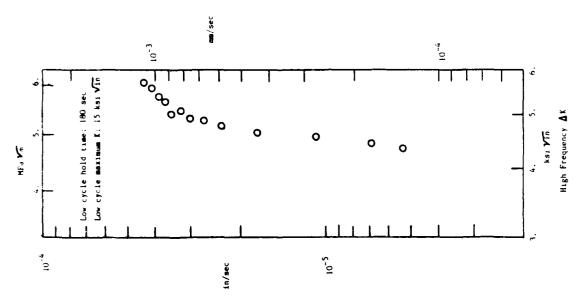
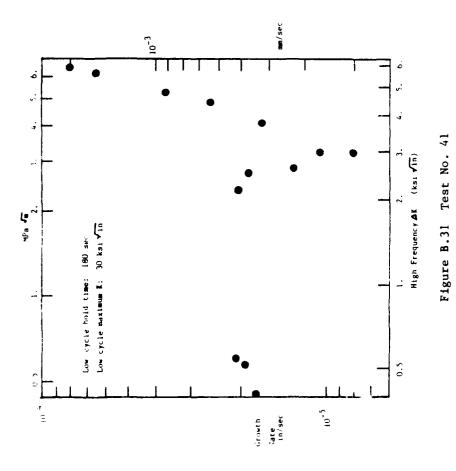
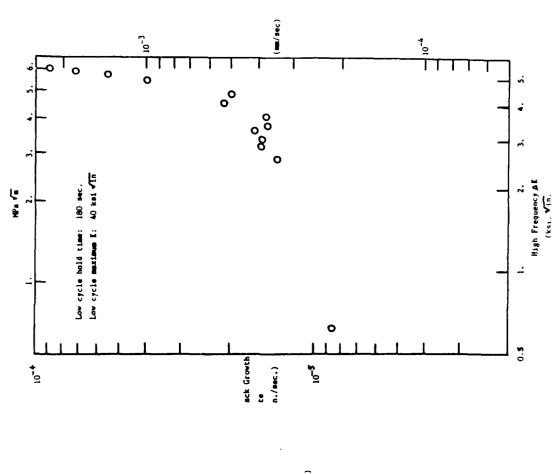


Figure B.30 Test No. 40







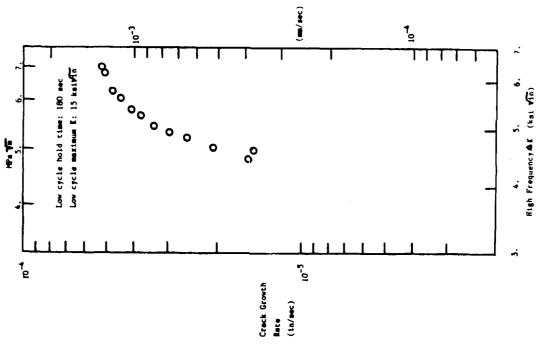
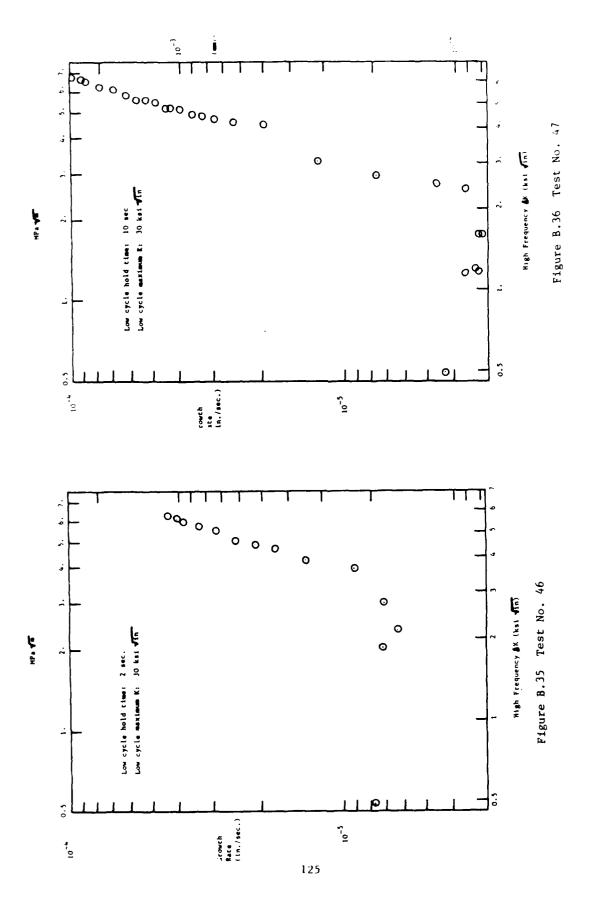


Figure B.34 Test No. 44



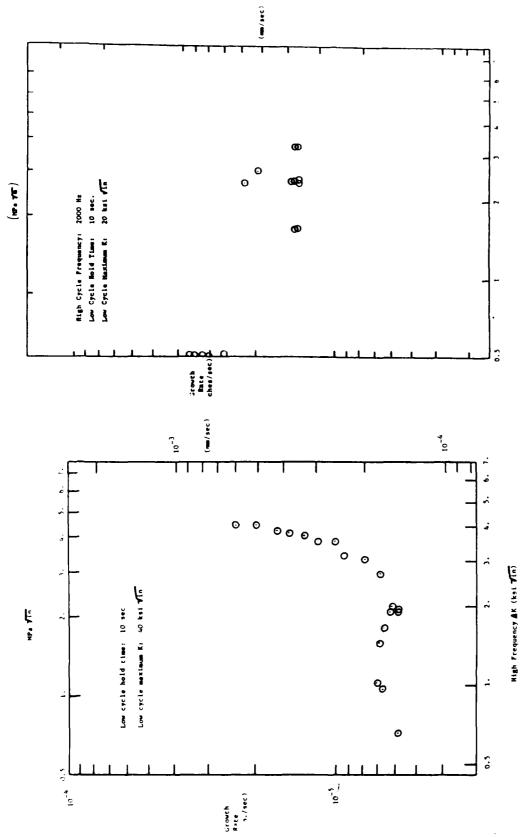


Figure B.38 Test No. 60

Righ Frequency &R (ksi fin)

Figure B.37 Test No. 48

126

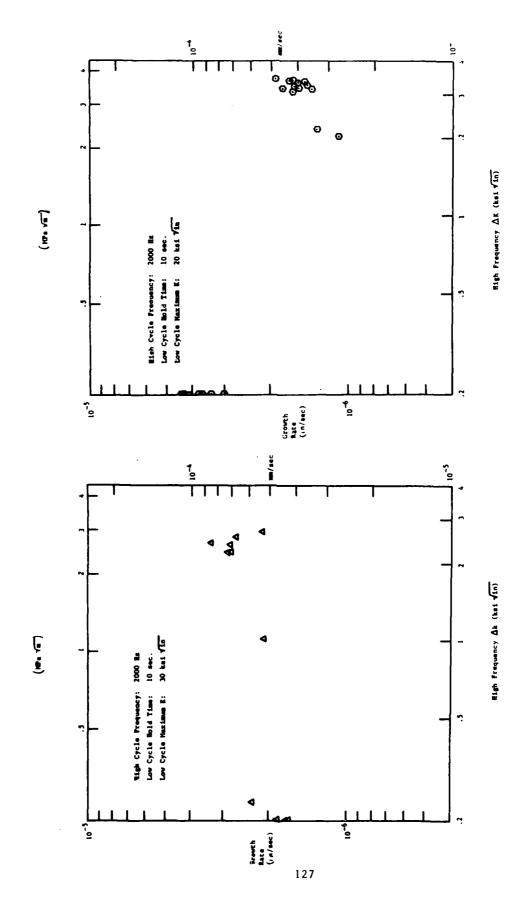


Figure B.40 Test No. 62

Figure B.39 Test No. 60

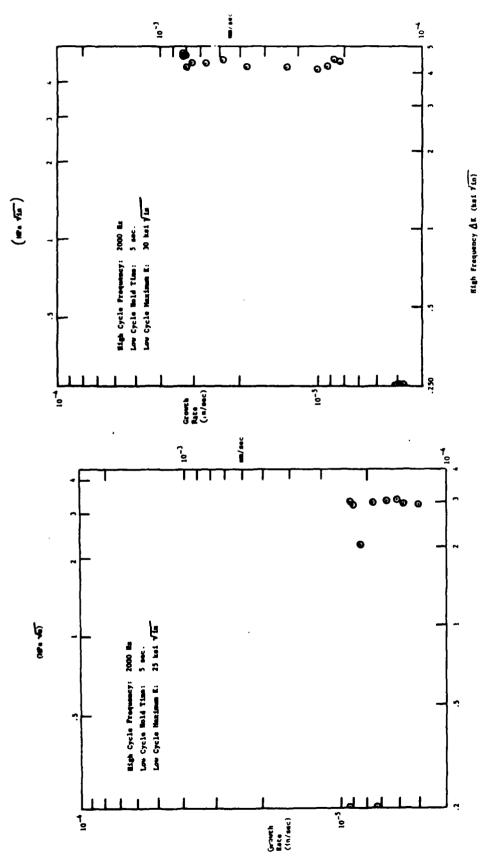


Figure B.42 Test No. 64

Righ Frequency AK (kei Vin)
Figure B.41 Test No. 63

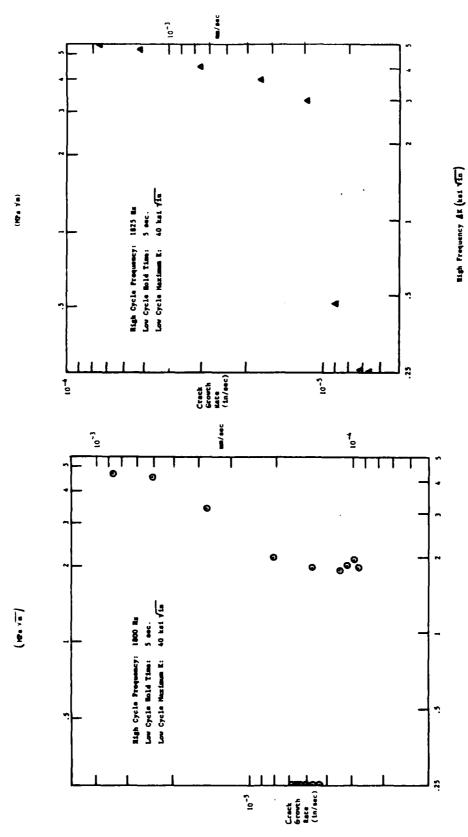
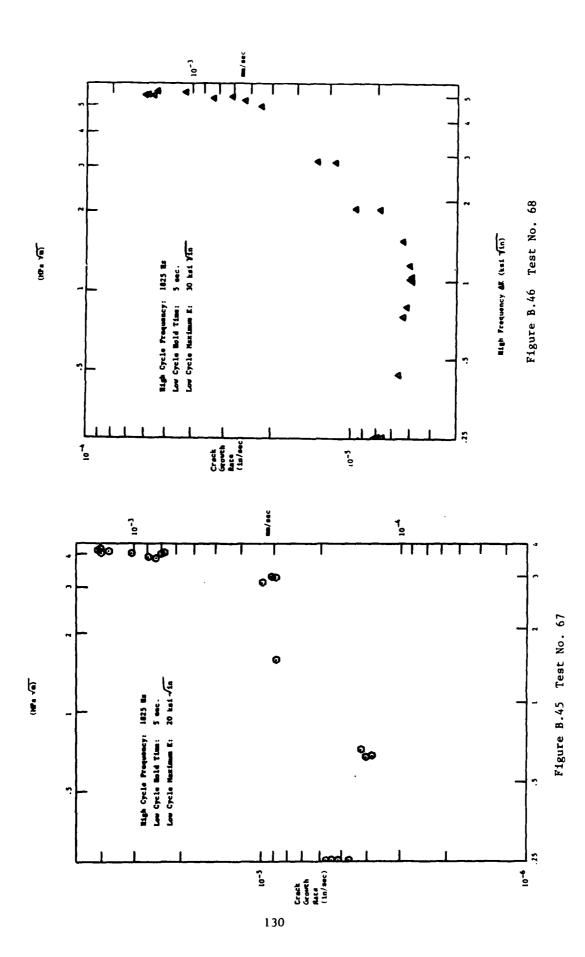


Figure B.44 Test No. 66

Figure B.43 Test No. 65

High Frequency OR (ket Vin)



#### APPENDIX C

DATA LISTINGS FOR ALL EXPERIMENTS

RECOLTS OF TEST NO. 1

Constant load test with a trapezoidal waveform having a  $60\,$  second hold time, no high cycle loading, R=0.1

Crack growth testing with low frequency loading only, 60 sec hold time, and  $R \approx 0.1$ RESULTS OF TEST NO. 3

TABLE C. 3

#### RESULTS OF TEST NO. 2 TABLE C. 2

Constant load test with a trapezoidal waveform having a 60 second hold time, no high cycle loading, R = 0.1

Growth Rate (in./sec)	0.00000123385 0.00000092738 0.0000007387 0.00000066074 0.0000006642
Low Freuquency AK (psi vin.)	16136.2617 16387.5547 16529.8945 16300.0469 16314.1562 16375.2969
Crack Length (in.)	0.1945 0.2000 0.2032 0.2080 0.2086 0.2097
Tine (sec)	2340.0000 7500.0000 11880.0000 19740.0000 20640.0000 22380.0000

Crack Growth Rate		0.2738E-05	0.3763E-05		0.4963E-05	5314E-0		71545-0		0.70105-05	73005	1070F	93755				0.21/35-04		0.2535.0	0.21002		2504F	2731E	0.28195-04	2519E	2615E	. 2388E	2070E	0.2090E-04	0.2228E-04	0.28545-04		0.4475E-04	0.5109E-04	0.5146E-04		0.5052E-04				7868E	. 1036E
AK (psi in. 1/2)	•	25046	25661	25979	26775	27134	27862	28819	30004	30259	30661	31397	31502	31679	32028	32776	32785	33399	33826	34380	35044	35457	36285	37326	38047	38192	38809	81 00 <del>7</del>	40865	41301	42618	43545	44753	46336	47656	48559	50398	51473	52877	53574	57730	28709
Crack Length (in.)	600	0.3302		•	•				0.4490		0.4611	•			0.4857	0.4986	0.4988	0.5092	0.5163		•			•	•	•	0.5910		0.6116				•	•	•						. 765	0.7827
Time (sec)	\$0691	1000	11000	10901	16606	62031	64671	67491	10491	71031	71991	73791	74031	74331	74751	75471	75471	75951	76191	76611	77091	77331	7.811	78351	78711	78771	79131	9896	17100	81137	(2110	11010	10010	76170	75470	62576	7/970	83091	63311	83421	63876	84081

RESULTS OF TEST NO. 4

Constant uncycled load, high cycle load of various frequencies

Tite (e)	Crack Length	Hax LF K	HF K Range	Growth Rate
, s	0.6205	27476 0000	( ) TO ( ) ( ) ( ) ( ) ( ) ( ) ( ) ( ) ( ) (	١,
3370.7698	0.6284	27637,0000	6563.0000	0.00011073000
5937	0.6369	26262,0000	7028.0000	0.00012606000
1919	0.6475	26560,0000	7021.0000	0.00014138001
7898	0.6594	26475,0000	6935.0000	0.00015327999
3879	0.6713	25879 0000	7056.0000	0.00016482000
7598	0.6845	24541.0000	7260.0000	0.00018151999
1318	0.6986	24092.0000	7490.0000	0.00020299001
7297	0.7144	23358.0000	7705.0000	0.0002772000
1018	0.7396	23643.0000	8251.0000	0.00022777000
24.78	0.7560	21938,0000	8400 0000	0.00021758000
1972.9419	0.7555	19520,0000		00005780000

RESULTS OF TEST NO. 7 TABLE (...

High Cycle Frequency: 246 Hz Low Cycle Hold Time: 60 sec Low Cycle Maximum K: 25 ksi vTn

Growth Rate (in. sec)

HF K Range (psi 'in.)

High Cycle Frequency: 248 Hz Low Cycle Hold Time: 10 sec

> Growth Rate (15.../840) 0.00011674033 0.0001185345 0.0001185345 0.000126535 0.0001267317 0.0001267300 0.0001267300 0.00001267300 0.00001267300 0.0000311699 0.00003027435

> MF Range (pai /fn.) (p

> Har LF (984 (48.)
> (984 (48.)
> 24.77. 23.7
> 24.77. 29.2
> 24.77. 29.2
> 24.77. 29.2
> 24.77. 29.3
> 24.77. 29.3
> 24.71. 85.4
> 24.11. 85.4
> 24.11. 85.4
> 24.15.2
> 24.90.773.4

0.4857 0.4927 0.5997 0.5069 0.5123 0.5235 0.5235 0.5580 0.5600 0.5600

Time	Crack Length	May 1.5 K	WF F Pence	A desired
(9ec)	(1n.)	(pet /In.)	(ps1 /In.)	
7984.6875	0.2884	21420.8984	787.9695	0.0000094181
9383.9336	0.3034	25230.0312	322.9622	0.00001158264
11084.0156	0.3252	25599.7070	226.9065	0.0000135139
12386 . 5664	0.3436	26294.6680	2122.8723	0.0000145559
13183.5195	0.3553	26434.4062	2349.8621	0.0000149763
14080.9570	0.3707	26770.3984	2298.0520	0.00001488024
14978.3984	0.3842	27066.2187	2610.2822	0.00001415124
15875.2070	0.3963	27487.7969	3088.2178	0.0000130747
16971.7266	0.4095	25899.5234	3429.1816	0.00001222489
18166.2070	0.4219	26296.1211	3541.4712	0.00001361158
19361.3203	0.4383	25894.9062	3646.9678	0.00001984589
20157.6406	0.4558	26672.8047	\$140.4961	0.00003097781
20555.8008	0.4675	27050.7266	5286.1211	0.00001640414
20754.8789	0.4750	27109.5391	5388.1836	0.00002467167
20854.1016	0.4815	26361.7422	2574.1467	0.00004010730
23646.2773	0.5050	25651.9141	5529.7539	0.00007742233
23745.5039	0.5127	25658.5820	5607.1758	0.00008925261
23844.0937	0.5218	25790.3672	5824.7070	0.00009586086
23943.3203	0.5312	25939.6523	5838.7500	0.00010083221
24042.5430	0.5420	25779.1484	5867.9297	0.00010941603
24141.1367	0.5532	25608.0781	6121.9414	0.30011525118
24240 .3594	0.5649	25686.8867	6164.8516	0.00012083116
24338.9492	0.5770	25452.4258	6397.6758	0.00012737671
24438.1758	0.5897	25350.6016	6505.2383	0.00013430661
24536.7656	0.6032	25332.5664	6529.3125	0.00014346938
24635.9922	0.6177	25388.5117	6964.9023	0.00015487349
24735.2148	0.6336	24974.6055	7135.6016	0.00016758432

\*Crack growth rate was not based on a sufficient range of crack length.

4117, 3789 428, 71328 428, 7330 4216, 7330 4216, 7330 421, 4453 4418, 5273 44

27021.7930 27786.2773 27709.9531 27659.7031

LABLE .

RESTANCE OF TEAT AND STANSON

Stab typle Fraguenc. 248 Hz U.s. ovole Hold Fime: 10 sec

	ı	<u>ئ</u> ے	_				_							_									_							7																		
Growth Rate	(10./mc)	9.00001193025	0.00001112623				•			•		0.00000333309	٠-,	٠.	0.00000613033					0.000005/2822			.0000053134		.0000030764	0.000005221/2		0.00000462104	٠.		0.00000429527	0 00000462211	٠.	.00000553	0.00000637227			.0000105353	.0000122636	0000133311	0.00001536643							0.00007786212
HE R Bang.	4. Teg	379 6965	7		273. 3506						-	1881.1353							7007 4387						2225.3672	2292 0698		2262.9773		2334.8945	122 1239				3360.5060			4206.9492	4239 .4687		0007.71	4457 1260	8680	2417	5538.9648	5578.3000		5659 2812
Has Of F	(per , in.	23661, 3945	234.75.0703	23330 406	.3165.2422			~ .			27.65					23600.2578	234 79 3433	23261.8047	0755.7.007	23786,5625	23.39.9102	23790.7578		23290 . 6023	23320 2622		23436.2812	23285.8359	23195.1484	23165.9844	23679.1172	23527.7852	23583.3750	23617.0547	23/63.23/6	23887.7461	23525.2734				22761.8933	23994, 3084						23056.8945
G-ack Length	9	0.2170				•			•	( T )	7.67				٠,			0.2757						0.2993						0.3261				0.3456	3560					0.3732						•	0.4151	0.6223
4	9	1197 6549			٠.			200.500				5692.3750					2000 Jaks				11379.1606	11977.9570	12773.5506	14.971 4.94	15069 1406	15866. 2812				19236-8281				27676 9653			25830. 312	26328.0469	20720.2656	27025.0LS		27722.1719	27821 6523	27920.4961		28120.3320		28319-444/

TABLE C.8 (Cont '3)

inge Growth Race in.) (in., sec.	.0703 0.00009162258 .1680 0.00009264356		0 0	8203 0 00004369611	0	4727 0.00005029375	6	0	7656 0.0000587: 24	33 0.00006168520	0	183 0.00006700812	9570 0.00007149023		_		1328 0.00008870705	6016 0 00009783010	7344 0.00010989855	Ö	0	
(pet 'in.)	5789.03 6553.16		4939.62			292			5917.76	5918.61	6027.8047	2109	6212.95	6386. 42	6381.5703	6702.48	6806.12	7022 60	6917.73		7091.49	
Nex (F. R.)	22791.0039 24008.7461 23373.3711	25795.6094	24674.6523	23826.7578	23596.6914	23527.7773	23439 1445	23290.9766	23541, 7187	23592.2891	23860 . 6094	23610.4687	23640.3672	23428 2345	23674. 5504	21299.6172	21322.9687	21509.1016	21455 3945		21507.5234	
Crack Length (in.)	0.4385	***************************************	0.4876	0.4955	0.5001	0.5049	0.5156	0.5213	0.5271	0.9384	0.5450	0.5513	0.5653	0.5726	0.5797	0.5872	0.5953	0.6044	0.6146	0.6257	0.6378	0.6269
Time (sec)	28517.8008 28617.1484 28716.4336	33973.8359	34073,0000	34271.2305	36370.4023	34566.6797	34667.8125	34766.9062	14965. 9687 14965. 1984	35064. 4648	35163.5977	35262.8203 35361.4766	35460.8867	35523.9844	35659.0620	35756.3086	35857 4687	35957.0078				36353,3359

\*A long dwell in the low cycle dominated regime for constant low-frequency  $\Delta K$  shows a decreasing crack growth rate.

TABLE (.9 (Cont'd)

	(in.) (psi /in.) (psi /in.) (in. sec)		0.4812 26322.9024 6742 2644 0.00015355332	24210.8125 7052.8281 0	1312.5078	24,301,9297 7640,3594 0	24,235, 3086 7944, 04,30	0.3739			C	IABLE C. 10	11 OF E334 30 31 11380	ACCOUNTS OF 1EST MO. 11	High Cycle Frequency: 605 Hz				(pet /In.) (pet /In.)		22738.3437 227.7065	22518.0781 422.0615	232.8022	22305.4687 255.0401	22328.7305 317.1313	22395.8477 320.8130	22224.3359 563.5928	261.5728	2505. 407 2777	14007 (42 24) 6978	21764 8300	22708.9492 606.9131	22872.7227 2311.0425	22761.3555 4308.6016	4317,6719	0.2669 22801.5547 4335.1328 0.00000545417	23005.9609 4456.	23114.5898 4519.4453	22878.8086 5679.0625	22419.9844 5883.8516		22294.5703 6259	2488,4687	22473.9727 5951	0.3552 22096.6094 5949.0352 0.00011367828	6290 7930	21816.7305 6472 ROOM				
	(308)	771. 01.694	60329 . 4570	604.29 . 3123	60528.5352	1197 1297	60/28.9844 60824.3870	0.07 : 97606										Time C	(8ec)		2290.0078	2987 . 1924	3/86 / 983 546 6117	/11C.000#	8/00.0000 8/47 FEES	6166 2014	1403.02/ 1403.1404	7587:5867	AGA BATEL	22164 4211	24.16. 1797	24261.8750	31947.7891	37426.5352	38916.7500	40112.5977	41206.7148	41505.1797	41902.9023	42002.2461	42101.7461	42201.1680	42300.5117	42399.8516	42499.7891	42599.0117	42698.1562				
248 Hz	oes n	OF E Lange Growth Bate		1580.7959 0.00000198970	Ö	Ö	o 0	24.2E 84.24 0 00000333861		<i>;</i> e				2991.0759 0.00000177671	71188100000 0 8178 8184				<i>•</i>	3010.6314 0.00000201505				0	•	6121.0742 0.00000734338	421.818* 0.0000/0/35* 4134.0199 0.00000661424	, 6		0 984	747	6122.4023 U.DGOGGGGA94 6281 1441 O.DGOGGTA893	4242.8516 3.00000795503	0	6097	6875 0.	4519.0898 0.00000/6/2/3	, ,	0	Ö	۰		0	-	0 0 0 0	2875	692	2222.0347 0.30002009184 2344 4043 0.30003383348	4602	0937	
High Cycle Frequency:	ie noto lime:	# 6 F 15 F 15 F 15 F 15 F 15 F 15 F 15 F 15		22325.0664 1580									22619 . 8437 2926	<b>1</b> 7				•	23159.9531 3509				23655.9570 3922			505	1281	000				23835,9805 6261			4				23536.5234 4694										3 3	7266	
High (		Crack Langth			0.2116								141 0.2497			6.274							1789 0.2884		0.2949							227 0.3390		1125 0.3501																0.4549	
		# (F		10746.1172	11863.9	3660.6	1327. 7422	17364.3	18750 1	20349	21946 1992	23343, 550	26640 6161	26130.2	775.45	28931.0	30327.7852	30627.3516	32223. 1523		39807.1	4.0403	40504.8789	41702.5	42500	2196.	4495, 3281	45093.8320		46589. L	47307.3	4.965.0033 4.0684.2227	49482.4	50000.3125	50578.9	51276.0547	52670 2500	53267.4	53864, 7305	5-561.8	55357.3	56053.3	56650.5	57,47.9687	1	38261.3	58738.7	19733	60031.7	60131.0078	

TABLE C. 11

RESULTS OF TEST NO. 12

High Cycle Frequency: 605 Hz Low Cycle Hold Time: 10 sec

TABLE C.12

RESULTS OF TEST NO. 13

Constant load, no cycling

GROWTH RATE	(inches/sec)	0.00000411751	0.00000573215	0.00000175572	0.00000696139	0.00000961069	0.00001040402	0.00001121570	0.00001191794	0.00001290539	0.00001347179	0.00001386898	0.00001417301	0.00001426718	0.00001525549	0.00001673370	0.00001811901	0.00001942298	0.00001582147	0.00001737023	0.00001682866	0.00001995124	0.00002138640	0.00002247342	0.00002456056	0.00002376697	0.00002576357	0.00002674197	0.00005098156	0.00010911345	0.00018643984	0.00018205498	0.00016690358							
HF K RANGE	(pet in.)																																							
MAX LF K	(ps 1 ta)	23961,6406	24174.6758	26662.5312	24753.0352	25381 7070	25693.0195	25964, 2461	26117.9219	26356.3125	26861.6758	27032.3320	27329.1953	27339.4492	27724, 2539	27902.6992	28219.9883	28531.8242	28767.6680	28978.1758	29255.1758	29572.7305	30180.0547	30800, 1250	31212.8242	31410.1250	32171.3984	32966.9180	32808.2695	34921.5625	40816.2148	39784.8711	47027.1250							
CRACK LENGTH	(Inches)	0.2869	0.2915	0.2965	0.2996	0.3037	0.3084	0.3127	0.3172	0.3225	0.3285	0.3347	0.3405	0.3462	0.3520	0.3585	0.3660	0.3699	0.3747	0.3770	0.3802	0.3887	0.3986	0.4080	0.4187	0.4274	0.4374	0.4503	0.4476	0.4817	0.5658	0.6039	0.6511							
TIRE	(Sec)	11620.3008	12676.3984	13522.0000	13944.3984	14366.8008	14789.1992	15212.3008	15634.6992	16057.1992	16479.6016	16902.1016	17325.1016	17747.6016	18170.0000	18592.5000	19014.8964	19226.1992	19437.3984	19649.3008	19860.5000	20282.8984	20705.3008	21127.6992	21550.1992	21973.3008	22395.6992	22818.1016	23240.6016	23874.8984	24508.3984	24720.0000	24931.2500							
GROWTH RATE	(inches/sec)	821819	564137	543995	525306	724804	154467	780310	706054	174450	80808	130064	530349	520370	157654	306677	11314	364495	909526	968979	380472	817993	500642	504.503	235798	189578	254430	132758	16091	569969	155028	559808	142890	562846	017189	00005480160	165937	712927	316225	187476
CHOM	(Inch	0.00003821819	0.00003664137	0.00003643995	0.00003625306	0.00003724804	0.00003754467	0.00003780310	0.00003706054	0.00003474450	0.00003480908	0.00003430064	0.00003630349	0.00003620370	0.00003557654	0.00003306677	0.00003011314	0.00002864495	0.00002909526	0.00003068979	0.00004080472	0.00003817993	0.00003600642	0.00006604503	0.00008235798	0.00008789578	0.00008254430	0.00007832758	0.00005444091	0.00002569969	0.00003155028	0.00003659808	0.00004142890	0.00004562846	0.00005017189	0.00005	0.00006465937	0.00007712927	0.00009316225	0.00011187476
HF K RANGE	(ps1 in.)	443, 1199	1004 2280	1010.5854	1130.5286	1440.7424	1489.1851	1538.5725	1666.4360	1677.3757	1766.1792	2013.2786	1629.3350	2441.6672	2701.9609	2517.5942	2572.6995	2668.1091	2598.4189	2654.8840	2667.4263	2737.8218	3483.4497	3488.5422	3582.3401	3528.1506	3697.1140	3926.9143	3931.1694	4060.8308	4178.9961	4020.960	\$725.7656	5638.9453	5647.0898	5803.3164	5875.2578	6063.6680	6125.2266	6480.4766
HAX LF K	(psi ta)	41686.8047	41386.2500	41204.4102	41213.6523	41035 8164	40854.7461	40848.4180	40331.3750	40315 2617	40063.4336	39997.0625	39881.4805	39860.0508	39785.5820	39656.7812	39510.9844	39449.9141	39117.0273	39105.0781	38689,6602	38890.0586	36960.1797	36927 . 1016	38685.2578	38605.7812	35115.4180	39345.3672	39683.5195	39081.9141	39681.0586	39650.9883	38875.9727	38664.5234	38612.3750	38171.4414	37606.6875	37043.4531	36564.9180	35917.7031
CHACK LENGTH	(Inches)	6775 0	5875 0	0.5527	0.5550	0 5593	0 5632	0 5671	0.5712	0 5747	0.5780	0.5810	0.5846	0.5884	0.5924	0.5960	0.5989	0.6016	0.6041	0.6071	9609.0	0.6150	0.6193	0.6201	0.6274	0.6382	0.6491	0.6609	0.6670	0.6658	0.6686	0.6721	0.6761	0.6807	0.6854	0.6900	0.6955	0.7021	0.7102	0.7202
1116	(Sec)	499.2798	599, 7678	699 6240	799 4800	899, 3359	999.1919	1099,6799	1198.9041	1298 7600	1398.6160	1498.4719	1598.3279	1697.5520	1796.7759	1897.2639	1996.4880	2096.3440	2196.2000	2296.0559	2395.2800	2495.1360	2594.3601	2694.2161	2793.4399	2893.2959	2992.5200	3091.7441	3191.6001	3291.4561	3390.6799	3490.5361	3589.7600	3688.9841	3788.2C )	3688.0640	396 . 2881	4085.8801	4185.7344	4284.9570

ABLE C. 13		
	<u>=</u> ;	
	ر د	
Ε.	Y	

RESULTS OF TEST WO. 20
High Cycle Frequency: 200 Hz
Low Cycle Hold Time: 10 sec

	CRACK LEMBTH	<u>ا</u> ا	¥ 8	GROWTH RATE			TABLE C. 14		
	INCHES 240143	P8141N 19275.6	7164	1,46917E-05		2	RESULTS OF TEST NO.	21	
	242142	15372.4	5151.03 5458.03	1.56078E-05		1	High Curle Specification	200 Hz	
	. 25302	14408.5	5419.15	1.45979E-05		Low C	Low Cycle Hold Time:	5 BEC	
	.259096	14909.1	10-70-00	2.05388E-05					
	270317	14502.7	5676.84	2.246@1E-05	TIME	CRACK LENGTH	ا اد	¥ ;	GROWTH FATE
	275052	14225.7	5474.89	2.3/380E-03	SEC 10150.7	INCHES 242241	PSI 41 N	A PACE	INCHES SE
	.280027		2011.23	2.78892E-05	11727.7	247325	14774.2	1000.00	3-1568
	.285311	1.0840.0	6069.58	2.968785-05	13303.9	.252022	14896.3	3879.15	3.1025.6-05
	267132		4.9809	3,211426-05	14755	.257116	14774.2	3930,39	3.445875-05
	304989		6125.34	2.964646-05	17215	. 265905	14802.6	3957.07	3.60E95E-05
	31075	13364.6	6059.57	2.81.50.00 - 0.00 - 300.000	18793.4	.271698	14816.6	3997.14	3.349.4E-00
21330.2	.310091	13023.1	6291.51	100 PE 00 PE	19612.9	.274558	14787.6	3975.68	3.22298E-04
	.316305	15120.1	4/84.34	0.000000000000000000000000000000000000	00017	64/8/2	14/04.5	40.00	3.75178E-06
	.317878	14901.0	6441.12	2.64979E-05	22001	266074	14743.0	4.26.41	4.311146-04
m (	9752910	24747	5979.44	2.97442E-05	24215.1	294100	14482.3	4133.04	8.82904F-06
	100711	15114	6260.5	3.17100E-05	25401.9	308728	14602.3	4259.48	1.05878E-05
	341087	14966.4	2609	0.000000000000000000000000000000000000	25980.1	.313666	14550.1	4346.51	9.32345E-00
•	340021	15152.5	6091.25	いつしいいいのかっち	26547.1	.319991	14678.1	4421.17	8.37080E-04
_	. 355339	14802.2	6628.76	3.80836870	27115.1	. 32363	14535.3	4459.72	7.07991E-03
24903.7	339099	14999.3	6574.97	5.48238E-00	27809.1	.326412	14318.1	4497.3	5.16632E-06
25101.9	.367231	14788.5	6522.24	4.48740F-05	29638.1	.335239	14390.1	4538.23	6.90702E-06
	371399	14591	AB4A.77	4.4544E-05	30457.1	.340864	13990.1	4559.37	8.345196-06
	19046	14436.4	6414.21	4.68673E-05	31088,1	946346	14259.8	4506.71	9.30251E-06
	72.00	14428	6831.84	5,13000E-05	32476.1	.362514	14242	4633.4	1.182728-05
	394021	14343.9	7168.61	5.45249E-115	33106.1	. 369923	14155.9	4712.29	1.34232E-05
	. 400344	14523.1	7017.95	5.797/1E-05	1.4/055	719//5:	9.0004	4/0/.00	10-11/4000 · 1
	. 406649	14283	6473.07	0.010 A 010	34242.1	- 50004	13/27.4	4774.05	1.44/40E-00
	.412764	4 · 9886 I	7001/	4.22454F-05	101/10	47404	1.1001	4002.40	1.707716-05
	418639	15767	/0:017/	0 - 900000 m	110000	*****	10401	10 H 10 H	30 30//0/1
•	.423618	13710.0	7151.96	4.96666E-05	34250.1	4189	130001	50101 6084.63	2.0742AE-05
•	41644	13560.6	7006.57	3.970356-05	36638.1	.427724	13631.4	5056.48	2,20227E-05
	434985		5220.6	2.77574E-05	37016.1	- 4363B7	13658.4	7568.14	2.43B01E-05
	9444	15247	2292.08	1.697916-05	37269.1	.44267	13306.1	5233.69	2,450498-05
_	. 448975	15364.4	5333.53	1.44/756105	37647.1	.452515	13600.6	5334.54	٠
27481.3	. 453807	15499.2	20.6/20	8.04164E-05	37900.1	.459012	13475.8	5371.77	2.532116 -05
•	48286	15012.0		A0-1040404.4	38278.1	. 468341	13297.6	5485.23	2,52942E-05
28073.7	45764	15273.3		4.24045E-06					•
_	. 465471	15151	4774.05	2.94880E-04					
_	. 46674	13227	4717.79	2.87439E-04					
	1842/4	1.10261	4980.29	2.99632E-06					
33616.0	470m64	13400.4	4664.43	2.73551E-06					
	484784		4739.94	2.38845E-06					
	468424	15314.4	4691.38	2.00577E-06					
	49173		4482.22	1.90823E-06					
87	.497001	15480.7	4426.87	1.63363E-06					

TABLE

. . .

RESULTS OF THE 23

RESULTS OF TEST NO. 24 TABLE C. 16

High Cycle Prequency: 200 Hz Lov Sycle Hold Time. 5 sec

								1	
	#	High Cycle Frequency:	: 200 Hz			H1gh	High Cycle Frequency:	200 H2	
	01	Low Sycle Hold Time.	\$ <b>8e</b> c			Low	Low Cycle Hold Time:		
					TIME	CRACK LENGTH	ŗ. X	*	SROWTH BATE
INE	CRACK LENGTY	ارة ا	¥	GROWTH RATE	SECONDS	INCHES	PSI AN	NIT-184	INCHES/SEC
		Z 17 1 18 4	X 10 10 10 10 10 10 10 10 10 10 10 10 10	1MCHES/3EC	4606.84	. 223925	15594.7	3179.65	5.05628E-06
137/3.0	24443	130/0.4	3417.12	7.71230E-108	5609.16	. 229521	15641.2	3244.61	5.20793E-06
7.0070	174407	× 1001	3446.74	20.000 00.00 40.000 00.00	6411.5	.234768	15729.5	3296.21	5.201356-06
2000	167107.	0.707	1444	4.04708E-05	7414.48	. 239853	15763	3413.37	4.89075E-06
17405.2	400000 400000	P 4000	2000:04 40.000	4.047085.04	1919	. 245684	15057.4	3682.07	4.14207E-06
710070./	1002/5		17.03.6	A. CBOASE-0A	11021.5	. 253005	14865.3	3409.08	3.34664E-06
1744.0	787773	14063.7	27.57.5	4.0070AF-0A	13029.8	. 258689	14948.7	3418.56	3.28950E -06
0.0484	707107	0 11 17	CE 1761	4 420245-04	14836.9	.26414	14853.5	3408.92	3.693398-06
24400.7	70474	4204	100.00	• •	16241.8	. 269325	14917	3761.31	4.24272E-06
0 00150	0 + 0 O O E	0011	1000	4-30215F-04	17445.	.27462	14676.8	3868.84	4.72856E-06
7.00.0	\$ 10 SC	44204	90.00	A. DADITAL OF	18649.4	. 280763	14458.4	3850.18	5.27330E-06
7.07.0	76.00	*****	CF 1/0F	A - 4 - 4 - 6 - 6 - 6 - 6 - 6 - 6 - 6 - 6	19651.8	. 286362	14493.5	3939.07	5.59561E-06
	0074.5	0.0111	10.400	10001C 4	20654.4	. 292063	14514.9	3946.46	5.96646E-06
5.007		0000	3743.01	10-11-10-10-1-1-1-1-1-1-1-1-1-1-1-1-1-1	21457.1	.297022	14548	3939	6.2190BF-06
1 1 1 1 1 1 1	19775	D. 75761	Þ٢	00-30-30-40-40-40-40-40-40-40-40-40-40-40-40-40	22460.4	. 303414	14338.8	3982.93	6.53483F-06
7.5/045	77077	6.49161	4277.47	7.28230E-08	23263.1	.308732	14445.2	4012.40	40-36464
35117.8	. 339332	4431.0	4367.34	1.31938E-03	24265.7	313691	14107	4077 22	PO-MARCE C
35463	143863	12006.9	- 1	1.47059E-05	25068.4	321781	14404.5	4089 20	7 204115-00
35809	17646	14829.6	4475.29	1.59329E-05	25869.9	327979	14024.1	4144 87	00-3111000
36105.8	400400	14789.2	4524.63	1./3392E-05	26672.2	31482	14311.8	¥170.5	400000
36401.9	. 339438	14828	4612.75	* . 85056E-05	27474.8	741017		4764	0.5880BE-00
36698.3	19029	14666.2	4390.06	2.04603E-05	28077.4	347.77	1 4 2 : 4	7504.50	00-30/004-6
36945.1	.370154	14534	4724.36	2.2471JE-05	70400	00 P P	0.7171	00.4004	1.051025-05
37192.4	.375874	14808.5	4853.29	2.561966-05	00000	*****	14234.4	00.4/54	1.188738-05
37389.8	. 381114	14747.4	5033.01	2.63292E-05	1.10000	0 1 0 0 1 1	0.00141	66.1944	1.20302E-05
37587.8	.386873	14460.6	5085.29	3.0808点€-05	0.70000	NOTE 07 .	14101.1	42/6.48	1.28988E-05
37735.9	. 391744	14622.7	5165.63	3.30694E 05	4.4000F	448704	13986.7	4556.17	1.413BOE-05
37933.3	398509	14419.3	5286.9	3.57565E-05	0.0000	70/4/5	14056.1	4240.04	1.61585E -05
38081.9	. 403913	14522.2	5295.75	3.89698E-05	1.78805	4D4400.	15062.1	4966.88	2.44796E-05
38229.8	. 409604	14273.1	5422.33	4.35185E-05	2.505.1	946.6	14908.8	5371.99	3.71809E-05
38378.1	. 416234	24434.9	5555.52	4.914778-05	0.1488.0	.401813	14949.7	2480.04	4.500000E-00
38476.9	.421292	14479.5	5785.21	5.264 '0E-05	31088.4	.411/62	15071.2	5628.37	5.08689E-05
.375.9	. 426942	14626.2	5873.24	5.41296E-05	31864.5	. 423788	14960.5	5782.32	5.64521E-05
38674.6	43279	14825.5	. г	J. 1338481-03	32089.8	. 435124	15170	5918.37	5.98691E-05
38773.4	438208	14910.3	4214.54	5.472956-05	32290.1	V 1447417	15193.8	6058.01	6.33767E-05
	*****			A 401475-04	32490.8	.44023	15178.9	6250.05	6.779565-05
30070.1	45144	13.47	4.046	A - A - A - A - A - A - A - A - A - A -	32692	.474052	15336.2	6413.88	7.36265E-05
4 44 4	A 100			1000 H	32892.4	.489203	15536.5	6621.06	8.156555-05
	10000	11011		10 101101	13093	. 506464	15222.5	6880.98	8.834045-05
37.60/ . 3	C47+0+.	14/2/11	16.0990	3.80334E-03	33293.7	. 524824	15302.8	7195.33	00 100 00 00 00 00 00 00 00 00 00 00 00
					33494.6	.544718	15342.5	7489.8	1.051316-04
					33695.5	.566616	15340.4	7823.79	1.140045-04
		•							10001111

TABLE C. 18

RESULTS OF TEST NO.

26 200 Hz 10 Bec Bigh Cycle Frequency: Low Cycle Hold Time:

GROWTH RATE INCHES/SEC	2.60107E-05	2.43103E-05	2.25147E-05	2.27702E-05	2.29000E-05	2.21918E-05	2.22762E-05	2.09159E-05	1.94536E-05	1.75716E-05	1.56418E-05	1.39363E-05	1.21865E-05	1.07715E-05	9.79167E-06	9.585915-06	9.61611E-06	6.58685E-06	4.52905E-06	6.73840E-06	1.18570E-05	2.27630E-05	3.46956E-05	4.09139E-05	4.38715E-05	4.96152E-05	5.34657E-05	6.34540E-05	6.76531E-05	7.17218E-05	7.67511E-05	8.33411E-05	9.00567E-05	1,00405E-04	1.10882E-04	1.21563E-04	1,325356-04	1.42347E-04	1.56928E-04	, 266E-04
¥ 8	368.538	370.757	400.22	416.665	427.059	444.469	435.19	1025.44	1045,55	1054.79	1078.61	1094.12	1103.8	1120.6	1127.71	1167.49	1192.79	1180.33	1828.52	2052.56	4043.77	4287.65	4375.27	4487.82	4530.77	4660.48	4706.52	4831.52	4919.57	4990.77	5085.51	5152	5263.77	5364.22	5494.63	5614.01	5767.28	5937.38	6137	6339.12
A PER SE	20646.3	20730.1	20749.1	20766.5	20909.5	20868.7	20951.2	21051	21135.3	21263	21337	21538.5	21475.1	21536.2	21685.7	21350.2	21726	21350.7	20889.9	20592.2	20497.5	20474.3	20348	19998.9	20268.7	20406.3	20460.6	20498.3	20644.6	20570.2	20661.2	20651.6	20474.1	20701.8	20440.6	20543.7	20857.3	20941.3	21080.6	21326.4
CRACK LENGTH INCHES	. 265289	.272813	.286124	.292666	. 29948	.306379	.311033	.317605	.323834	.32927	.335894	.341657	.34661	.352082	.357041	.364605	.37402	.386309	.414256	.419472	.429797	. 43503	.438857	.446038	. 45067	.460062	.464475	.475906	.482789	.48982	.497457	. 504959	.513324	.522867	. 533467	. 545155	. 557852	.571526	.586322	. 603259
TINE	1387.89	1604.03	2279.08	2575.89	2872.87	3170.42	3368.47	3669.29	3970.15	4271.24	4672.2	5073.03	5474.5	5975.8	6578.16	7280.9	8083.49	9889.53	21026.1	22129.9	23334	23735.3	23936	24137.2	24237.6	24438.4	24538.8	24739.5	24839.9	24940.1	25040.5	25141.4	25241.7	25342.2	25442.5	25542.8	25643.2	25743.6	25844	25944.3
						GROWTH RATE	INCHES/SEC	9.6631BE-06	1.19874E-05	1,44594E-05	1.79219E-05	2.20731E-05	2.72575E-05	3.33616E-05	3.47997E-05	3.66119E-05	3.82572E-05	3.94627E-05	4.14606E-05	4.38213E-05	4.62507E-05	4.84076E-05	5.04204E-05	5.26406E-05	5.52307E-05	5.83182E-05	6.31815E-05	7.12344E-05	3.51581E-05	1.03244E-04										
	. 25	900				į ¥	PgI LIN	4183.33	4356.9	4479.3	4703.94	4754.94	4824	4959.78	5040.66	5122.27	5149.12	5224.06	5293.93	5364.33	5436.1	5499.68	5612.34	5716.72	5820.48	5891.86	5977.56	6406.62	6742.58	7157.55										
TABLE C. 17	RESULTS OF TEST NO.		nign cycle frequency: Low Cycle Bold Time:			ינ ני	PBI LEN	15232.4	14956.2	15104	15076	14992	14754.2	14656.9	14789.1	14923	14611.7	14588.6	14841.9	14564.1	14551.3	14290.3	14338.3	14254	14145	13969.1	14082.9	14282.7	13962.8	14004.1										
	2	1 4	10 4 C			LENGTH		<b>.</b>		.327102	.331765	. 337393	.344727	. 354088	. 359804	.366015	.371875	. 377934	.384462	.391241	.398564	. 406035	. 413939	. 422356	. 430881	. 4397	. 449067	. 458976	.470238	. 484288										
						13E	SECONDS	3610.3	1251.17	4731.53	52	5372.7	692.85	6013.3	6173.6	6333.65	6493.8	6653.9	6814.35	6974.95	7135.05	7295.3	7455.6	7615.85	7776.05	7936.25	8096.35	8256.35	1416.6	9576.85										

RESULTS CT TENT MO\_\_27
Migh cyale Francis regs 200 Hillion Gycle Hold Times - 10 sec

RESULTS OF TEST NO. 28
High Cycle Frequency: 200 Nz

1						ļ		iļ.	
	CRACK LEMBTH	ا د د	نار د	BROWTH RATE		H1gh C	High Cycle Frequency:	200 Nz	
	E Marie	E . T .		I MCME 3 / SEC		S AOI	Low Cycle Hold Time:	5 Bec	
10.00	233403	20184.3	571.125	4.90UU4E-06					
2000	247:84	20241	307.300 ABC 500	40000000000000000000000000000000000000	1				
7010.04	231449	20201.7	301.454	3.82553E-04	1158	CRACK LENGTH		¥	GROWTH RATE
7210.73	258409	20343.	1748.82	2.77500E-06	SECONDS	INCHES	412	312	INCHES/SEC
12429.4	.245484	20550	2027.91	2.70423E-06	2648.78	.240046	20179.4	749.787	7.74330E-06
14236.4	.270472	20694.7	2077.96	2. F3555E-06	70.070	.245896	20174.1	766.559	8.19528E-04
15042.0	273175	20730.7	2108.57	3.20174E-06	24.04.0	. 251334	20137.5	•	8.55140E-06
1748.	.20121	20843.9	2159.4	3.54126E-04	AU4004	.256387	20267.3	782.697	8.49905E-06
1.0001	286363	20597.9	2193.35	0 - BC048B-0	10.1440 10.440 10.440	748787	1997	789.463	8.61791E-06
20360.3	241671	20094	3157.88	4.0566BE-06	4700 171	150/07	20244	803.888	•
21363.1	-04042	20061.3	3312.70	4.2008/E-00	7298.74	/04C/2	20240.4	CZZ-018	8 38608E-06
24075.4	107604	20021.3	33/7/26	4.110000	9299.19	205110	20531.7	911.10/	7.7003ZE-00
25380.4	31364	20171	3556.58	4.25586F-06	9349.29	300798	20024.	2353.34	8.24692F-06
24565.3	.310621	20350.4	3612.37	4.31478E-04	11204.7	.310807	20273.2	2634.49	4,54244E-06
27890.8	.324578	20135.9	3666.41	4.12126E-06	13040.5	.31660	20428	2693.3	3.27657E-06
29497.8	.331132	20288.3	3737.94	3.93100E-06	14766.3	.32178	20309.6	2761.4	3.25186E-06
32307.1	341345	20370.4	4552.75	4.31815E-06	14371.4	.327097	20190.7	2015.23	3.51192E-06
12711.1	34705	20259.5	4355.14	5.69834E-06	17926.6	.332611	19076.1	2974.32	3.71742E-06
34614.2	.352225	20047.4	4413.06	7.45441E-06	19281.4	.337992	20045.7	3044.12	3.89707E-06
1977	.35745	20211	4422.77	9.26753E-06	20636.7	.343395	1966.2	3158.71	3.95179E-06
7.0140	.363327	20179.5	4496.42	1.0/335E-03	21992	.348923	19799.5	3534.2	4.18675E-06
7.029.7	30724	17724.7	4328.74	1.1447BE-03	23146.	. 35355	19863.9	1268.99	4.70819E-06
34722.3	373383	20454.7	4298.05	1.200078-00	24703.1	.361112	20004.1	4160.64	6.32707E-06
7/3/1/2	20000	1.67707	1001	10-1004CF	2330	.364951	19897.4	4297.7	B.12654E-06
3/843.4	10171	20142.4	8411.78	1.495486-05	2000.8	.370271	20096.6	4266.62	9.83634E-06
10520	146841	20044.8	4764.84	1 .89203E-05	74640	0010/9·	19/93.5	42/9.03	1.11210E-05
# 724	401032	20071.4	4819.02	2.05254E-05	27744	7101/5.	1,10071	44.04.10	1.22304E-05
39029.9	. 409307	20174.6	4671.8	2.20164E-05	27215.4	1000°	E 7000	46/0.54	1.536625103
39330.8	416034	20227.9	4907.95	2.35808E-05	20117 3	397384	19709.7	4482.02	1.57873F~05
7.1054	. 42122	20144.9	49:37.48	2.36545E-05	28468.9	. 404541	19872.4	4629.51	1.55083E-05
1,000	42847	19904.3	5054.33	2.57.008E-05	28769.9	. 409051	19649.5	4649.59	1.52263E-05
	791964	17612.0	21.0110	0014000/P	29373.1	.417111	19593.2	4796.14	1.36610E-05
	461447	10542.4	42.45.78	3.26606E	30524.1	. 431645	20440.7	5414.36	1.66776E-05
40634	459744	19579.2	5386.65	4.10950E-05	30474.4	434204	19859.2	6049.05	2.41597E-05
41036.7	74886	19654.8	5545.32	4.50665E-05	31007.1	101111	8.4007	6021. <b>4</b> /	4:164/46-00
41237.3	.478156	19548.9	5646.13	4.80814E-05	31077.1	448787	19471.9	42.4.150 42.4.100	10-10000V-4
41430	468004	19434.7	5735.92	5.119666-05	31227.4	456238	20144.8	5368.82	
41338.	.493115	19241.2	5772.14	U. 24046F-03	31377.8	. 464868	19849.8	6493	5.76234E-05
	/ * * * * * * * * * * * * * * * * * * *	0.08741	10000	00121007010 1 0114110101	31478	.470301	19589.5	6673.9	5.93147E-05
4.0004	509926	10101	Ses . • S	00-10000000000000000000000000000000000	41578.1	476389	19495	6737.04	6.35132E-05
41939.	514162	19470.6	4117.85	6.75129E-05	31,28.0	. 466253		6839.66	
42040.1	.523435	19485.9	6246.27	7.24997E-05	31928.8	50211	19650.	7014 14	6.30333E-03
42140.3	.531042	16941	4387.59	7.80731E-05	32029	512143	19790.8	7142.33	
42240.7	.538931	19556.7	6394.84	٠	32079	.518288	19545	7201.2	1.68188E-04
42340.7	547512	19412.5	6790.22	9.31693E-03	32129.4	.526682	19193.7	7295.67	2.09250E-04
4.1424	**************************************	1,004.	7072.73	1.028306.104	32179.7	.537159	18970.8	7432.43	2.58190£-04
42442.4	.00/0/0	19307.8	7559.28	1.274586-04	32229.9	.5506	19205	7622.74	3.41643E-04

7
ن

Migh Cycle Frequency: 200 Hz Low Cycle Hold Time: 5 sec

High Cycle Frequency: 200 Hz Low Cycle Hold Time: 5 sec

TABLE C.22 RESULTS OF TEST NO. 31

GROWTH RATE INCHES/SEC	1000000	1.013088-00	1.3/0//6-03	1.37997E-05	1.25640E-05	1.09053E-05	9.76339E-06	8.9329BE-06	8.22917E-06	8.33220E-06	9.39563E-06	1.16632E-05	1.37667E-05	1.61338E-05	1.86552E-05	2.13422E-05	2.52906E-05	2.79442E-05	3.06652E-05	3.35301£-05	3.56431E-05	3.78655E-05	4.06528E-05	4.41215E-05	4.70995E-05	5.02227E-05	5.37809E-05	5.6627BE-05	6.05860E-05	6.52224E-05	7.17634E-05	9.21304E-05	1.00003E-04	1.09941E-04	1.17643E-04	1.25042E-04
HF K		468.443	C*/ 910	1327.95	1365.15	1395.49	2239.28	2327.9	2366.76	2409.42	2668.94	3444.71	4098.53	4267.44	4401.18	4451.28	4559.26	4658.13	4714.49	4750.7	4806.47	4914.85	4965.07	5024.22	5118.24	5229.38	5341.94	5427.09	5496.01	5582.71	5658.76	6020.88	6572.11	6840,15	7173.98	7694.4
LF K PBI VIN		29768	3003	30448.3	30372.2	30149	30963	30826.2	30716.6	31343.1	23008	30334	30034.3	29869.7	30083.3	29684.7	29820.1	29812.9	29922.5	29624.6	29303.2	29083.2	29155.3	29219.7	29334.2	29039.4	28879.1	28620.4	28378.3	28478.2	27996.2	28468.4	29053.5	28627.6	28513.9	27846.9
CRACK LENGTH INCHES		00/00PM	190000	342973	372169	.301229	.389321	89245	. 406783	. 414905	. 422804	. 43293	. 439168	. 446843	.457679	. 465095	.475411	.483282	. 492122	.500603	. 509234	. 516525	. 528346	.539072	.548211	.557866	.568476	.579901	. 588624	.597864	. 407898	. 61383	.630383	.641767	.633391	.679745
TINE SEC		2001.00	72.0/07	3774.34	4485.42	5295.14	6105.77	7018.19	8133.33	9248.61	10262.4	11327.6	11885.3	12545.5	13052.7	13459	13916.6	14220.8	14525.6	14779.8	15034	15288.1	15541.6	15795.8	15999.2	16202	16405.5	16608.3	16761	16913.1	17065.9	17167.6	17319.9	17421.9	17523.5	17726.6
	SHOWIN RATE	A MCMES/SEC	1.52431E-05	1.40442E-05	1.38679E-05	1.290778-05	1.241936-03	1 30673E-05	1 · 1 VOZ4E - 05	1.101876-05	7.8366/E-06	7.176/BE-06	0.31/6/E-06	00-100001	3.20084E-06	3.276705-06	2.48710E-06	3.86281E-06	4.32524E-06	4.97800E-06	3.20176E-06	3.73781E-06	0.82521E-06	7.78010E-06	1./1221E-05	Z-35067E-05	3.3/89/E-05	3.7278BE-05	4.10/702-03	4 - 4 5 KB VE - 03	4.79476E-05	3.16290E-05	3.43949E-05	6.19547E-05	6.82441E-05	7.22722E-05
3	1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1	E	340.203	914.4FF	377.720		103.100	/10.00F	404 404	418 118	412.480	1632 36	1474	10.070	77.70	20.1675	374.44	2300 000	70.00/2	144.767	21/.272	7/7:000	104.707	764.00	7777	4401.72	A774. AR	4074	4040.30	4040 . V	30.00AF	70.190	11.0117	2183.5	3402.64	2000
¥	\ \ \ \ \ \ \ \ \ \ \ \ \ \ \ \ \ \ \		1.906.1	21861.2	2215A. A	22111.4	20221.7	19934.0	20024.0	19987.4	19978.5	19809.1	20180.4	20390.1	20437.B	21010.1	21001.4	21042.4	20144	19936.3	19470.5	10004	19747.7	20310.7	1 0041.T	19989.7	19951.1	20254	19679.1	19777.B	0.000	10756.4	10001	10071	20004.4	
CRACK LENGTH	INCHES	270248	201215	304048	. 41663	. 328192	. 338991	.354705	. 344402	.376936	. 384753	. 393241	. 408799	.419784	. 43298	. 444594	. 454303	.467388	463744	. 491487	.501109	.51004	. 522503	. 534645	.54262	.549394	.559197	840498	.57746	. 586715	.596852	. 604651	.613393	.622978	634079	
TINE	SEC	2534.11	3531.94	5125.49	6022.22	6919.44	7666.46	9058.19	10103.3	1000		13262	16717.9	20327.1	24003.5	28005.6	31052.0	34254.2	37912.2	39458.3	41252.8	42749.3	44495.8	45713.9	46070.1	46374.3	46628.5	46881.9	47085.4	47288.2	47491	47643.8	47796.5	47948.6	48100.7	

TABLE C. 23

RESULTS OF TEST NO. 32

High Cycle Frequency: 200 Hz Low Cycle Hold Time: 5 sec

RESULTS OF TEST NO. 33

TABLE C. 24

2,94116E-05 2,95931E-05 2,490391E-05 2,490391E-05 2,40998E-05 3,12651E-05 3,12651E-05 3,13052E-05 4,18041E-05 4,75454E-05 5,48045E-05 6,86089E-05 1,00388E-04 1,15192E-04 1,53136E-04 GROWTH RATE INCHES/SEC 200 Hz 5 mec High Cycle Frequency: Low Cycle Bold Time: PSI VIN 400057 3 4000957 3 4000957 3 4000957 3 39628 8 39628 8 39628 3 39628 3 39628 3 39620 5 11190.28 11190.28 11190.28 12252.78 22657.64 22657.64 3420.69 1.09281E-05 1.04281E-05 7.70614E-06 6.47261E-06 6.47261E-06 6.47261E-06 6.47261E-06 6.47261E-06 7.81884E-05 7.81372E-05 7.2272E-05 BROWTH RATE INCHES/SEC PSI (IN 4452.934 4457.337 11771.111 11771.111 117777.111 11777.111 PSI (IN 2028.5) 30028.5 30028. LENGTH CRACK LE
INCHES
3758039
3841748
404456
442884
442884
4445898
446435
452423
524233
524233
538165
538165
538165
538165
538165
538165
538165
538165
538165
538165
538165
538165
538165
538165
538165
538165
538165
538165
538165 11hE SEC 32747.2 32746.9 4334 4334 4334 4334 4334 10383.8 10383.8 115034 115034 115036 115036 11602.7 11865.9 118102.7 11866.2 221716.9 221716.9 221716.9 221716.9 221716.9

High Cycle Frequency: 200 Hz Low Cycle Hold Time: 5 sec

High Cycle Prequency: 200 Hz Low Cycle Hold Time: 5 sec

RESULTS OF TEST NO. 35

TABLE C. 26

GROWTH RATE	INCHES/SEC	AC. 7. HEA.	00131000	1.48558E-05	1.48289E-05	PO-SEGOLV	10 100 100 1	1 . 40180£ -05	1.34332E-05	1.21808E-05	1.09363E-05	9.97757E-06	9.18619E-06	8.72959E-06	8.57454E-06	8.30781E-06	8.10419E-06	7.82721E-06	A0-314041 C	20 101 101						
¥	NI VIN		515.53	297.527	326.257	100	1/4:419	339.132	335.925	1100.57	1120.84	1520.39	1545.55	1751.14	1771.08	1678.35	1967.86	1998.74	10.4400	× 100×						
4	NI 7 184		4./9207	20240.8	20314.3		20107	20152.7	20139.7	20758.4	20529.9	20468.7	20238.1	20079.2	20684.3	20704.5	20779.8	21317.8		41914.7						
CRACK LENGTH	TACKES		25022.	.229704	ATRRIC.		717847.	.257422	.266295	.277634	.287598	296608	.307081	.316761	.326537	.336814	346413	AAFSSE		907995						
1116	338		2000.02	2611.59	1328. A7		7.0096	4470.13	5110.4	5991.47	6886.41	7788.17	8924.91	10070	11224.6	12387.2	13558.9	14440		16091						
<b>1</b>	u	93	50		3 :	ņ	50	50	100	9			e e	ľ	ŗ	ŗ	2 2	-	9	2	20	2	7	3	7	*
GROWTH RATE	INCHES/SE	2.23906E-	2.20114E-	2 20114E	100000	Z.13393E-	1.97302E-	1.87303E-	1.792036-	1.74827F-	1.734205-	1.679436-05	1.455486-	1.450456-	1.494406-	- B6222E-	74746		3.23/2/E-	4.46391E-	6.44529E-	8.30216E-	1.095856-(	1.27637E-	1.47433E-(	1.70636E-
<u>}</u>	PSI VIN	572.952	570.943	407 - 406		7.88ZZ	2366.4	2378.97	2432.46	2730.55	2754.37	2862.72	3045.8	1005	1171.42	1304.75	1400 74	76.14.15	70.77	4130.38	5165.45	4008.17	6479	4810.64	7009.25	7465.49
(F R	PSI VIN	39264.7	4. 20001	* *****	7	39846.5	39444.6	39520.7	39414.9	10200	38734.7	38427.3	38310.2	17574.5	17811	17720.4	17077		2027	38897.1	39573.9	38967.3	38116.4	37972.4	37984.6	36834.4
CRACK LENGTH	LACKES	. \$20774	527852	0 0 0 1 1 1	D	. 542783	. 551942	559348	566893	. S74184	410101	586244	0.000	A0210B	41179	A17424	404100	*****	77170	. 638525	. 648211	. 65456	.663386	.649264	.476018	.683478
11ME	360	1199.83	1499.44	1700 47	10.44.1	2156.28	2616.13	3024.81	3433.5	1042.19	4250.86	4711.42	5119.4	5528.70	4000.00	4440.2	1000	10.000	70.017	7369.61	7574.3	7675.95	7778.3	7830.17	7881.34	7932.52

#### TABLE C.27

# RESULTS OF TEST NO. 36 High Cycle Frequency: 200 Hz Low Cycle Hold Time: 2 sec

TIME	CRACK LENGTH	LF K )	) ¥	GROWTH RATE
SEC	INCHES	PSI VIN	PSI VIN	INCHES/SEC
582.251	.52279	36239.4	794.299	6.241B1E-05
975.532	.545328	40318.8	665.92	5.08084E-05
1173.72	.554126	40687.7	646.157	4.54792E-05
1373.04	.560293	40479.3	614.271	3.74973E-05
1787.29	.576002	41010.B	3818.71	3.75249E-05
2000.23	.583788	40642.2	4023.53	3.59086E-05
2429.35	. 598895	40870.7	4199.52	3.46251E-05
2863.25	. 599264	39471.7	4258.6	5.857675-05
3301.49	.629442	41145.1	4559.42	9.51374E-05
3743,34	47A014	44868.0	5044.74	AC TACAGE

I ABI E 'N

RESILENCY TROUGH (100 Hz. Bank CV. Le Holls Comm. 100 Hz. Bank CV. Le Holls COM. 100 Hz. Bank CV. Le Holls CV. Bank CV. Bank

		GEOUTH RATE	INCHES, SEC	14865F-05	1.45503E-05	1.517386-05	1.49006E-05	1.41691E-05	1.332426-05	1.253625-05	1,20062E-03	1.138626-03	1.07.3000	1.00370E 03	00.000 00 00 00 00 00 00 00 00 00 00 00	A.07910E-06	7.88914E-06	6.26715E-06													INCHES/SEC	5.60387E-06	5.81127E-06	5.32295E-06	4.44742F-06	2.37025E-06	1.95697E-06	3.14442E-06	5.89327E-06	1.32590E-05	3.73400E-03	0.07/0/10.0				
	200 Hz 180 sec	i L	Por A	200	307.702	144.700	307.132	312.66	331.125	402.849	331.785	417.871	1143.38	1113.61	1128.57	17/4:03	10.1121	2595.84							•	40	200 H2			:	PSI VIN	1988.55	2069.45	2217.8	7757 01	1116.94	3456.37	4356.08	5008.89	5310.19	5500.3	37.35.76				
	High Cycle Freugency: Low Cycle Hold Time:		1 to 1 to 2 to 2 to 2 to 2 to 2 to 2 to	KT 1 101	20318.6	20140.3	701/8:	20025	20024.3	19962.3	19729.7	19794.2	19917.2	19930.5	19964.1	19941.2	14445.1	1001	•					TABLE C.31		RESULTS OF TEST NO. 40	High Cycle Frequency:	Low Cycle Hold Time:			TH LF K PSI JIN	20803	20791.8	20952.9	20826.2	19397.9	19413.7	199.4	199 10.7	19820	20564.6	20175.4				
ı	H1gh Low	1	CRACK LENGTH		.227391	.235284	.247346	174557	759507	.282697	.291224	.300104	.308193	.317269	.326865	.337612	.345934	3216/2	. 30200								H	Low			CRACK LENGTH INCHES	.224916	.232454	.244775	.254731	.350457	17881	792767	.405611	.415805	,423876	, 428343				
			TIME	<b>3£</b> C	3061	3601	4321	4861	1040	0121	78.47	8282	9002	9903	10984	12066	12967	13688	15310												TIME SEC	4862.16	6124	8466.53	10628.8	27383.6	33149.8	64.000.0	48466.4	49367.7	49728.1	49908.2				
		GROWTH RATE	1.28245F=05	1.22103E-05	1.159446-05	9.77332E-06	8.72546E-06	8.21591E-06	8.06245E-06	6.15975E-06	8.73030E-06	1.047036-03	1 - 2 0 4 4 4 K   0 5	1.101001-0.1 1.1014001-0.3	1.741815-03	1.92647E-05	2.07186E-05	2.30221E-05	2.60315E-05	2.96901E-05	3.46969E-05	4.05215E-05	4.555/E-03	5.091586-05	5.57455E-05	5.94511E-05	6.45650E-05	7.07119E-05	8.92416E-05	1.01465E-04								0 + 4 · 1 · 1 · 1 · 1 · 1 · 1 · 1 · 1 · 1 ·	CROMPIN AND FINE TO THE TRANSPORT OF THE		4.1665581100	4:1/14/E-05 4:24424F-05	4.15735E-05	3.96830E-05	3.83825E-05	1.1387.1
÷	:	HF K	205.125	108. 180	310.353	1280.97	1308.29	1003.53	1109.83	1307.28	1364.53	1396.94	3300.20	10.41.01	37.22.47	4210.14	4321.05	4435.76	4285.71	4211.27	4918.64	5059.39	5156.63	5388.45	5530,66	5739.16	5957.54	6148.06	6684.8	7193.5				NO. 38		: 200 Hz	) SG 7	1	PST ~IN		746.57	704.079	2261.8	2377.18	3930.93	07.00
High creite Frague C.	Low CV. Le Boll & Claric	PSI VIN	30417 8	20404	20450.2	20012.3	19948.2	20138.7	20392	20371.8	20517.5	20686.8	20782.2	20//07	20/72.7	21014.8	20778.5	21047.5	20623.3	20727.8	20633	21452.1	21319.1	21336.7	21601.1	21418.7	21362.9	21953	21367.3	21432.5		TARLE C. 29	110	RESULTS OF TEST IN		High Cycle Frequency:	cycle note time		7 24 12 24		39136.9	11040	37148.4	37049.2	36778.2	23404.
48.4	Total Control	CRACK ! ENGTH INCHES	201.43	20007	219856	217148	244773	. 254139	.263076	.271473	. 280698	908067	.300853	\1860£	. 3202y4	140045	1300 KB	360387	.370187	.381638	.394916	.410228	.419416	42887	450148	.46262	.475961	.489891	.522147	.541936				(		H 18	<b>*</b> 07		THENETH THENETH		.65165	1100.	.681145	.690337	.699051	./1/434
		TIME	2404.93	7017.22	3835.74	8477 10	A316.86	7442.93	8576.71	9716.42	10662.7	12016.7	12046.3	13649.2	14357.5	130/0.7	14041	16510.2	16936.1	17361.9	17787.7	18213	18426.5	18639.1	19975.1	19278.8	19491.8	19704.7	20130.7	20343.7									¥1.	; ;	850.736	70.00.1	1523.46	1760.19	2000.68	74.7.47

~
TABLE

3	
1	
$\sim 1$	
<b>₽</b>	
- 1	
-	
201	
TEST	
⊢l	
. 1	
<u>e</u> /	
- I	
)	
vΩ	
5	
٠,١	
⊋)	
إزب	
91	

High Cycle Frequency: 200 Hz Low Cycle Hold Time: 180 sec

RESULTS OF TEST NO. 43

TABLE C. 34

High Cycle Frequency: 200 Hz Low Cycle Hold Time: 180 sec

GROWTH RATE Inches/SEC	1.555146E-05	2.08054E-05 7.54490F-05	2.98083E-05	6.104049F-000 6.77070F-000 6.09064F-000	4.42672E-05 4.87286E-05	5.03253E-05 5.11316E-05	
HF K PSI VIN	4426.15	4756.66	4937.99	5265.25	5699.63	6322.17	
PSI JIN	16069.6	16166.7	15640.6	15890.2	16682.7	16533.5	
CRACK LENGTH INCHES	.270508	.29243	.309112	. 334067	.363509	.415276	
11ME SEC	6276.52	8069.9 8428.43	8786.94 9145.48	9504.62	10221.7 10938.8	11298 11477.3	
חדויים מין	INCHES/SEC	2.01955E-05 1.90932E-05	2.06432E-05 1.85947E-05	1.28805E-05 7.95625E-05	1.05250E-05 1.6859E-05	2.54616E-05	5.09255E-05 6.41874E-05 7.89586E-05
<u>T</u>	510.935	551.435	2149.6	2565.89	2915.88 3772.74	4457,26 4832,91	5614.28 5614.85
						30861.4 29904.7 29781.4	1.5.54
CRACK LENGT	INCHES . 456555	.472873	.524105	.534229	.568245	.57724 .586283 .600734	.620827
						21098.5	

### TABLE C.35

# RESULTS OF TEST NO. 44

High Cycle Frequency: 200 Hz Low Cycle Hold Time: 180 sec

	104 Cacle Mold Tites:	100 001						
CRACK LENGTH LF K HF K GROWTH RATE 7165.76 INCHES 951 J4928 354051 14928 354051 14928 355051 15671.7 4422.95 3570816 15591.1 4518.02 37791.5 370816 15591.1 4518.02 37791.5 3781.5 1558.9 37791.5 15923.2 4763.47 378279 15739.4 4801.52 2.67585E-05 11110.6 37279.4 37279 15739.4 4801.52 372700E-05 11110.6 37270.6	con cycle motor time:	sec nor		TIME	CRACK LENGTH	_	Ŧ	GROWTH RATE
CRACK LENGTH LF K  INCHES  14428  14528  14528  14528  15671.7  16671.7  16				SEC	INCHES	•	PSI	TACHES/SEC
INCHES PSI (IN PSI (IN INCHES/SEC 9241.4) 346051 14928 4321.8 5.35195E-06 9779.17 359031 15671.7 4429.95 6.89649E-06 9137.64 3790816 155791.1 45180.02 11.67271E-05 9676.08 379115 15968.9 4607.83 1.67673E-05 11310.6 379115 15973.2 4763.47 2.32200E-05 11310.6 37927 15739.4 48031.5 2.32300E-05 12903.6 403837 16227.6 4908.9 3.23090E-05 16403.1 4379687 16400.2 4648.15 3.44885E-05 16489.6 453359 16400.5 5386.42 3.48721E-05 16489.6 453359 16400.5 5551.35 4.07448	_	¥	GROWTH RATE	7165.76	.513305		470 567	A TOP TOP TOP TO
346051 14928 4321.8 5.35195E-06 9779.17 339031 15671.7 4429.95 6.89449E-05 9137.64 379015 15591.1 4518.02 1.07271E-05 9576.08 379015 15968.9 4607.83 1.69453E-05 10393 38219 15923.2 4763.47 2.322000E-05 10310.6 39279 15739.4 4801.32 2.99493E-05 12186.2 39279 16209.7 5680.98 3.23990E-05 15903.6 427049 16402.7 4948.15 3.44895E-05 164131.1 439787 16340.1 5251.35 4.07448E-05 16489.6 453359 16489.5 5551.35 4.07448E-05 16688	_	Per Sty	CUCY CLITCHA				101.020	8.3342E-00
3346051 14928 4321.8 5.35195E-06 8779.17 4359.03 155971.7 4429.95 6.895495E-06 9137.64 370816 15591.1 4518 0.2 11.67271E-05 9575.04 370816 15968.9 4607.83 1.69453E-05 11110.6 15933.2 4763.47 2.32200E-05 11110.6 15933.7 16227.4 49031.2 2.45285E-05 12903.6 40383.7 16227.4 4908.15 3.23090E-05 12903.6 45340.1 5521.3 3.44885E-05 16489.6 16310.4 453359 16480.5 5386.42 3.83525E-05 16489.6 16489.6 40348.7 150349.7 16489.6 1648	•	E1. 10.	INCHES/ SEC	841.41	.523347		2606.96	1.35011E-05
.359031 15671.7 4429.95 6.89649E-06 9137.64 .370816 15591.1 4429.95 6.89649E-05 9756.08 .370816 15951.1 4607.83 1.67425-05 10393 .377115 15961.2 4763.47 2.32200E-05 12186.2 .377116 15739.4 4833.52 2.6783E-05 12186.2 .403837 16227.6 4900.5 2.99493E-05 12903.6 .415136 16209.7 5080.98 3.23090E-05 15400.6 .427049 16402.7 4948.15 3.44895E-05 164131.1 .453359 16420.1 521.48 3.64271E-05 16489.6 .453359 16420.1 521.35 4.07448E-05 16668.8		4321.8	5.351955-04	8779.17	. 529613		2869.61	1.556825-05
370816 15591.1 4518.02 1.07271E-05 9676.08 1570816 15591.1 4518.02 1.07271E-05 10393 1.094315 15927.2 4607.83 1.69453E-05 10393 1.0943279 15739.4 4831.2 2.32200E-05 12186.2 12186.2 463837 16227.6 4900.5 2.99493E-05 12903.6 427049 16409.7 4948.15 3.48885E-05 16431.1 453339 16489.6 16489.6 16489.6 463339 164399 16489.6		4420.05	4.00440E-04	9137.64	.536767		2027.82	1 110405
379115 15948.9 46010.0 1.694516-05 10393 1.694119 15958.9 46010.0 1.694516-05 11110.6 15973.2 4763.47 2.322006-05 11110.6 15973.2 4763.47 2.322006-05 11110.6 15973.1 15973.2		000	1000	9676.08	.547145		10.00	00-U-00-00-00-00-00-00-00-00-00-00-00-00
3379115 15948.9 4407.83 1.674518-05 1735 384119 15923.2 4743.47 2.322008-05 12186.2 39279 15739.4 4811.52 2.675858-05 12186.2 403837 14227.6 4900.5 2.994938-05 12903.6 415134 14209.7 5080.98 3.230908-05 13620.6 427049 14402.7 4948.15 3.448958-05 16489.6 453359 14480.5 5384.42 3.839258-05 16489.6 467478 14490.7 453359 4.07448-05 16688.8		70.0164	1.0/2/1E-03	10101			7/10/75	70134/440.1
384119 15923.2 4763.47 2.32200E-05 11110.6 39279 15739.4 4831.52 2.47586E-05 12963.6 403837 16227.6 4900.98 3.23090E-05 15903.6 415136 16209.7 4948.15 3.44885E-05 16131.1 427049 16400.1 5251.48 3.44891E-05 16489.6 453359 16430.1 5251.33 44771E-05 16489.6		4607.83	1.69453E-05	7070	A/0/00:		3362.06	1.39910E-05
.39279 15739.4 4831.52 2.67585E-05 12903.6 463837 16227.6 4900.5 2.99493E-05 12903.6 12903.6 441513.6 16209.7 5080.99 3.23090E-05 15520.6 16402.7 4948 3.44885E-05 164131.1 453359 16480.5 5386.42 3.8371E-05 16489.6 16489.6 47748E-05 16668.8 1640.8 470.8 470.8 483.5 470.8 670.8 16489.6 470.8 670		4763.67	2.32200F-05	11110.6	.565103		3421.96	1.25462E-03
.403837 14227.6 4900.5 2.99493E-05 12903.6 .415136 14209.7 5080.98 3.23090E-05 13620.6 .427049 14402.7 4948.15 3.44885E-05 16131.1 .439987 14340.1 5251.48 3.44871E-05 16489.6 .453359 16489.5 5386.42 3.83925E-05 16688.8		4831,57	2.67585F-05	12186.2	.877035		3630.83	1.48081E-05
-415136 16209.7 5080.98 3.23090E-05 15520.6 -427049 16402.7 4948.15 3.44891E-05 16131.1 -439897 16440.1 5251.68 3.64891E-05 16489.6 -453359 16429 5351.33 4.7448E-05 16688.8		4900.5	2.09403F-05	12903.6	.587697	•	4112.72	2.11307E-05
.427049 16402.7 4948.15 3.44885E-05 16131.1 .439987 16340.1 5251.68 3.64571E-05 16310.4 .453359 16480.5 5581.48 3.83925E-05 16489.6 .467475 16429 5551.35 4.07448E-05 16668.8		5080,98	3.23090F-05	13620.6	.604154		4439.72	1.864225-05
.439987 16440.1 5251.68 3.64571E-05 16310.4 .453359 16480.5 5386.42 3.83925E-05 16489.6 .46747 16429 5551.35 4.07448E-05 16668.8		4948.15	3.44885E-05	16131.1	.646879	41929.3	4890.97	4.000566-05
-453339 16489.5 5386.42 3.83925E-05 16489.6 -467475 16429 5551.33 4.07448E-05 16668.8		5251.68	3.645715-05	16310.4	.654503	-	5138.16	5,531,355-03
.467475 164329 5551.35 4.07448E-05 16668.8		5386.42	3.83925E-05	16489.6	.662041	-	5254.83	7.18609E-05
TO STATE OF THE PROPERTY OF TH		5551,35	4.0744BE-05	16668.8	.677036	•	5351.43	8.90449E-05
00:500C 5:00/07 5070t.		5663.06	4.33939E-05					

High Cycle Frequency: 200 Hz Low Cycle Hold Time: 2 sec

High Cycle Frequency: 200 Hz Low Cycle Hold Time: 10 sec

					TIME	CRACK LENGTH	LF K	HF K	GROWTH RATE
TIME.	CRACK LENGTH	ريا د	Z.	GROWTH RATE	SEC	INCHES	PSI IN	PSI IN	INCHESSEC
360	INCHES	Pg1 IN	PSI IN	INCHES/SEC	5461.17	.331421	29549.9	490.83	4.58923E-00
94.11.25	147844	1045H	484.944	7.71884E-06	10114.9	.349872	29645.8	1151.54	3.65665E-00
11242.7	1000 E	30454.2	498.092	7.54679F-06	12366.4	.357237	29927.9	1191.84	3.37474E-06
13344	AF8781	10415	1840.42	2.2227F-06	15232.3	.36623	29790.2	1184.59	3.23888E-06
1738.7	104701	30200.2	1987.4	5.42116F-06	18099.3	.37468	29804.5	1608.21	3:19975E-06
14101.0	40400	10264.7	2164.03	6.38648E-06	20863.6	.383359	29969.8	1596.01	3.25549E-06
14031.8	.411551	30124.3	2621.05	6.42596E-06	23014.1	.390238	29730.4	2384.34	3.6473:E-00
17914.3	42175	29734.5	2685.74	7.1698E-06	25370	750995.	29663.7	2461.62	4.66809E-06
19172	432534	29567.9	3388.67	8-14-46E-06	27827.8	.412084	30160	2636.87	7.72139E-06
20010.8	439094	29523.5	3607.5	9.36793E-06	28749.8	.41973	29584.8	3008.95	1.26483E-05
21058.4	.449634	29443.3	3675.86	1,154816-05	29364.3	.427309	29254.6	4083.25	1.96343E-05
21897.4	459458	29595	3840.79	1,37587E-05	29774.4	. 435027	28971.4	4161.16	2.56312E-05
22526.1	.468197	29333.7	4155.86	1.57161E-05	30081.7	.443211	29231.8	4358.64	2.95346E-05
23155.1	478834	29249.5	4280.29	1.77438E-05	30388.9	. 453052	29918.4	4422.33	3.35075E-05
23783.7	490936	29235.4	4364.68	1.97976E-05	30594.3	.460127	29014.1	4519.87	3,537486-05
24203.5	409381	29276.3	4508.81	2.11949E-05	30901.5	.471618	29330,3	4669.44	4.30391E-05
24623.2	508352	29213.1	4589.21	2.29015E-05	31106.3	.480364	29208.1	4725.68	4, 20667£-05
25042.2	519167	29072.4	4637.84	2.49247E-05	31311.2	. 469113	29701.1	4819.65	4.50001E-05
25441.0	1000A	20849. A	4801.47	0.200828-00	31516.4	.498612	29954.6	4894.85	4.87157E-05
25881.4	3,000	28594.5	4946.53	2.92414F-05	31721.3	.508798	28504.7	5029.51	5.23991E-05
26301.3	553586	28804	5051.03	3.157946-05	31926	.519764	28770.8	5074.79	5.72547E-05
26720.7	. 367326	28673.4	5151,52	3.41831E-05	32130.9	.531742	29543.9	5316.53	6.280a1E-05
27140.5	582096	28551.5	5317.62	3.67852E-05	32335.8	.545126	28954.1	5524.83	7,042916-05
27350.3	589866	28566.9	5415.38	80-800-80 M	32541.1	. 560165	29985.7	5636.9	7.84051E-05
27560.3	598148	28556.5	5502.31	4.01892E-05	32746.1	.576952	28480.3	5880.48	8.74923E-05
27770	404729	28516.2	5514.7	4.214415-05	32848.6	. 586211	28966	6026.09	9.20887E-05
27979.A	415448	28403.2	3687.94	4.19207E-05	32951	. 595864	28805.6	6204.64	9.73516E-05
28189.4	.42505	28492.B	: 761.75	4.62213E-05	33053.5	.606048	29297.3	6315.1	1.033506-04
					33155.9	.616676	28853.7	6482.96	1.08053E-04
					33258.3	. 628268	28520.9	6596.82	1,15626E-04

# DATE FILMED - 2-85